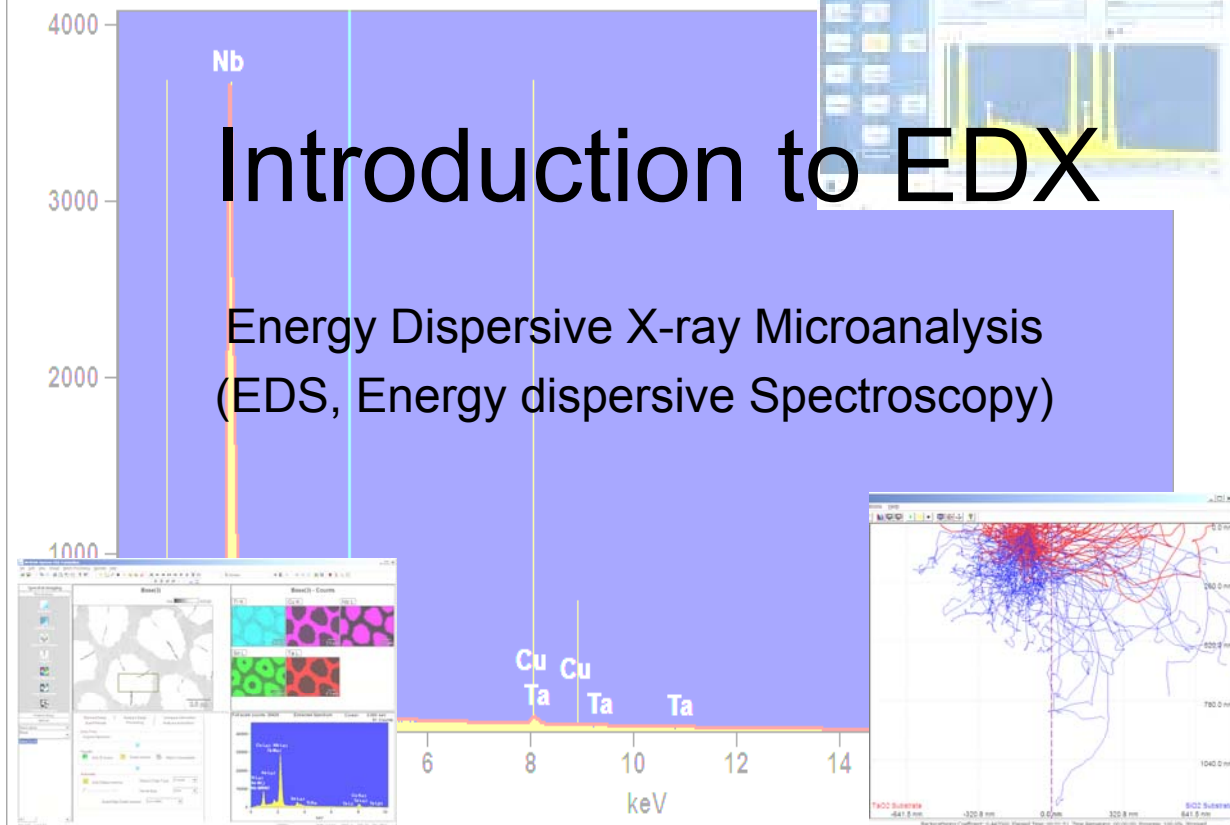


Full scale counts: 3678

Base(2)_pt2
Synthetic Spectrum



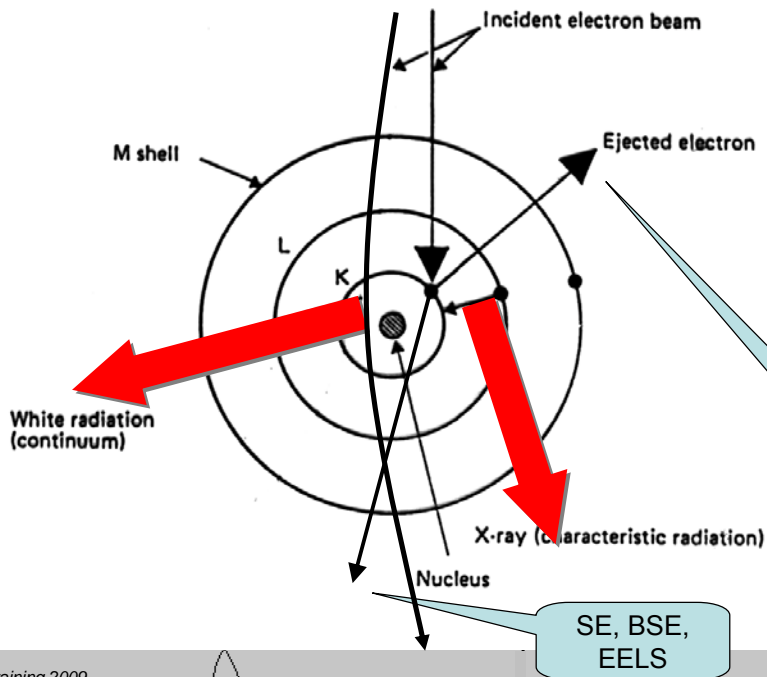
Basics of EDX

- a) Generation of X-rays
- b) Detection
Si(Li) Detector, SDD Detector, EDS (\leftrightarrow WDS)
- c) Quantification
EDX in SEM, Interaction volume
Monte-Carlo-Simulations
EDX in TEM

[d) EDX in SEM-STEM]

Inelastic scattering of electrons at atoms

$$E_{\text{electron_in}} > E_{\text{electron_out}}$$



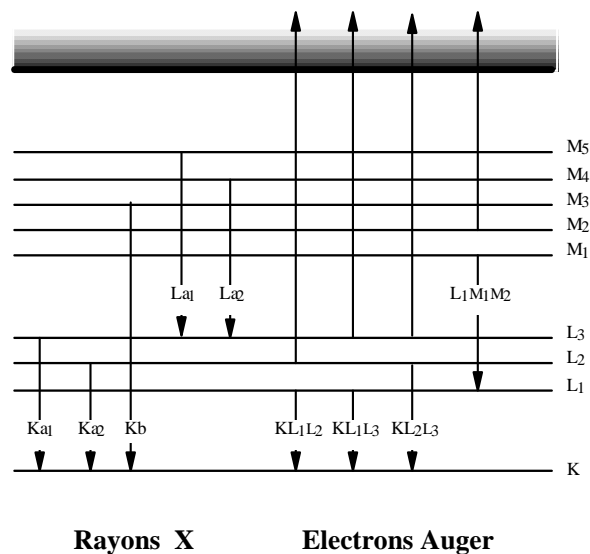
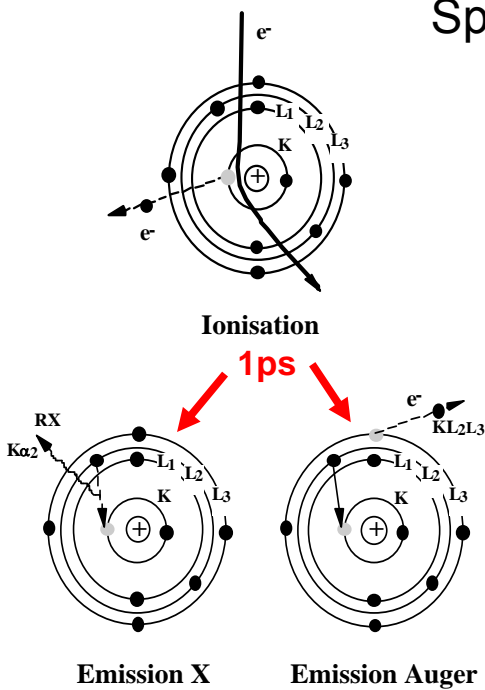
• **Continuum X-ray production (Bremsstrahlung, Synchrotron)**

• **Inner shell ionization**

SE

SE, BSE, EELS

Core shell ionisation: chemical microanalysis by X-ray, Auger electrons and Electron Energy Loss Spectrometries



Rayons X

Electrons Auger

Emission of characteristic X-ray and Auger electron

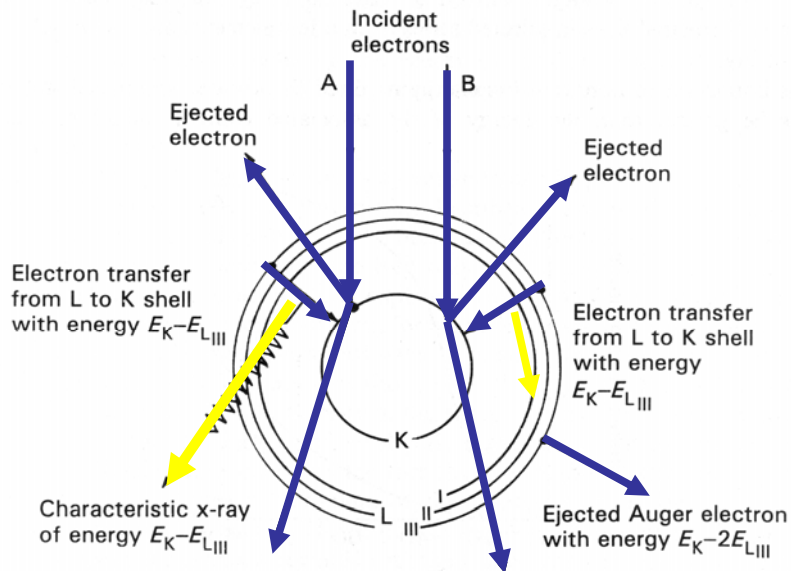


Fig. 2.3. Schematic diagram showing emission of characteristic x-ray by electron A and emission of Auger electron by electron B.

Designation of x-ray emission lines

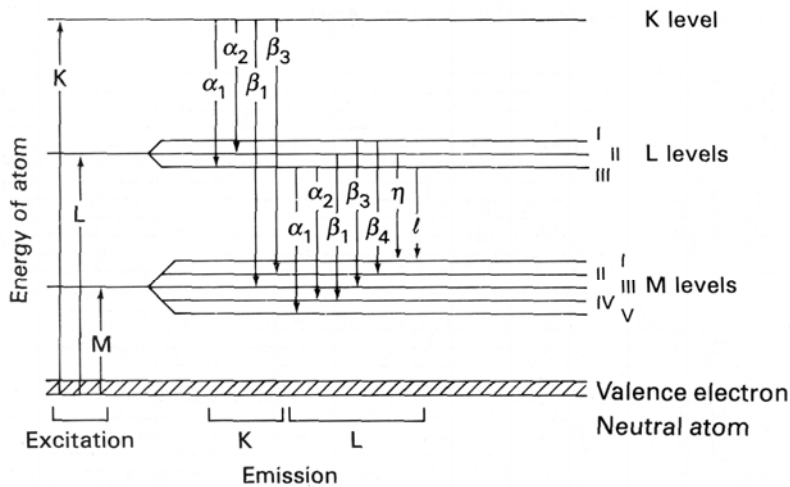


Fig. 2.2. Schematic diagram showing common x-ray emission lines with their designation for an element with atomic number (Z), where $29 < Z < 37$.

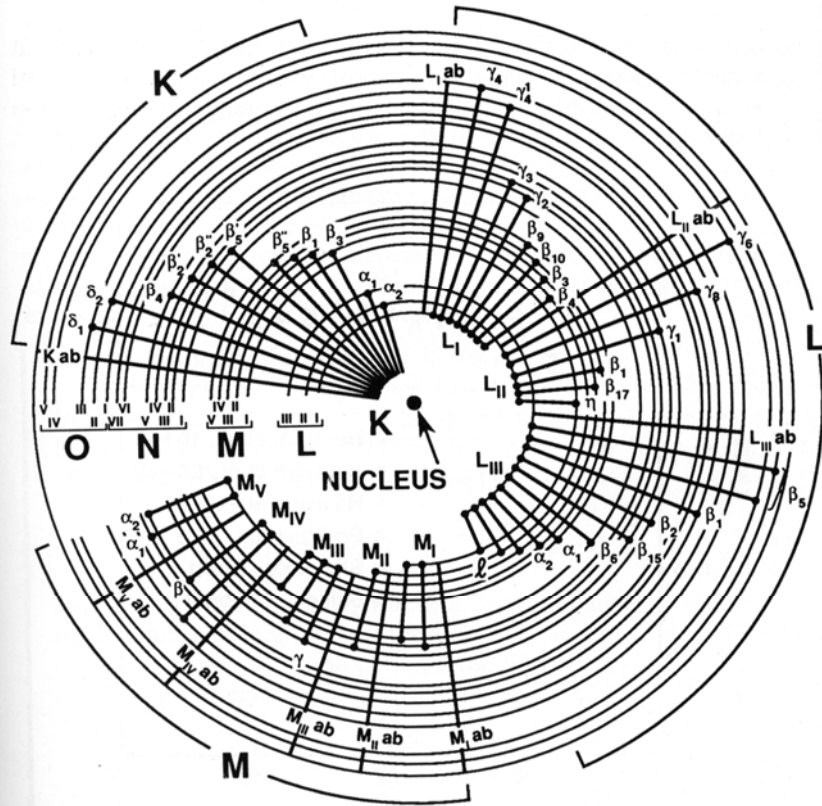
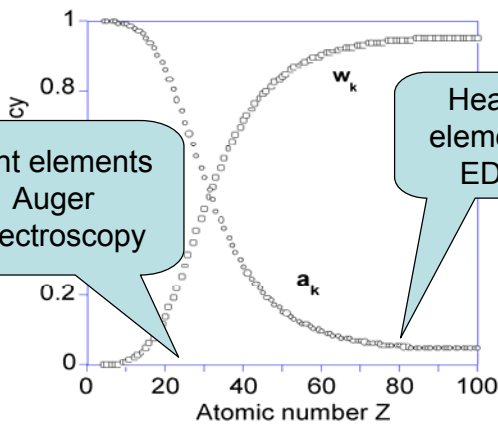


Figure 3.37. Comprehensive energy-level diagram showing all electron transitions which give rise to K, L, and M x rays (Woldseth, 1973).

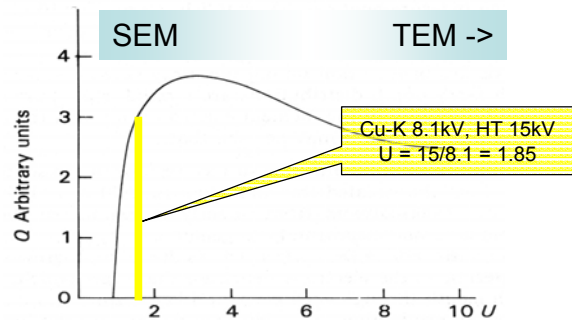
Efficiency of X-ray generation

Relative efficiency of X-ray and Auger emission vs. atomic number for K lines



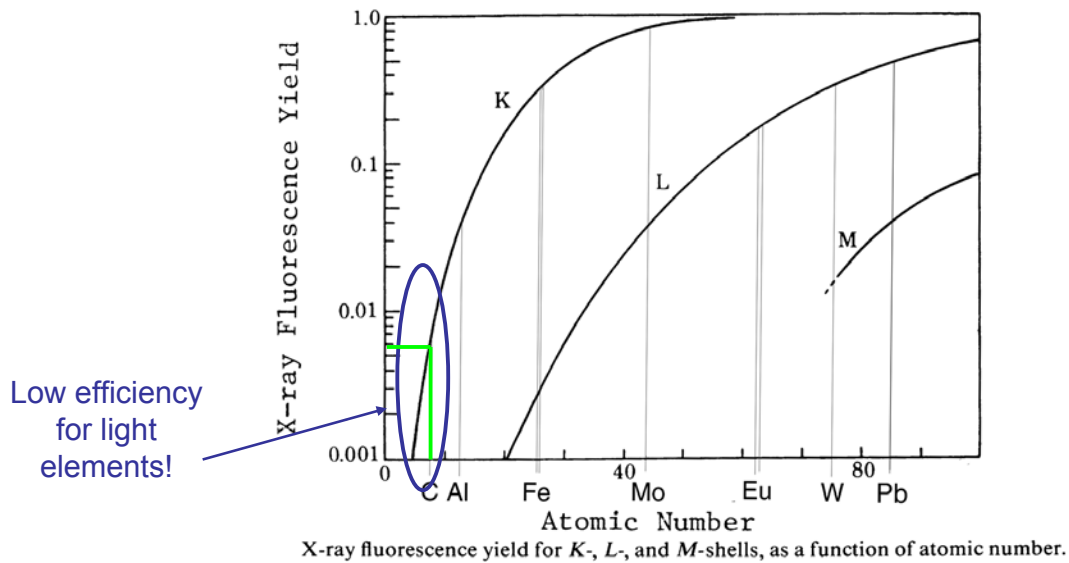
Light element atoms return to fundamental state mainly by Auger emission. For that reason, their K-lines are weak. In addition their low energy makes them easily absorbed.

Ionization cross-section vs. overvoltage $U = E_0 / E_{edge}$
(electron in \rightarrow X-ray out)



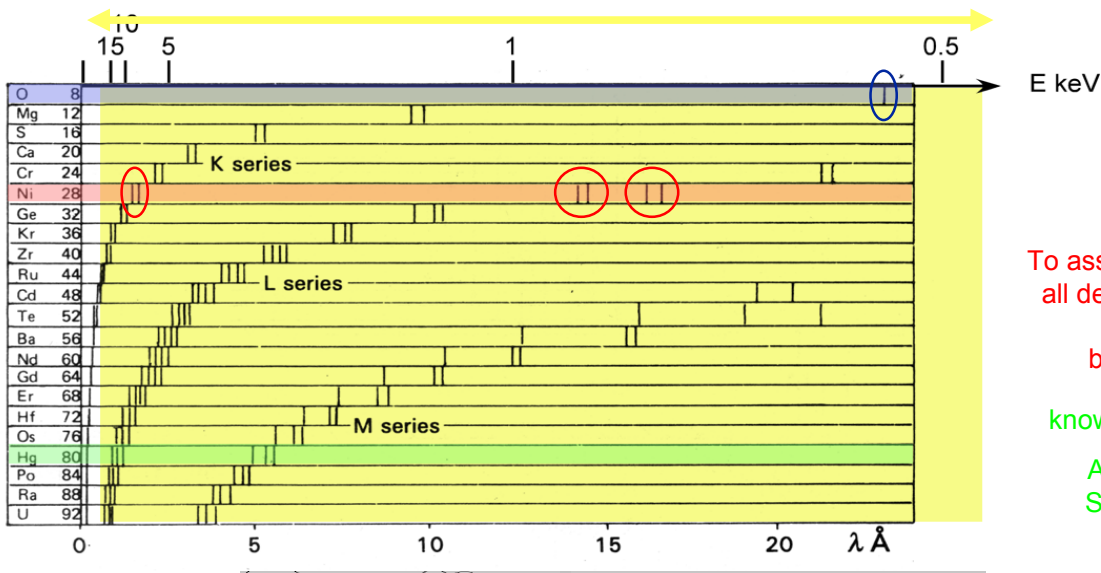
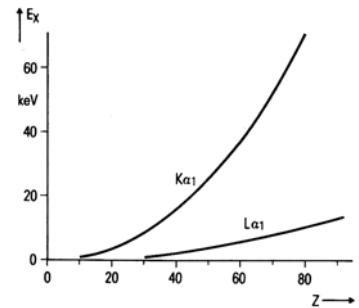
To ionized the incident electron MUST have an energy larger than the core shell level $U > 1$. To be efficient, it should have about twice the edge energy $U > 2$.

X-ray production vs. atomic number Z



Characteristic lines: Moseley's Law

EDS range ~ 0.3-20 keV



To assess an element
all detectables lines
MUST
be present!!!

known ambiguities:

Al $K\alpha$ = Br L1
S $K\alpha$ = Mo L1

Moseley's law for K-series

Frequency ν of X-rays emitted from K-level vs. atomic number

$$\nu = 2.4810^{15} (Z-1)^2$$

$$E = h\nu \text{ et } \lambda = c/\nu$$

with the Planck constant: $h = 6.626\,068\,76(52) \times 10^{-34} \text{ J}\cdot\text{s}$
and $1\text{eV} = 1.6 \cdot 10^{-19} \text{ J}$

Energy of characteristic X-ray
-> Element
Qualitative EDX-Analysis

So, lets measure the X-rays emitted from my sample and determine the composition !
But how to detect it ?

b) Detection of X-rays (EDX)

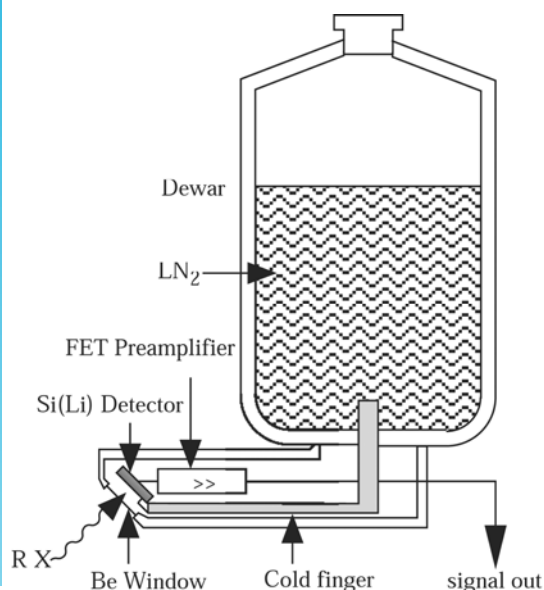
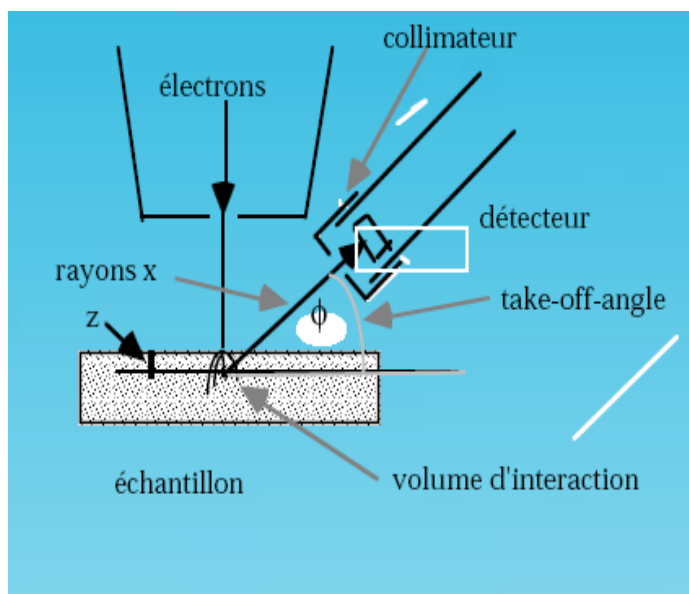
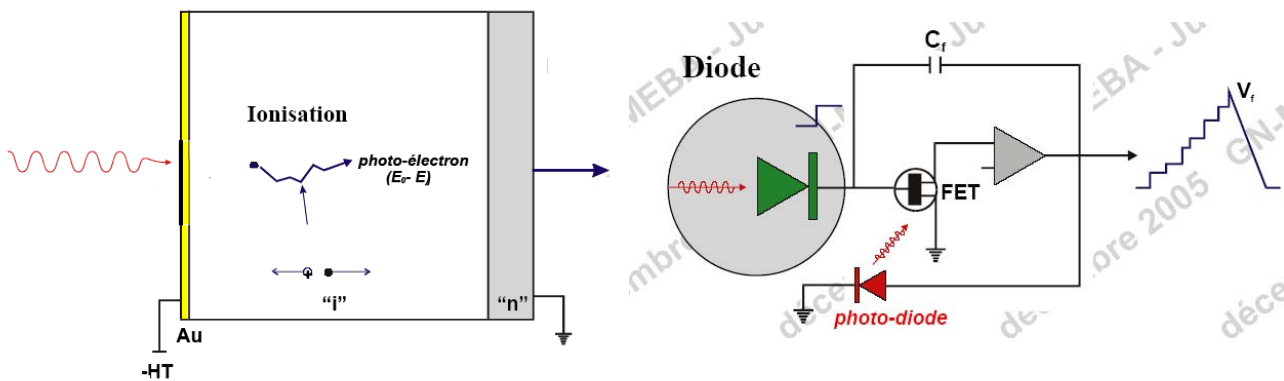
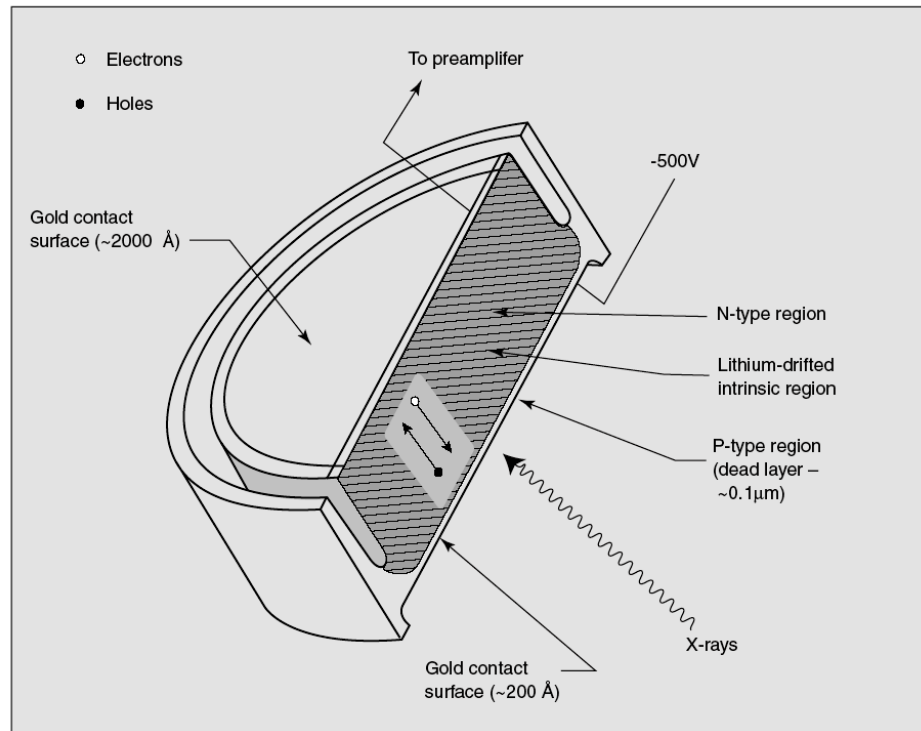
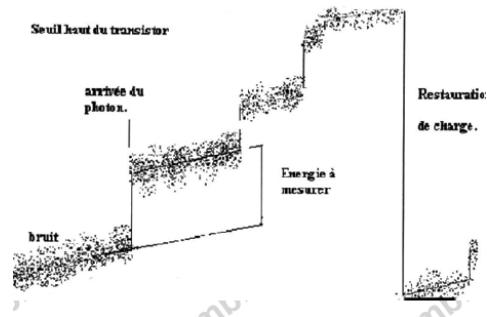


Figure 4-2. Cross section of a typical lithium-drifted silicon detector. X-rays create electron-hole pairs in the intrinsic region of the semiconductor; these charge carriers then migrate to the electrodes under the influence of an applied bias voltage.

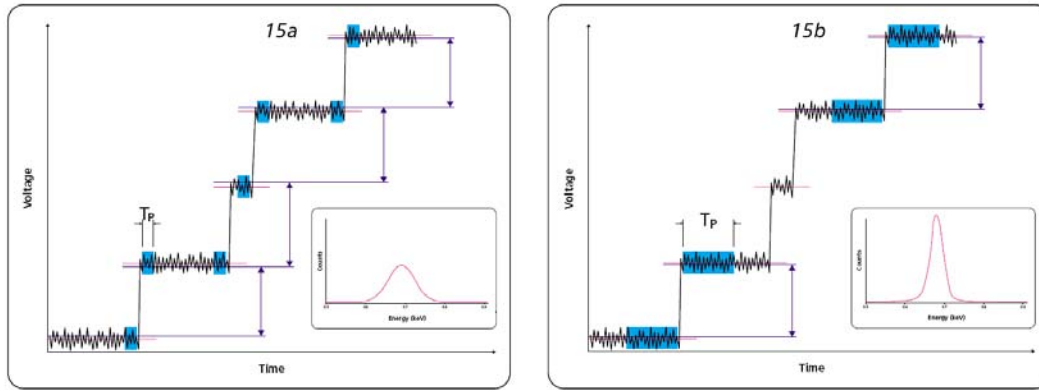


X-Ray energy conversion to electrical charges:
3.8eV / electron-hole pair in average
 electronic noise+ imperfect charge collection:
130 eV resolution / Mn Ka line

- Detector acts like a diode: at room temperature the leak current for 1000V would be too high !
- The FET produces less noise if cooled !
- Li migration at room temperature !
- ->Detector cooling by L-N



Pulse detection and analysis



Pulse detection: Charge ~ energy

Shorter time constant = process time to analyze voltage
 -> peak broadening (lower energy resolution)

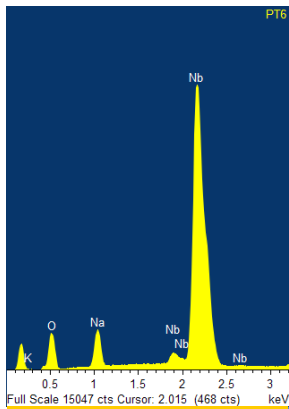
Longer time constant (higher energy resolution)
 -> pulse rejection (dead time)

Process time constant

resolution
peak identification



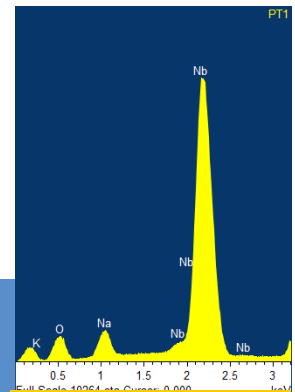
beam current
count rate



Long

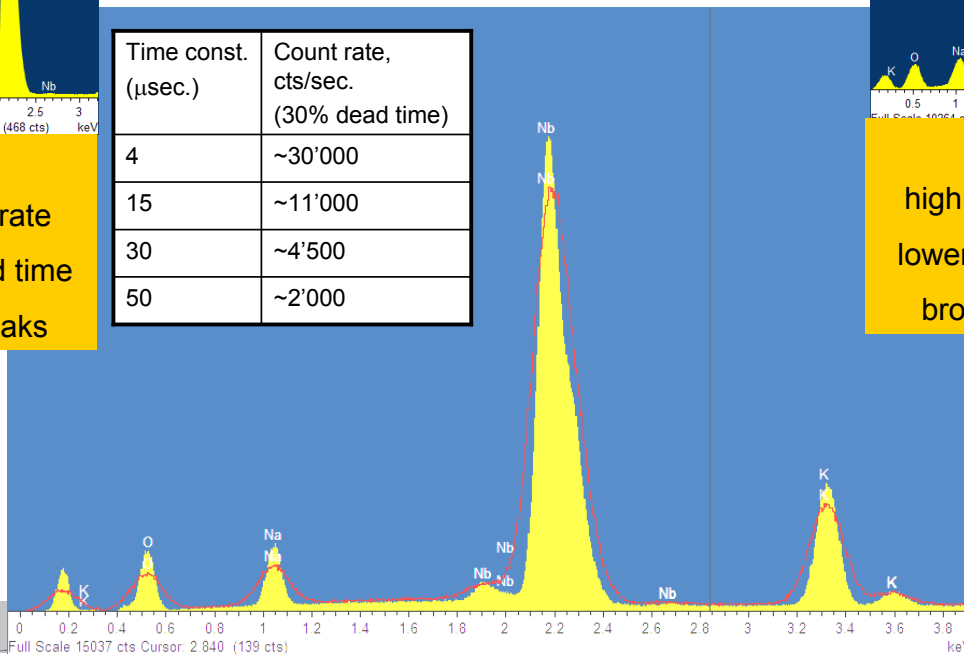
low count rate
Higher dead time
narrow peaks

Time const. (μ sec.)	Count rate, cts/sec. (30% dead time)
4	~30'000
15	~11'000
30	~4'500
50	~2'000

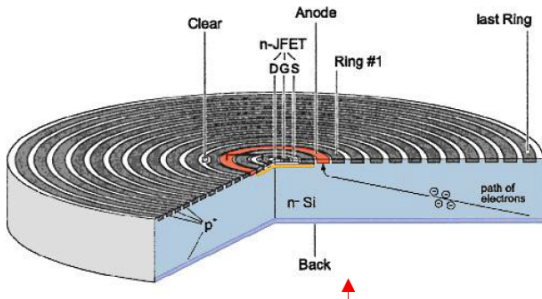


short

high count rate
lower dead time
broad peaks

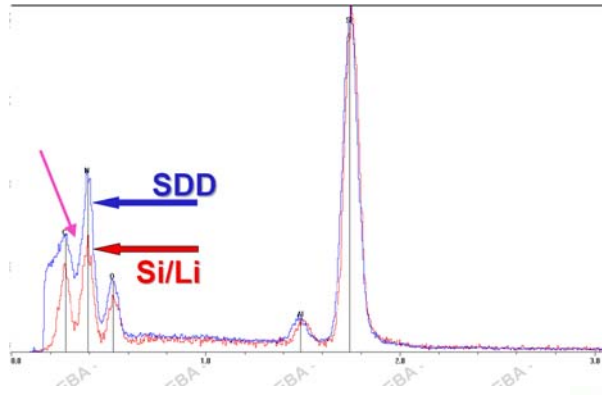


Silicon Drift Detectors



Extremely fast (up to 100'000 counts/sec.)
 No L-N cooling required
 Similar priced as Si/Li detectors

Peak tail at lower energies
 Lower resolution for light elements



Detection and artifacts

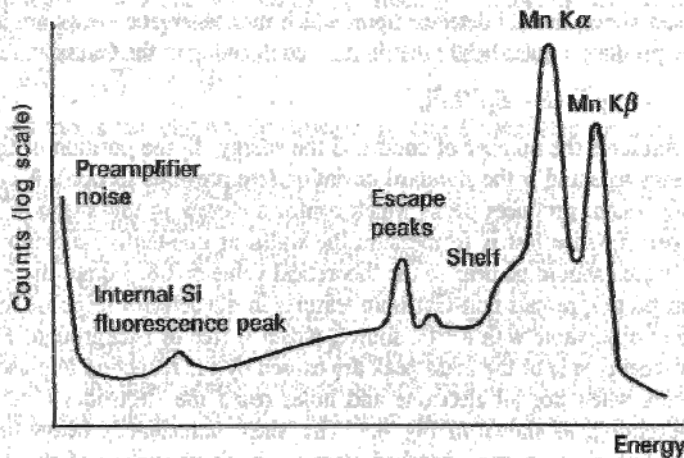
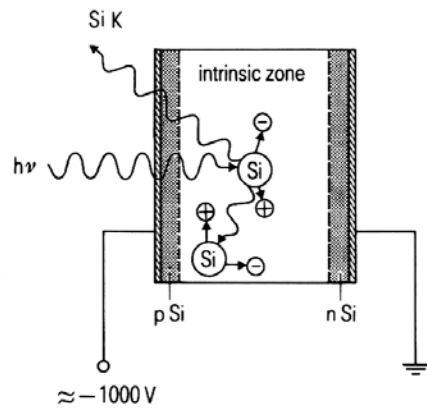


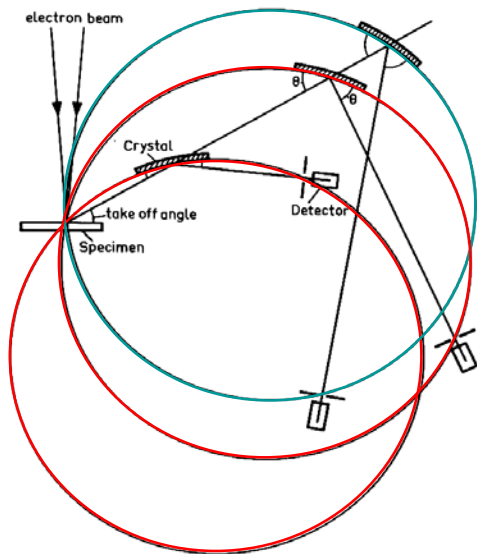
Fig. 4.9. Detector response for Mn K x-rays from ⁵⁵Fe source.



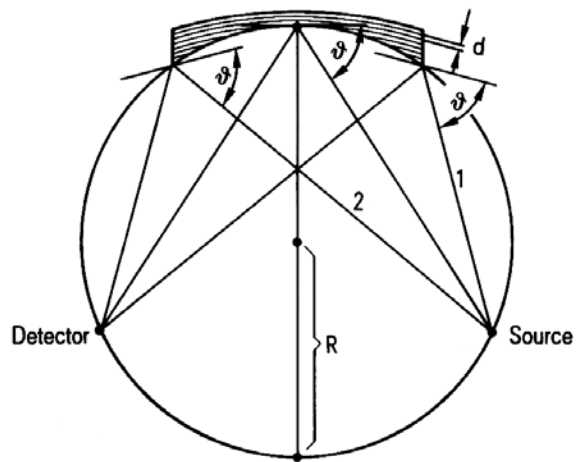
X-Ray energy conversion to electrical charges:
 3.8eV / electron-hole pair in average
 electronic noise+ imperfect charge collection:
 130 eV resolution / Mn Ka line

Take care when looking for "trace" elements (low concentrations). Don't confuse small peaks with escape peaks!

Wavelength Dispersive Spectroscopy (WDS)

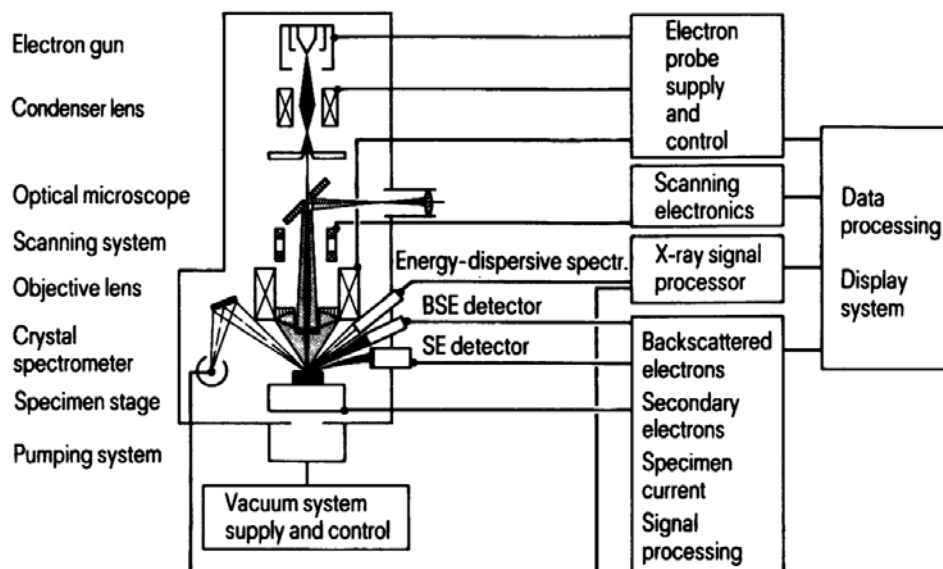


The specimen, the diffracting crystal and the detector stay on the **Rowland circle**. To scan the wavelength, this circle rotates around the specimen to satisfying the Bragg law

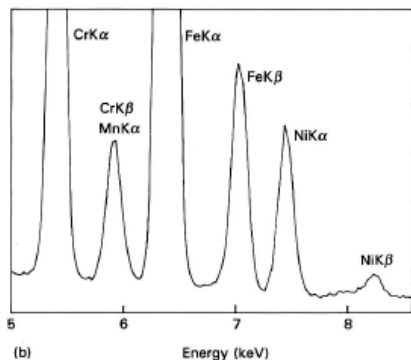
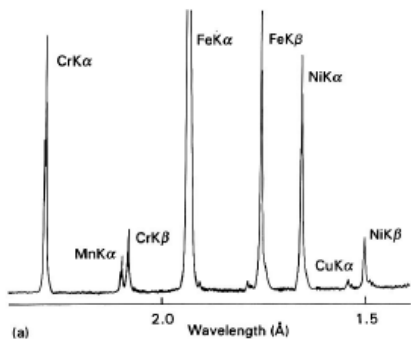


Johansson focusing spectrometer:
 The diffracting crystal is bent with a curvature radius double of that of the Rowland circle
 The crystal surface is cylindrically ground to the radius of the Rowland circle

Electron Microprobe EPMA (Electron Probe MicroAnalyser) with WDS



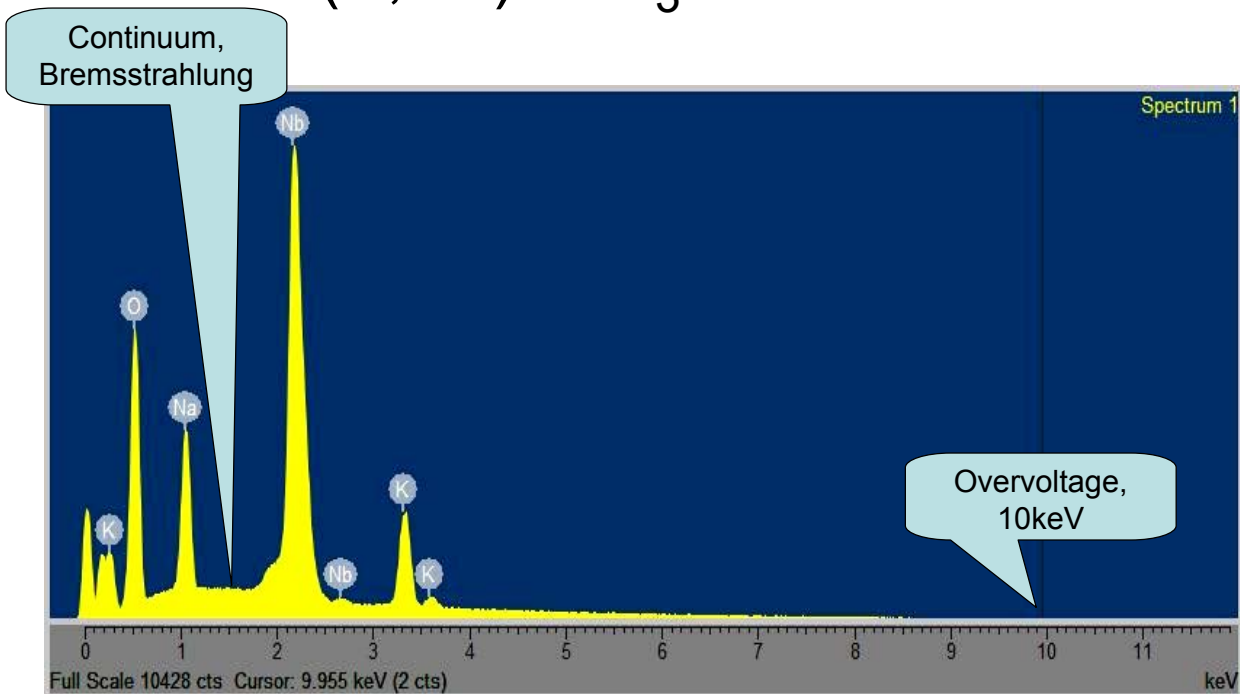
EDS <-> WDS



g. 6.7. Comparison of spectra from a steel (1.7 wt% Mn) at 20 kV recorded by (a) WDS and (b) EDS; note the Mn Kα peak is not resolved with EDS.

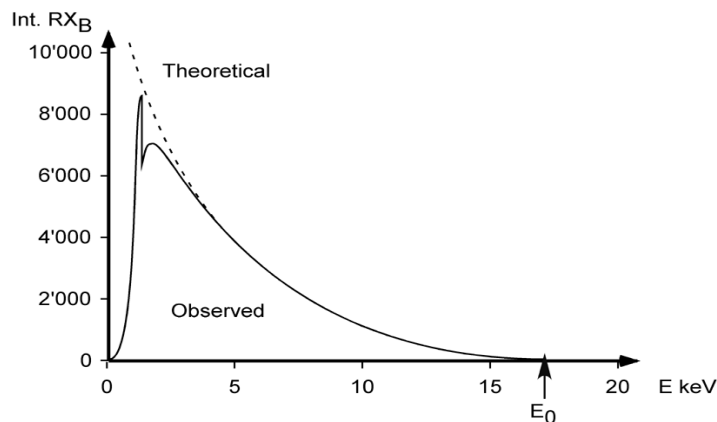
	EDS	WDS
<i>Energy Resolution</i>	80-180eV	~5eV
<i>Acquisition time</i>	1 min.	5-30min.
<i>Use</i>	Easy	difficult
<i>Standardless Analysis</i>	++	difficult
<i>Peak to background ratio</i>	100:1	1000:1

(K,Na)NbO₃



Bremstrahlung (background)

- when a charged particle (des-) accelerates or changes direction, it emits an electromagnetic wave.
 - This is widely used to produce synchrotron radiation
- On a bulk sample of atomic number Z:
 - $N(E)$ is the number of photons of energy E , E_0 the energy of the incident electron and K the Kramers constant

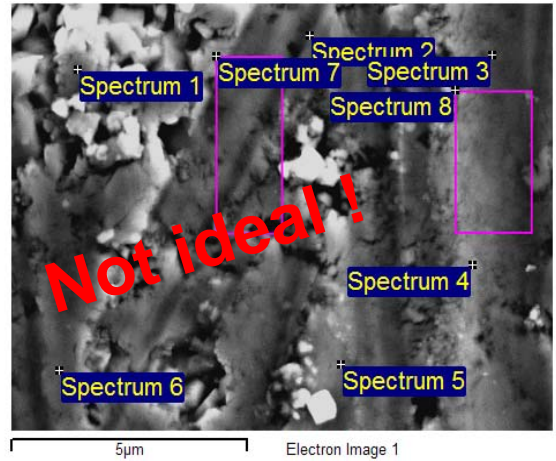
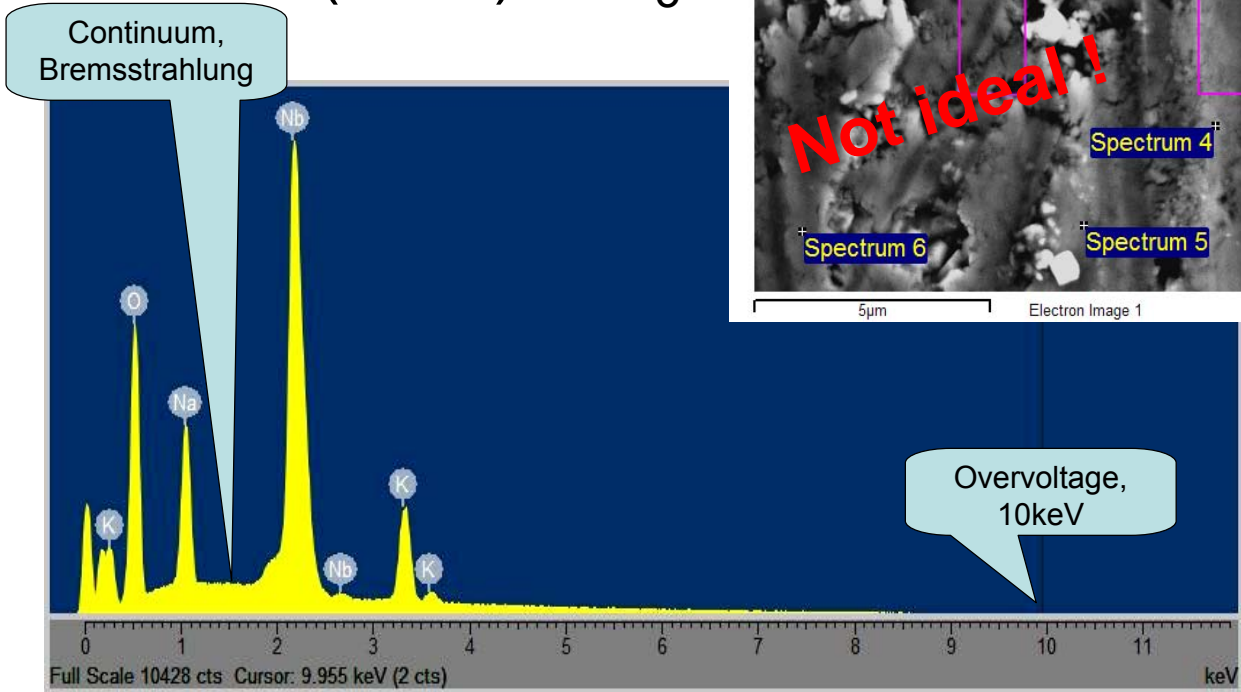


$$N(E) = \frac{KZ(E_0 - E)}{E}$$

EDS in SEM

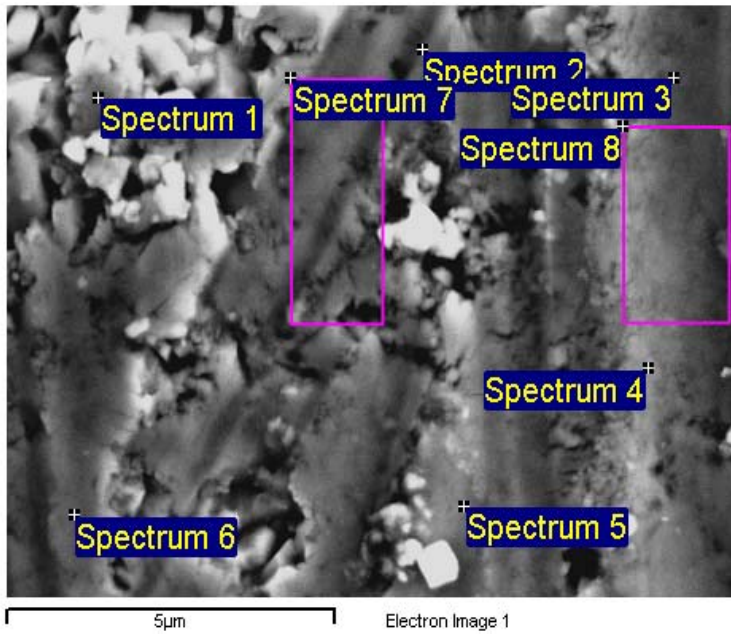
- Acquisition under best conditions
 - Flat surface without contamination (no Au coating, use C instead)
 - Sample must be homogenous at the place of analysis (interaction volume !!)
 - Horizontal orientation of the surface
 - High count rate (but dead time below 30%)
 - Overvoltage $U = E_0/E_c > 1.5-2$
- For acquisition times of 100sec. :
detection of ~0.5at% for almost all elements

(K,Na)NbO₃



Duane-Hunt limit

(K,Na)NbO₃



Spectrum	Na	K	Nb	O	Total
Spectrum 1	8.19	10.18	20.70	60.93	100.00
Spectrum 2	9.59	8.66	20.75	61.00	100.00
Spectrum 3	7.82	9.54	21.13	61.51	100.00
Spectrum 4	9.79	9.37	20.36	60.48	100.00
Spectrum 5	8.86	9.35	20.77	61.02	100.00
Spectrum 6	9.46	9.07	20.63	60.84	100.00
Spectrum 7	8.89	10.25	20.37	60.49	100.00
Spectrum 8	8.60	9.40	20.86	61.14	100.00
Max.	9.79	10.25	21.13	61.51	
Min.	7.82	8.66	20.36	60.48	

c) Quantification



- First approach:
compare X-ray intensity with a standard (sample with known concentration, same beam current of the electron beam)

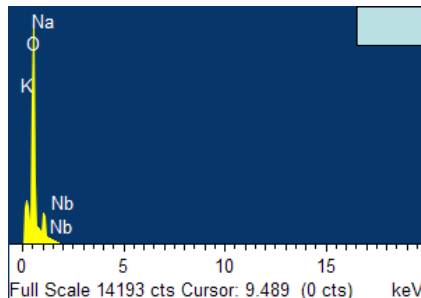
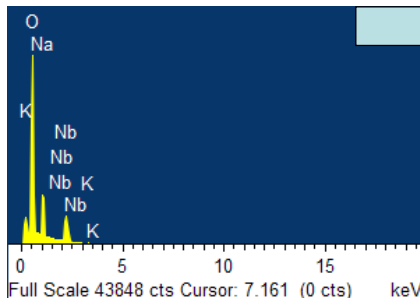
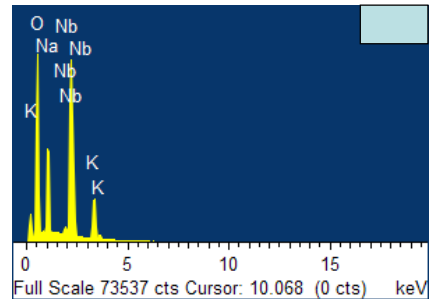
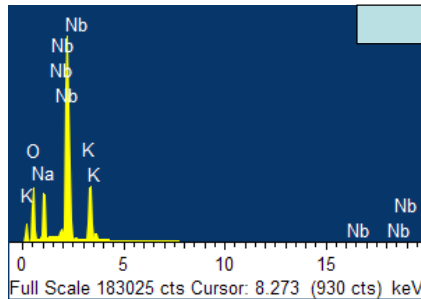
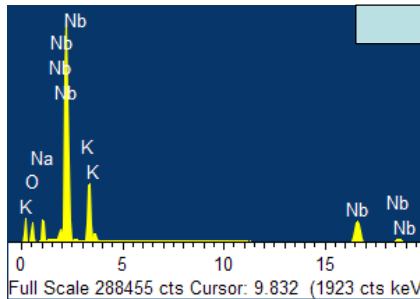
c_i : wt concentration of element i

I_i : X-ray intensity of char. Line

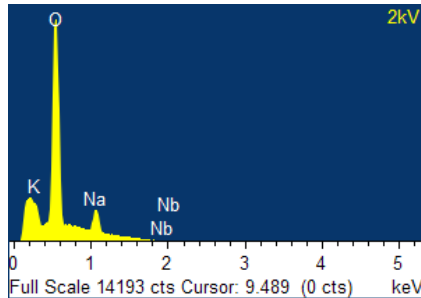
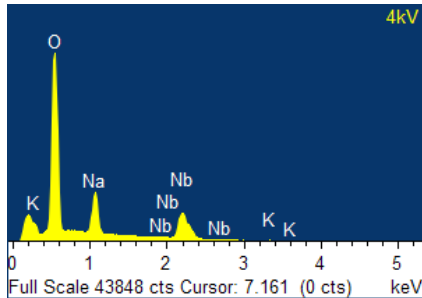
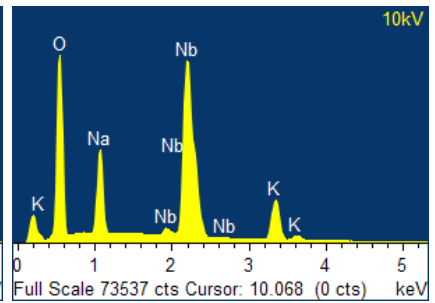
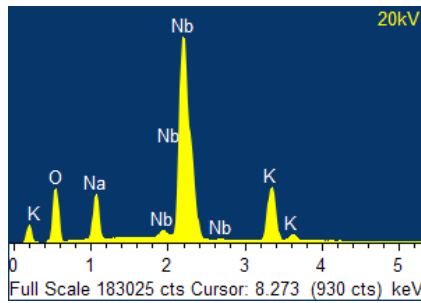
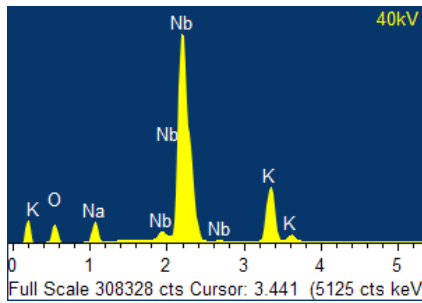
k_i : concentration ratio

$$\frac{c_i}{c_i^{std}} = \frac{I_i}{I_i^{std}} = k_i$$

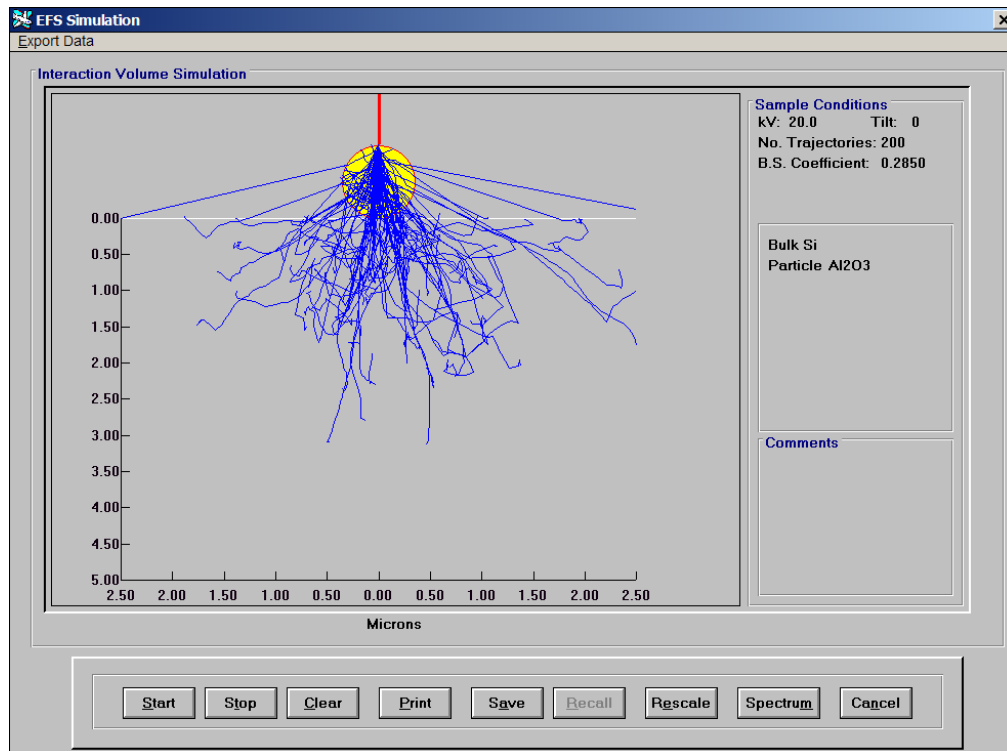
Intensity ~ Concentration...?



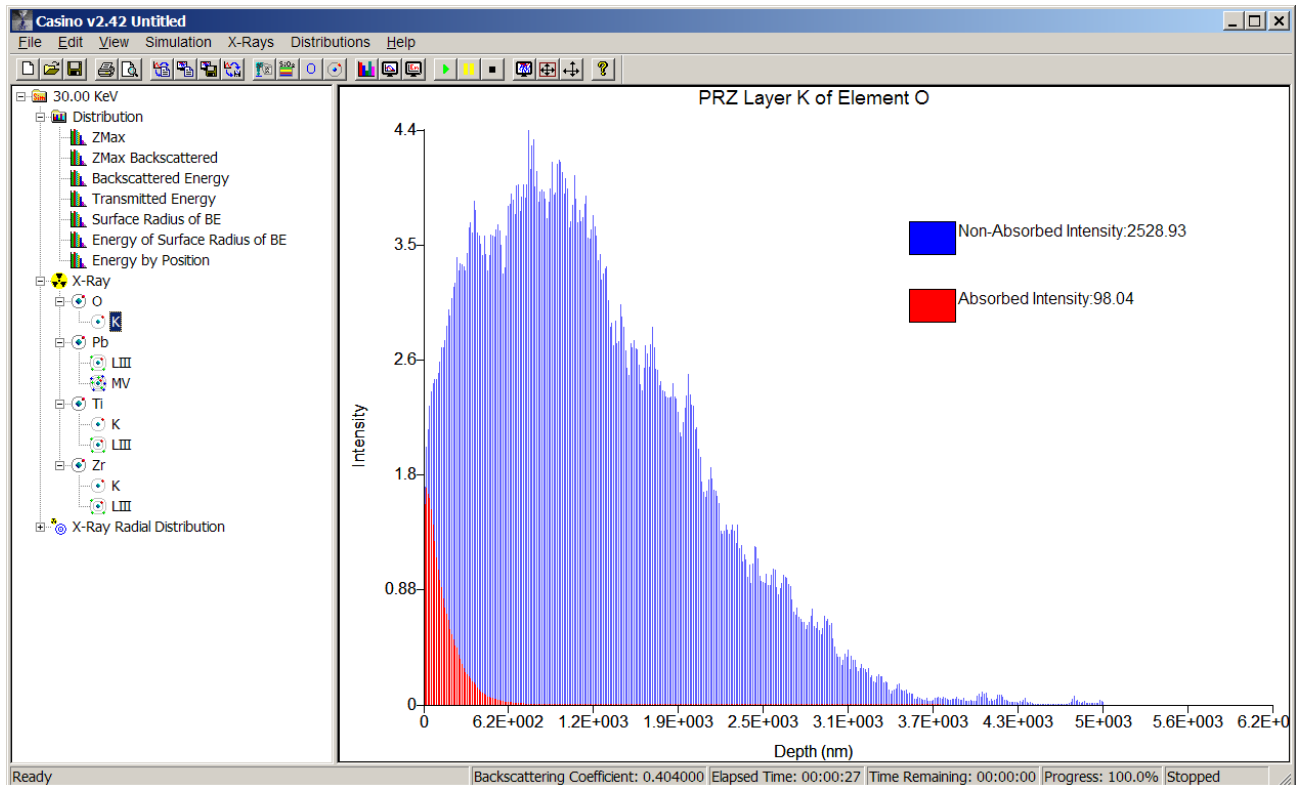
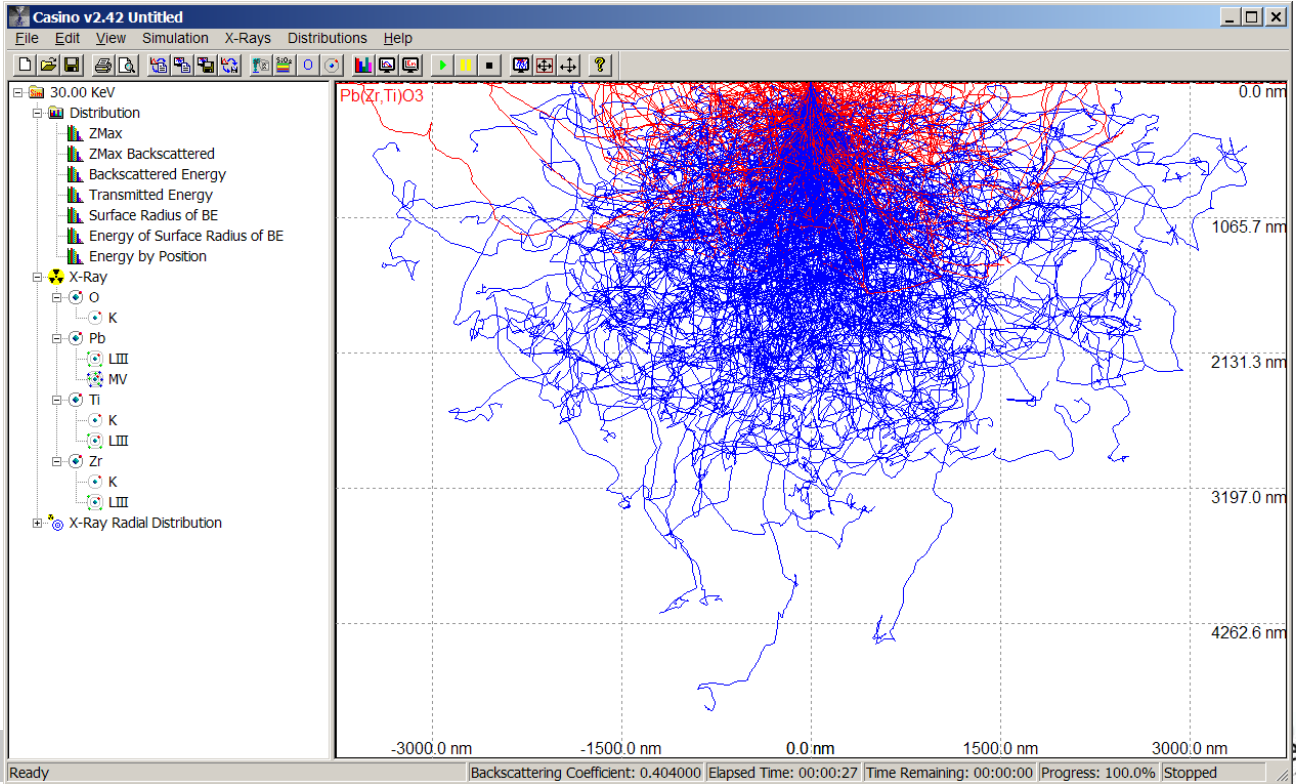
How many different samples...?



Electron Flight Simulator



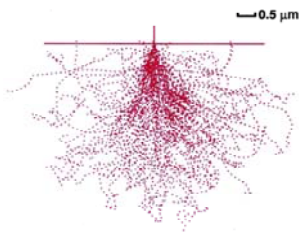
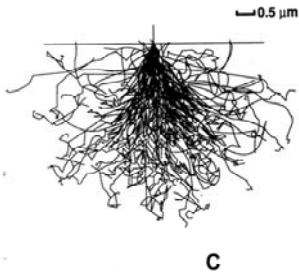
Casino



Quantification

When the going gets tough.....

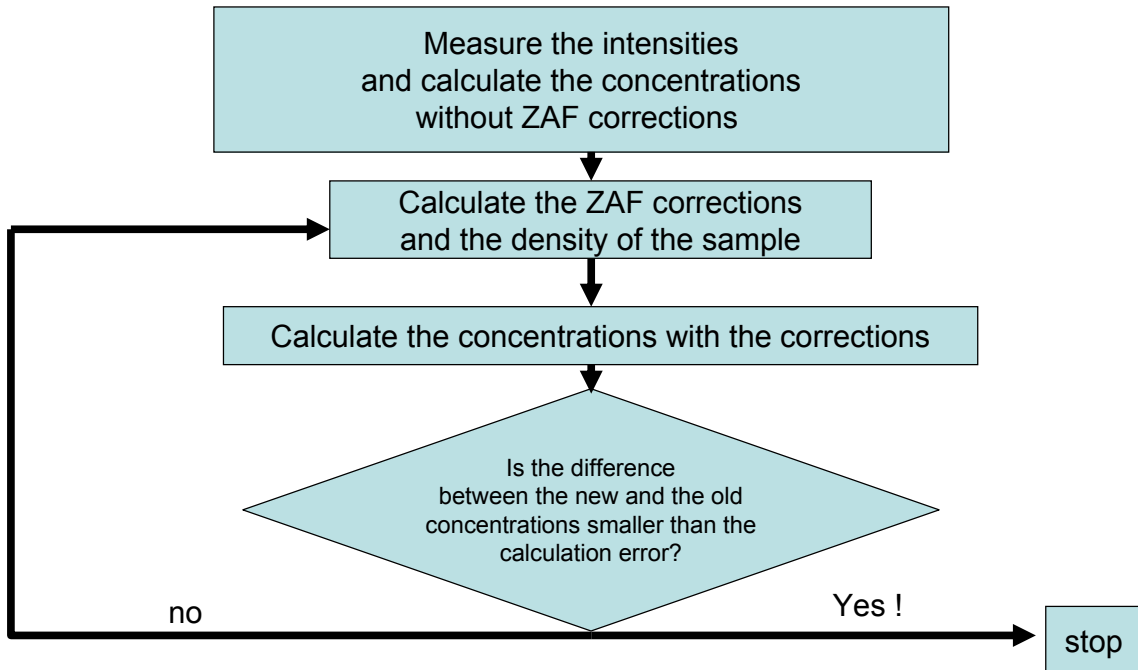
Correction matrix



$$[Z \times A \times F] \frac{C_i}{C_i^{std}} = \frac{I_i}{I_i^{std}} = k_i$$

- "Z" describe how the electron beam penetrates in the sample (Z-dependant and density dependant) and loose energy
- "A" takes in account the **absorption** of the X-rays photons along the path to sample surface
- "F" adds some photons when (secondary) **fluorescence** occurs

Flow chart of quantification



Correction methods:

- ZAF (purely theoretical)
 - PROZA Phi-Rho-Z
 - PaP (Pouchou and Pichoir)
 - XPP (extended Puchou/Pichoir)
-
- with standards (same HT, current, detector settings)
 - Standardless: theoretical calculation of I_{std}
 - Standardless optimized: « hidden » standards, user defined peak profiles

Quantitative EDX in SEM

- Acquisition under best conditions
 - Flat surface without contamination, horizontal orientation of the surface (no Au coating, use C instead)
 - Sample must be homogenous at the place of analysis (interaction volume !!)
 - High count rate (but dead time below 30%)
 - Overvoltage $U=E_0/E_c > 1.5-2$
- For acquisition times of 100sec. :
detection of ~0.5at% possible for almost all elements
- **Standardless acquisition** possible with high accuracy (intensities of references under the given conditions can be calculated for a great range of elements), test with samples of known composition, light elements (like O) are critical...
- Spatial resolution depends strongly on HT and the density of the sample

Demo NSS/INCA

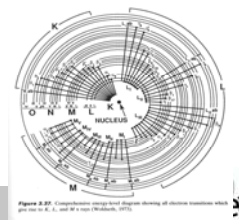
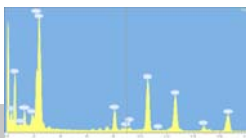
- Peak finding, synthetic spectrum
- Spectrum imaging (extraction of elements)

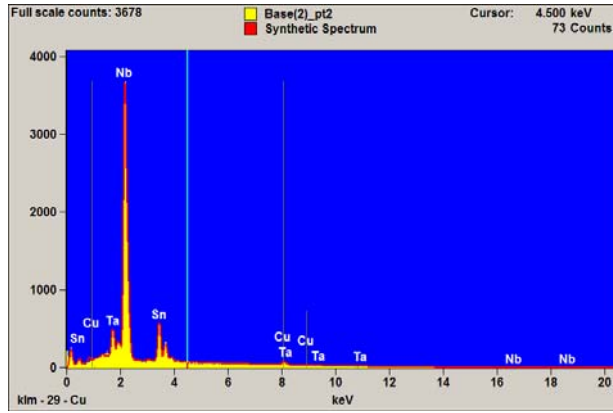


Modern EDX systems:

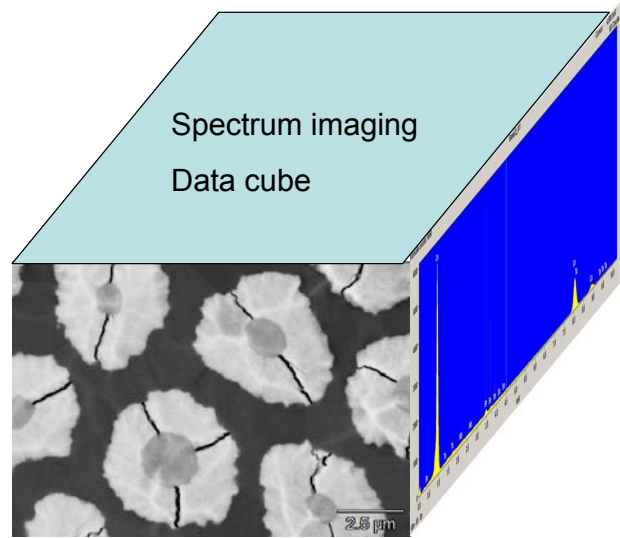


- **User friendly** interfaces
- New and more **powerful electronics** (stability of calibrations, higher count rate)
- **Drift compensation** for long acquisition times (element mapping on CM300 at high mag, "sitelock")
- **Synthesized spectra** (spectrum overlay) easier identification
- Advanced element mapping: **Spectral imaging** (data cube), selection of elements and regions post-acquisition
- Powerfull **reporting** and Export tools (Word, Powerpoint, html, tif etc.)

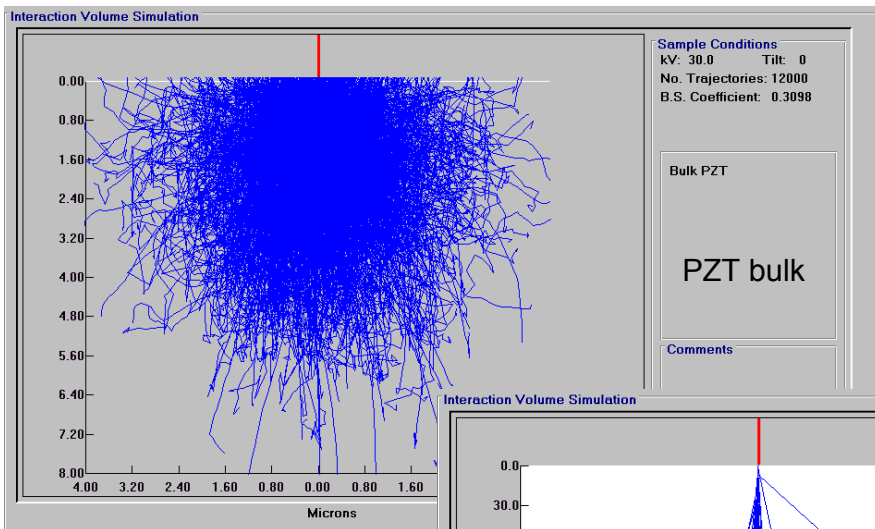
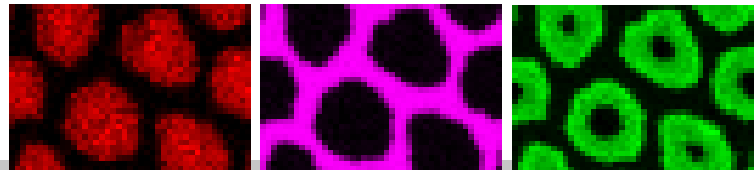




Synthesized spectrum

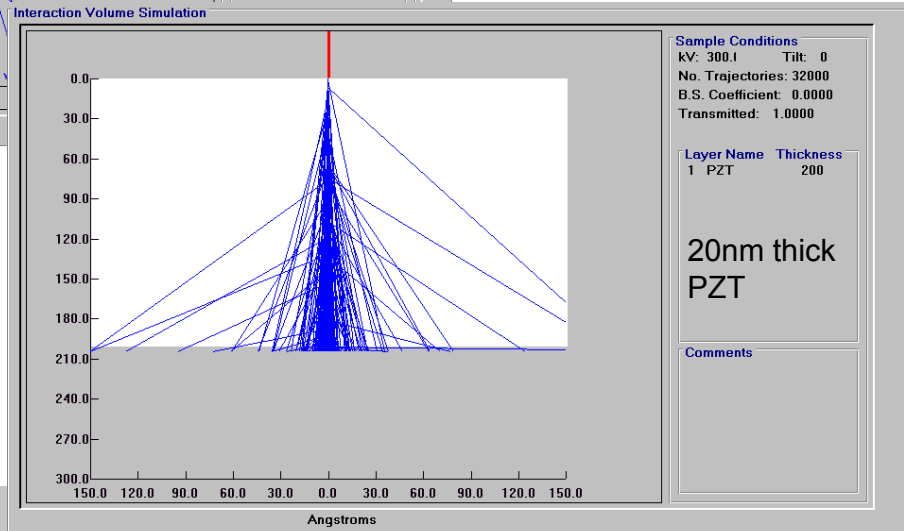


Extraction of element maps



High spatial resolution !

EDS in
TEM

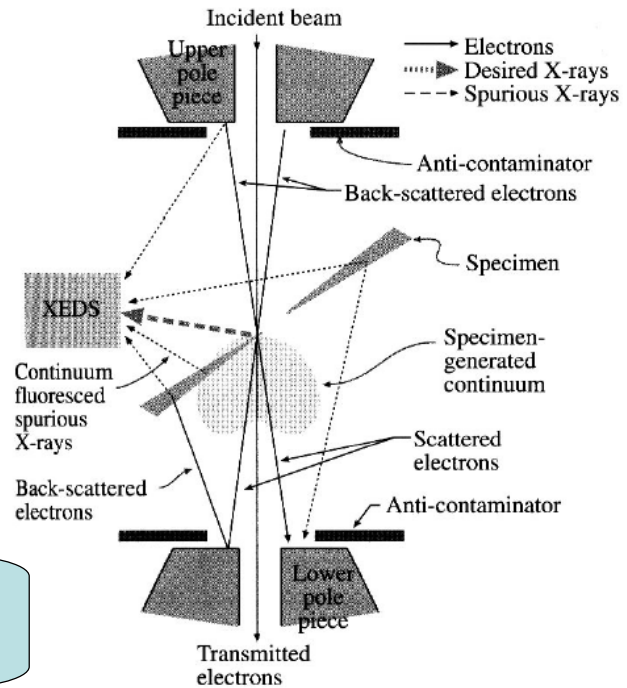


EDS in TEM

- Thin samples -> correction factors weak (A and F can be neglected)
- Very weak beam broadening -> high spatial resolution ~ beam diameter (~nm)

High energy: artifacts !

If only there wasn't this specimen preparation.....

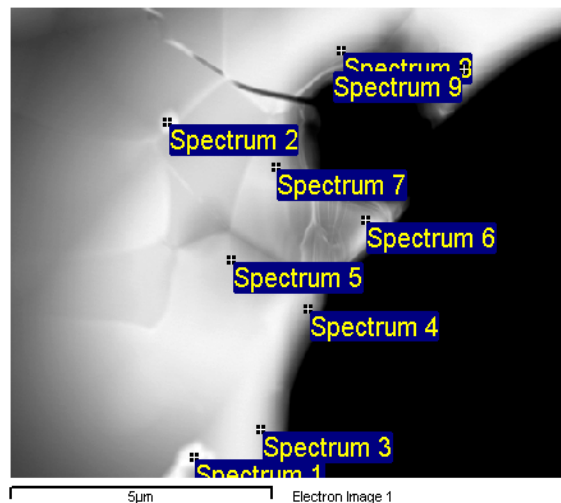


STEM point analysis PbMg_{1/3}Nb_{2/3}O₃ (bulk)

Processing option : Oxygen by stoichiometry (Normalised)

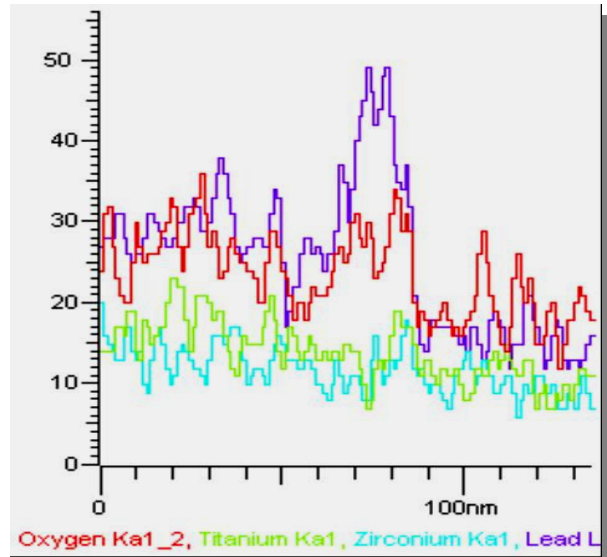
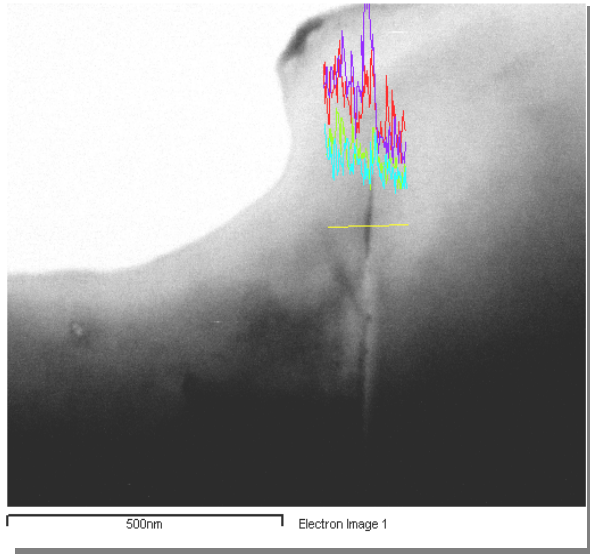
Spectrum	Mg	Si	Nb	Pb	O	Total
Spectrum 1	30.02	13.32			56.66	100.00
Spectrum 2	19.15	7.96	4.11	11.72	57.06	100.00
Spectrum 3	6.01		12.49	22.13	59.37	100.00
Spectrum 4	5.65		12.39	22.67	59.29	100.00
Spectrum 5	5.63		12.48	22.52	59.36	100.00
Spectrum 6	5.98		13.66	20.11	60.25	100.00
Spectrum 7	5.55		12.45	22.66	59.34	100.00
Spectrum 8	5.49		12.96	21.84	59.72	100.00
Spectrum 9	5.63		12.19	23.04	59.14	100.00
Max.	30.02	13.32	13.66	23.04	60.25	
Min.	5.49	7.96	4.11	11.72	56.66	

All results in Atomic Percent



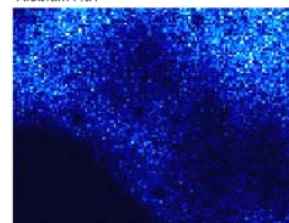
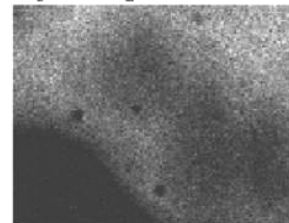
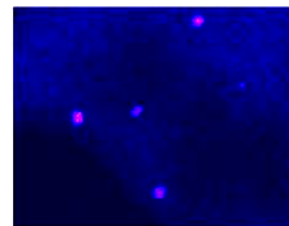
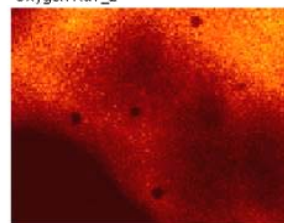
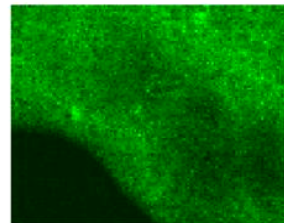
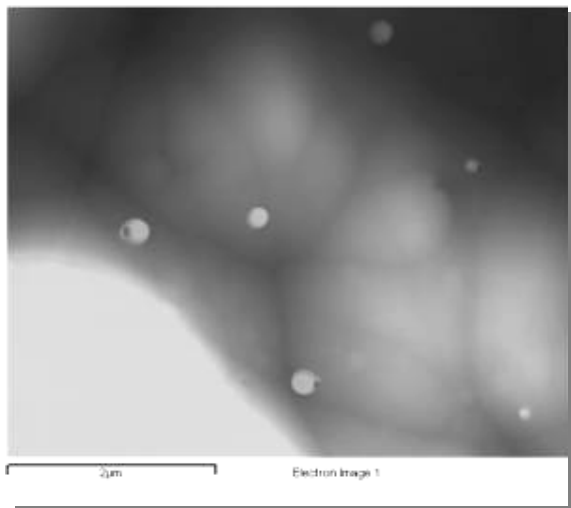
STEM linescan

Pb(Zr,Ti)O₃ (thick film), slight Pb excess



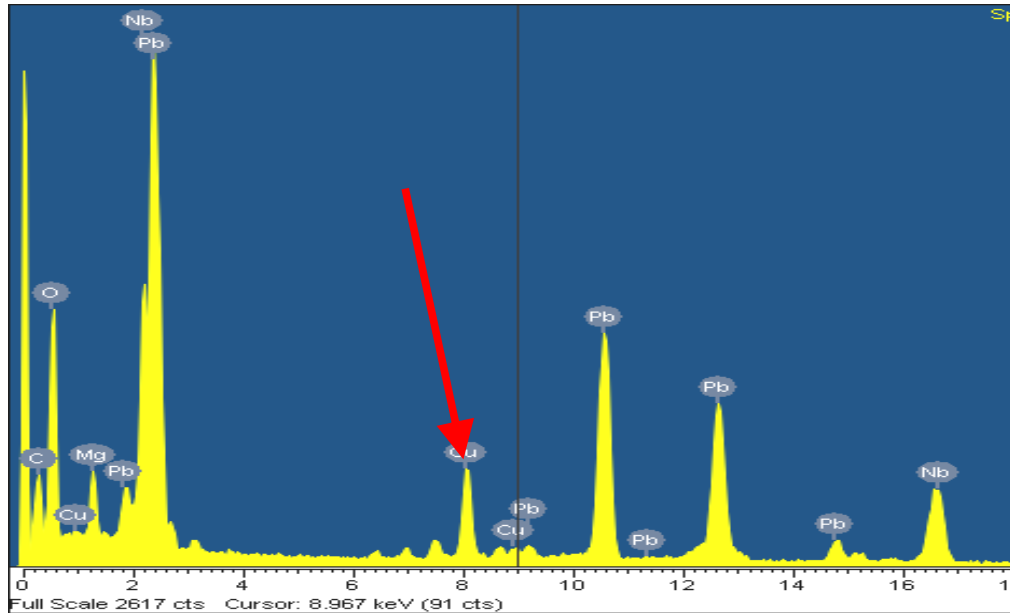
STEM Element Mapping

PMN/PT 90/10 (bulk)

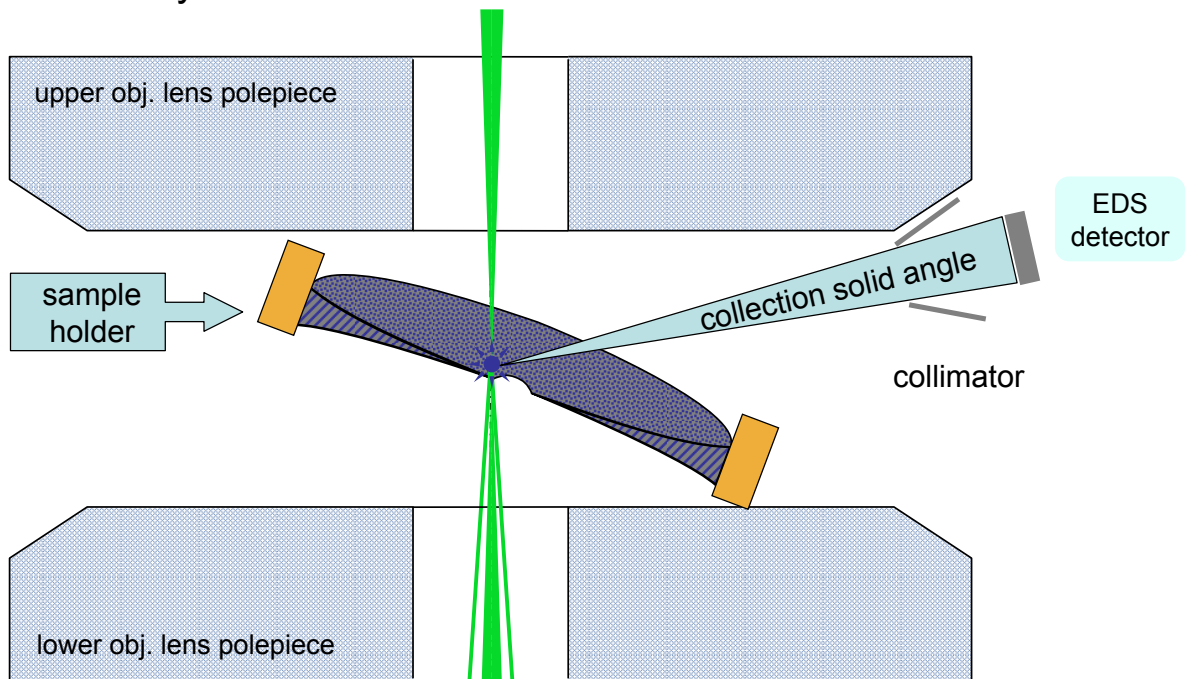


Artifacts

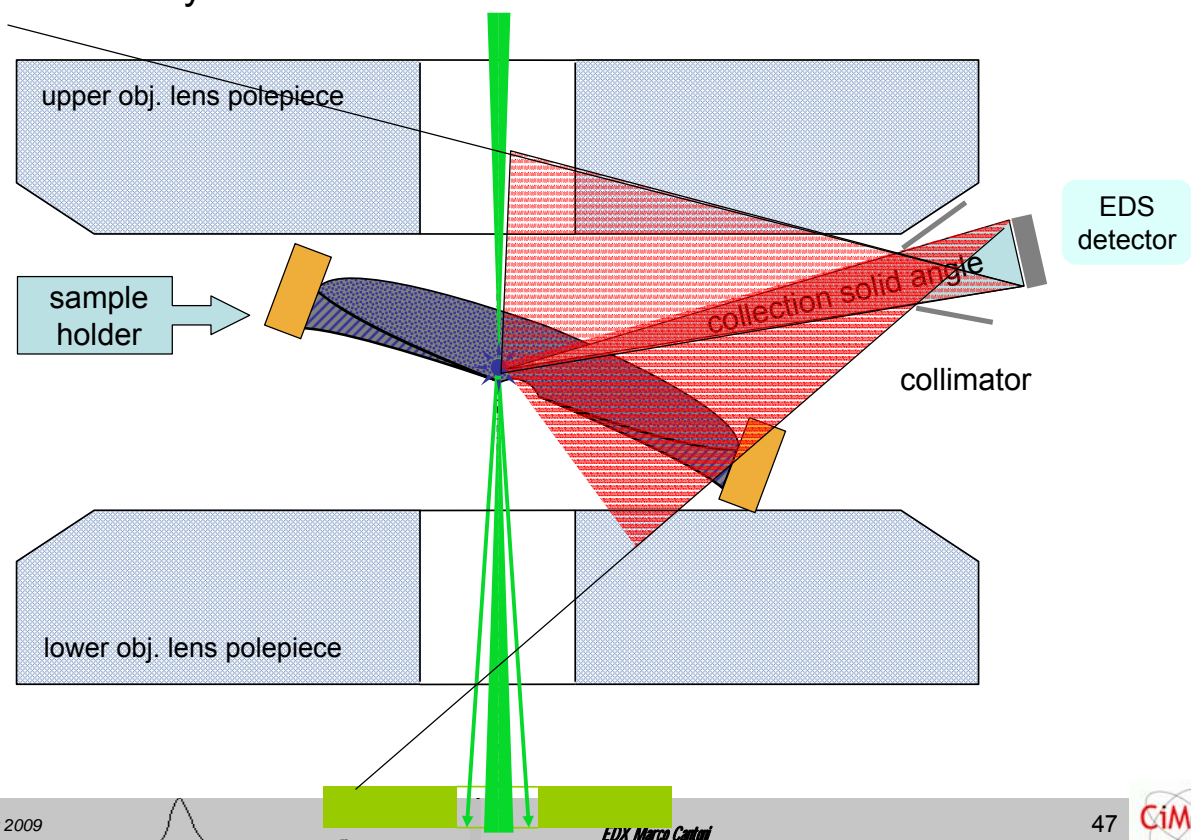
how to recognize/minimize them



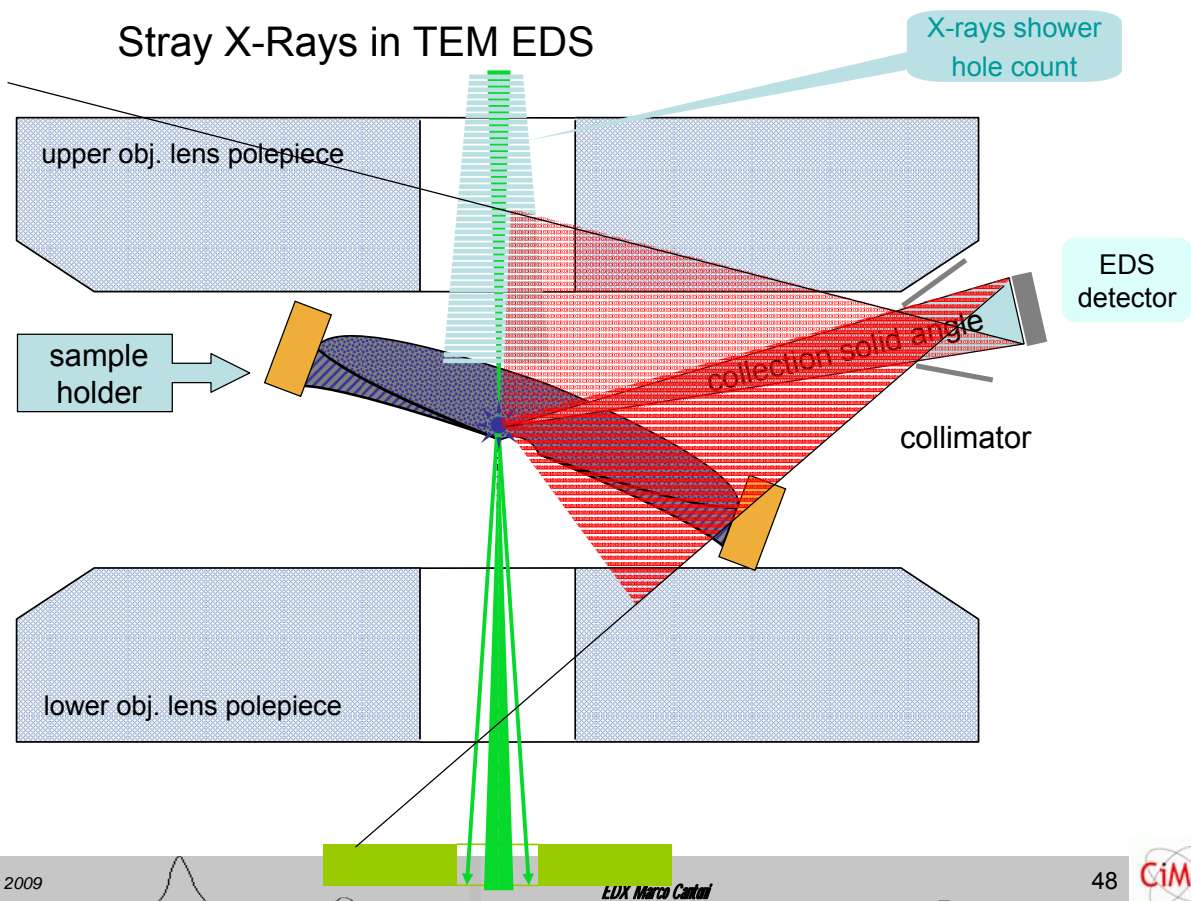
X-Ray collection in TEM EDS



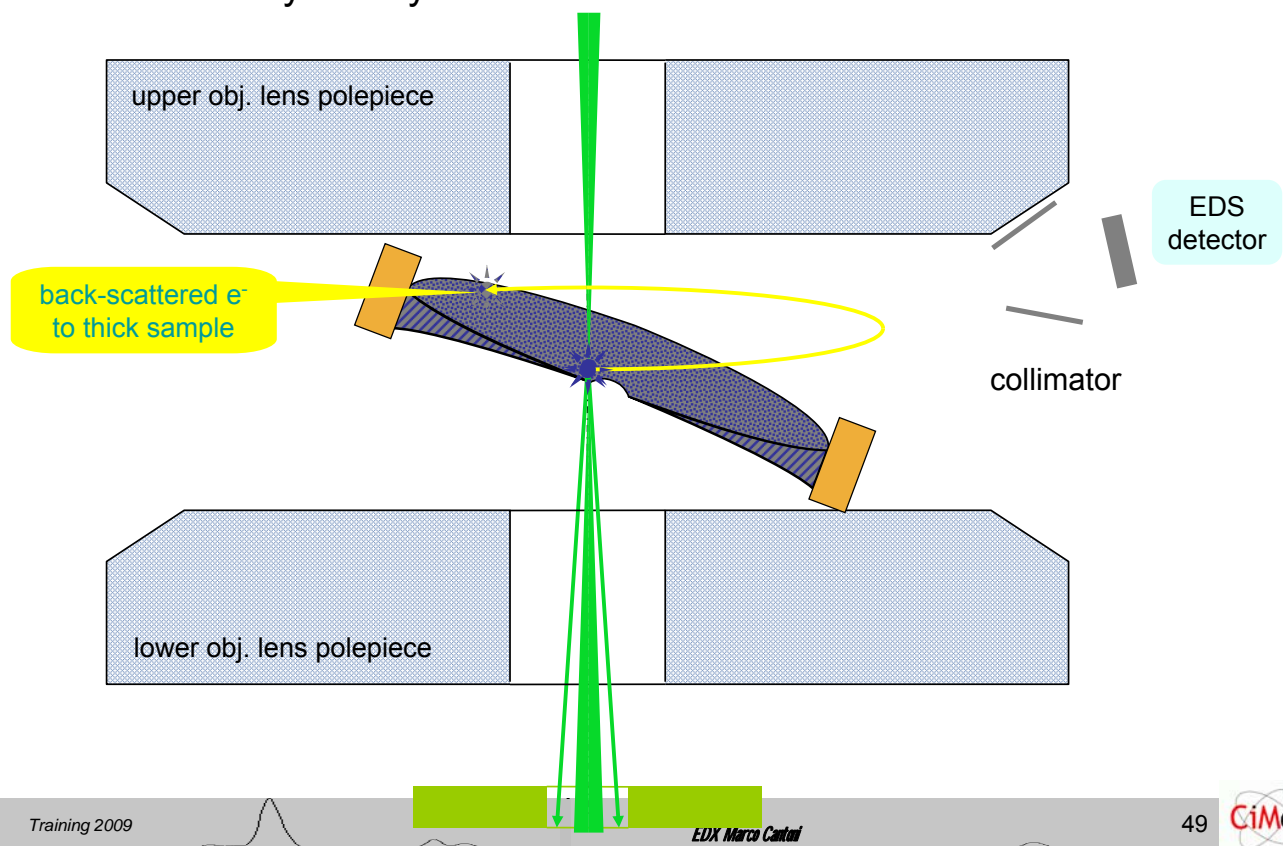
X-Ray collection in TEM EDS



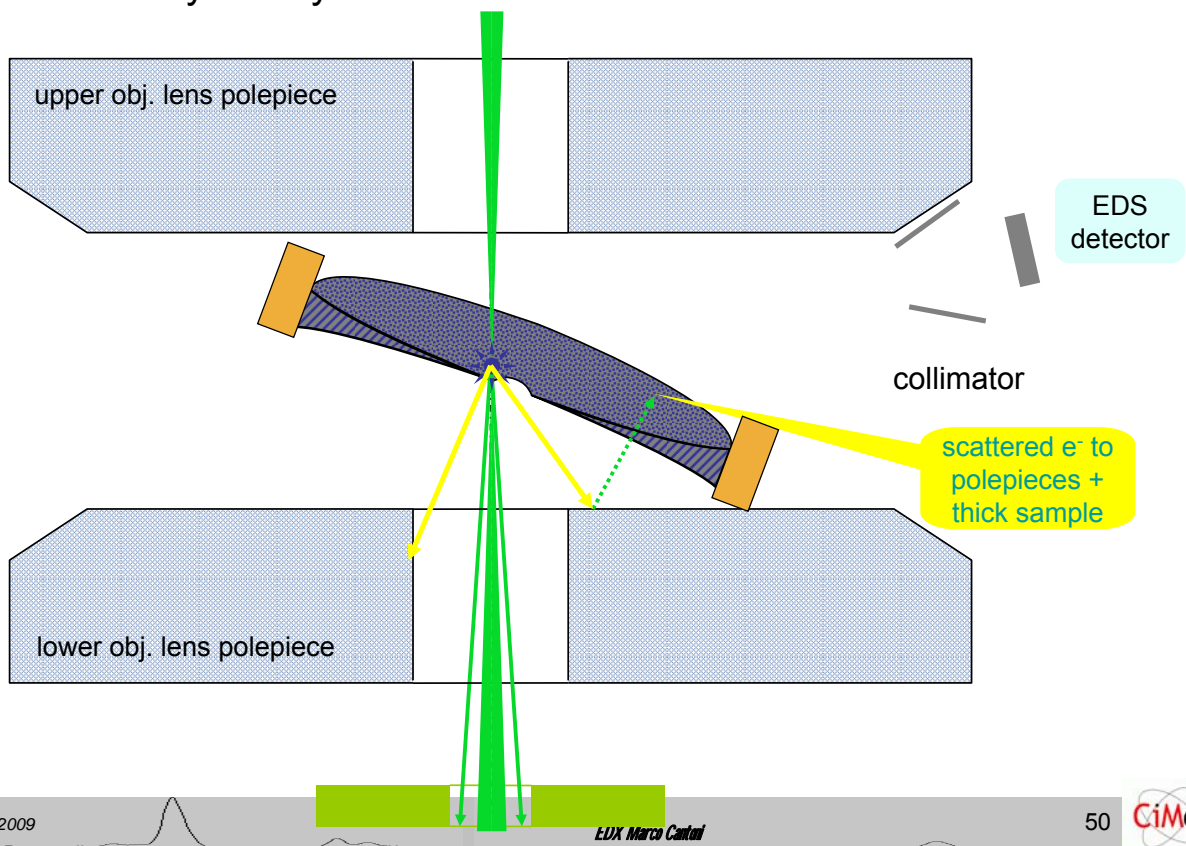
Stray X-Rays in TEM EDS



Stray X-Rays in TEM EDS

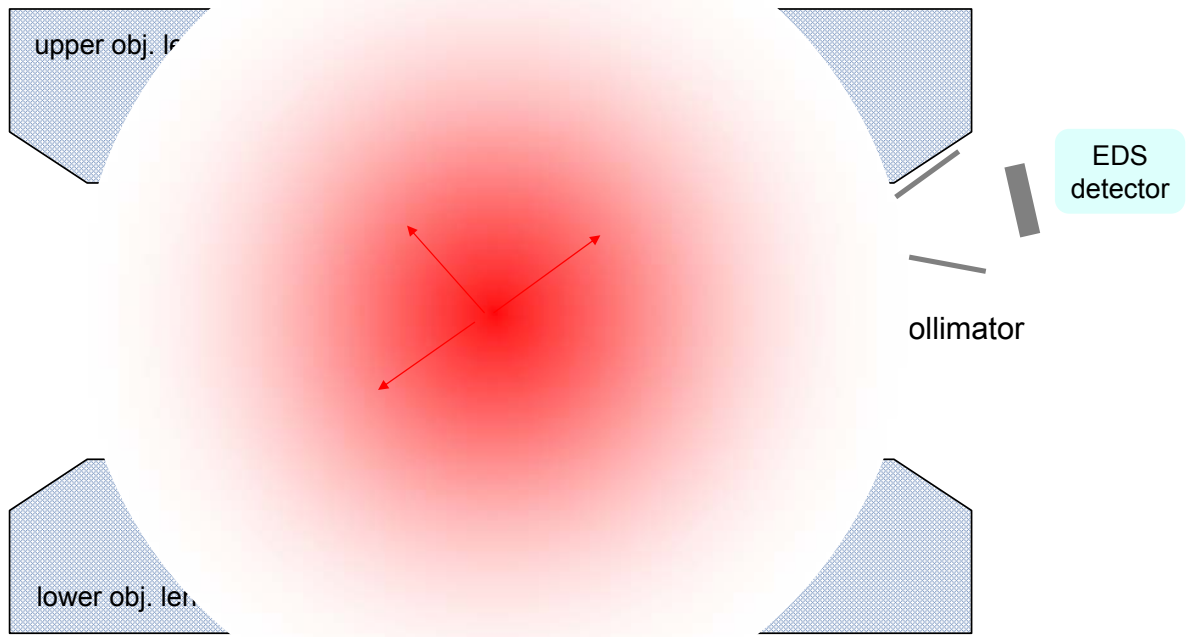


Stray X-Rays in TEM EDS

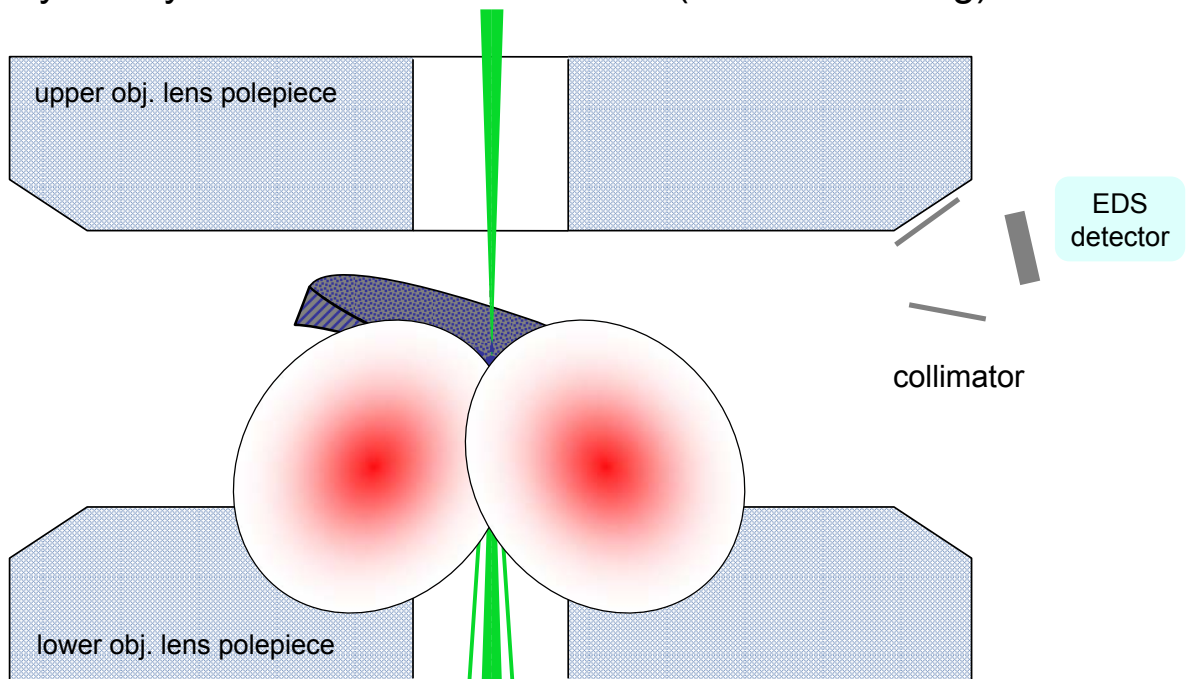


Stray X-Rays

Characteristic x-rays



Stray X-Rays in TEM EDS: continuum (Bremsstrahlung)



Ideal samples:

- FIB samples: almost uniform thickness, small sample size (less bulk material around)

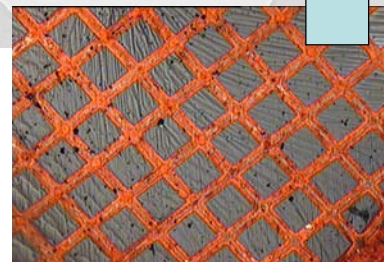
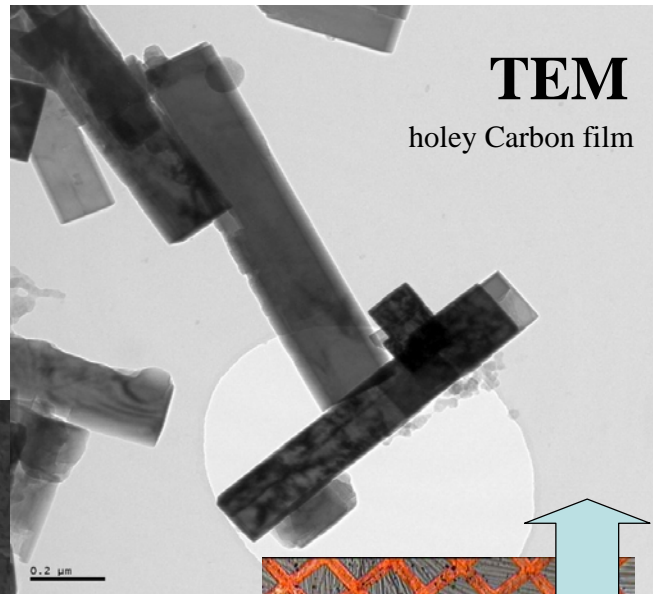
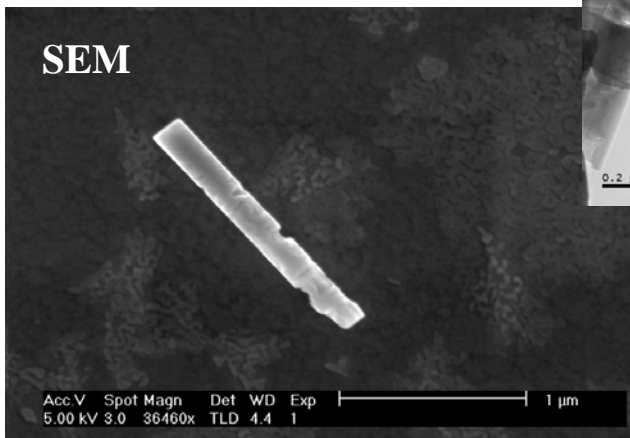
EDS in TEM

- Thin samples -> correction factors weak (A and F can be neglected), quantification “easy”
- Very weak beam broadening -> high spatial resolution ~ beam diameter (~nm)
- High energy -> artifacts
- Sample preparation, sample geometry.....

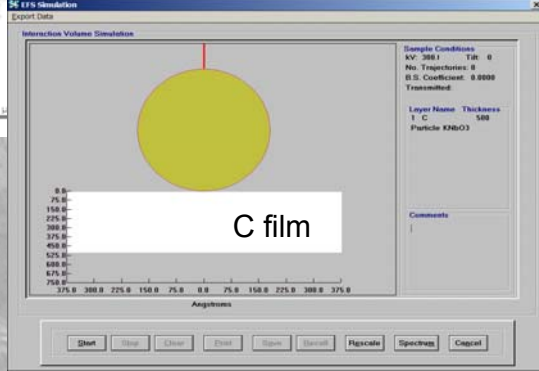
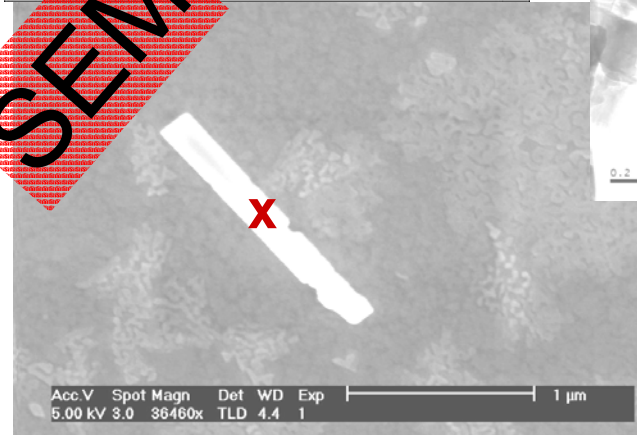
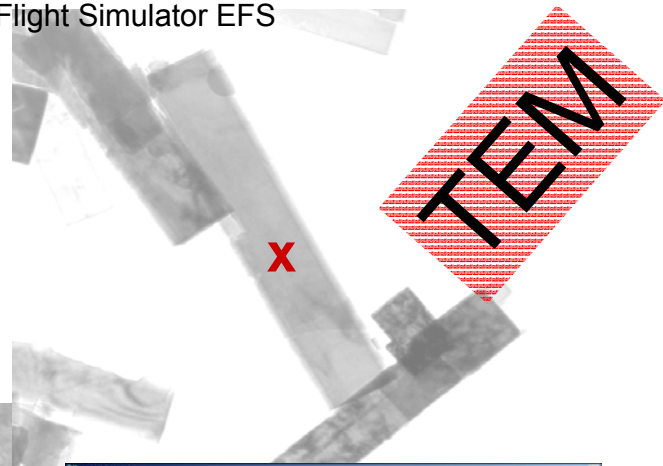
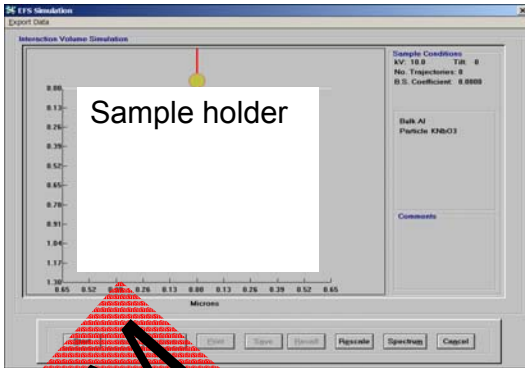
Bonus

**EDX of
Powders
nano particles
SEM or TEM ???**

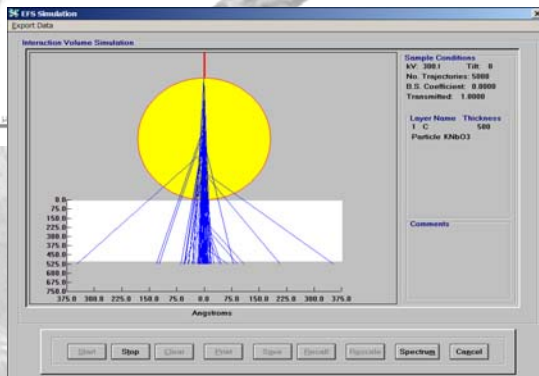
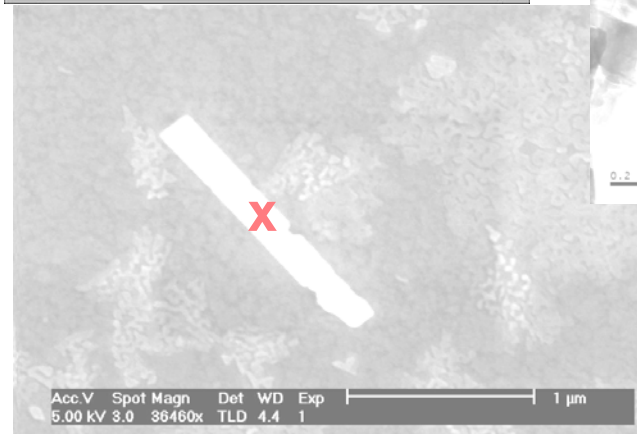
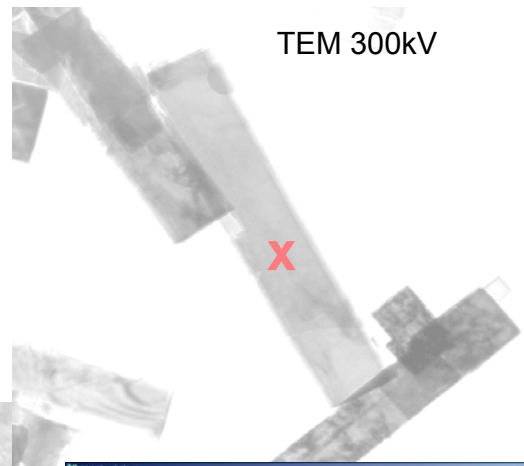
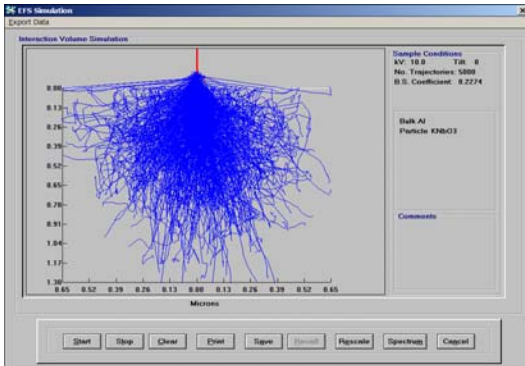
KNbO_3
Nano-rods



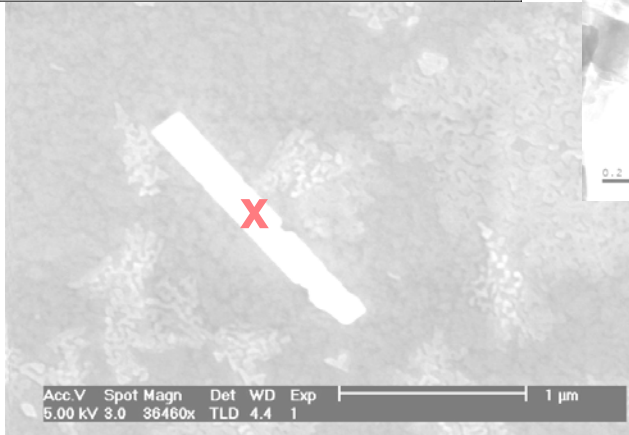
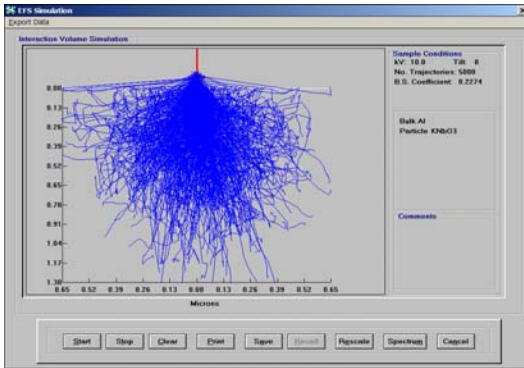
Simulation Electron Flight Simulator EFS



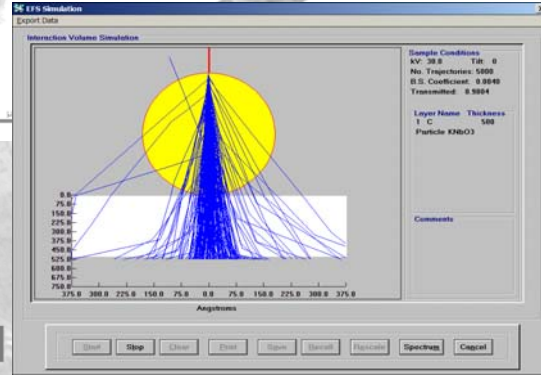
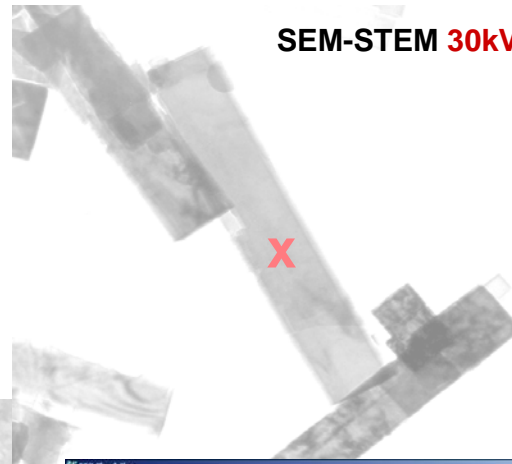
SEM 10kV



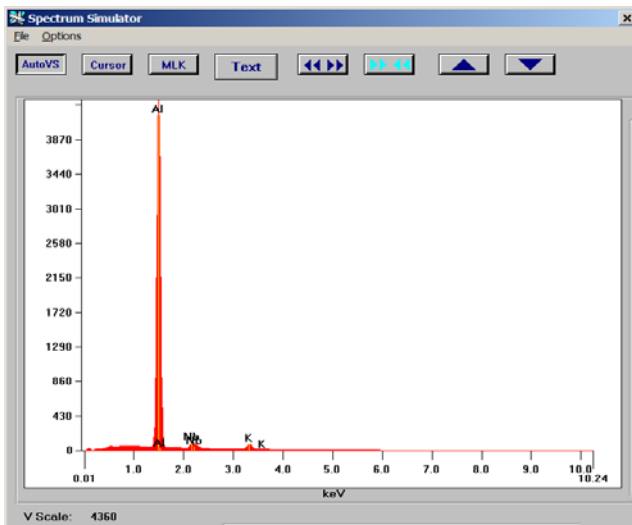
SEM 10kV



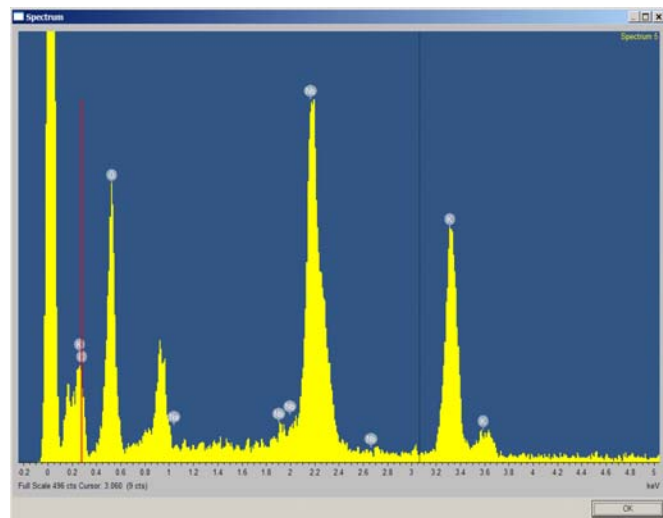
SEM-STEM 30kV



SEM 10kV



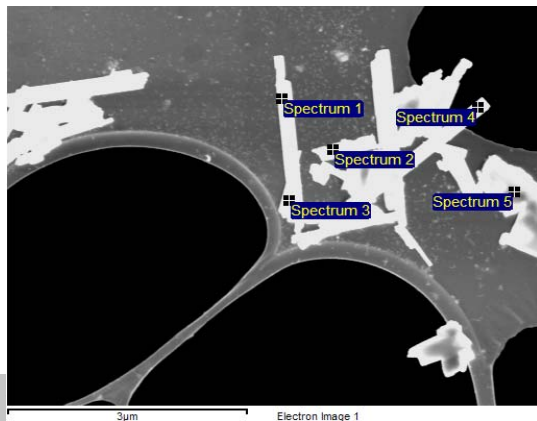
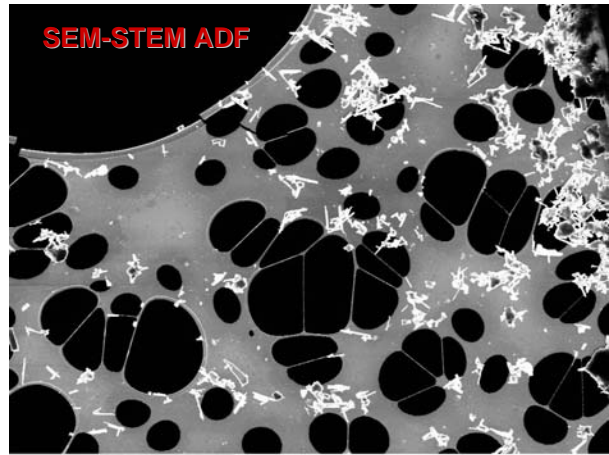
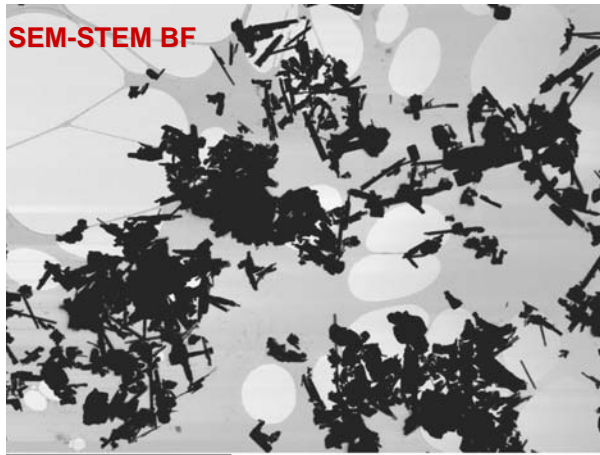
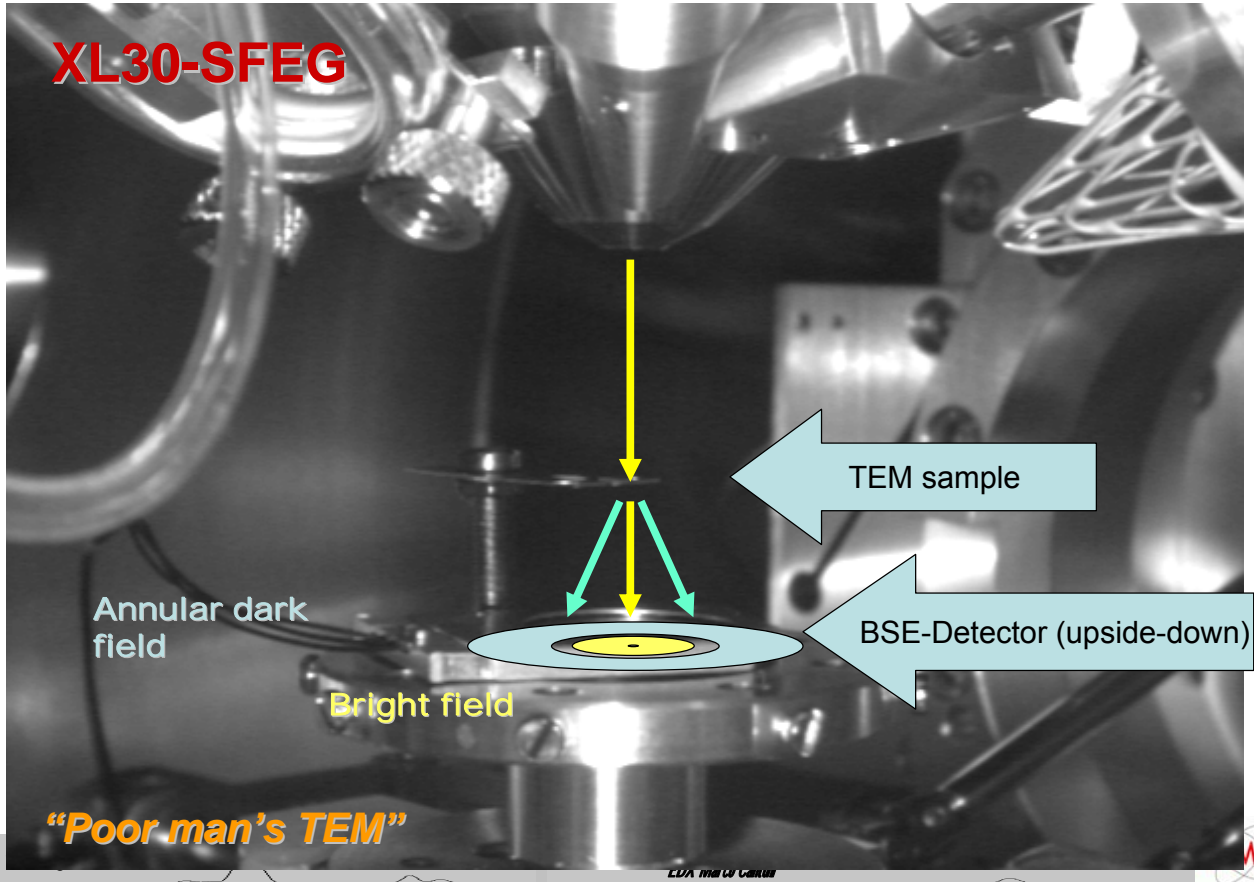
STEM 30kV



- Interaction volume \gg particle size
- Sample holder analyzed
- No deconvolution possible
- Low HT = limited energy range for ionization energies

- Easy interpretation
- “no” contribution from substrate (C)
- MBTF corrections for quantification

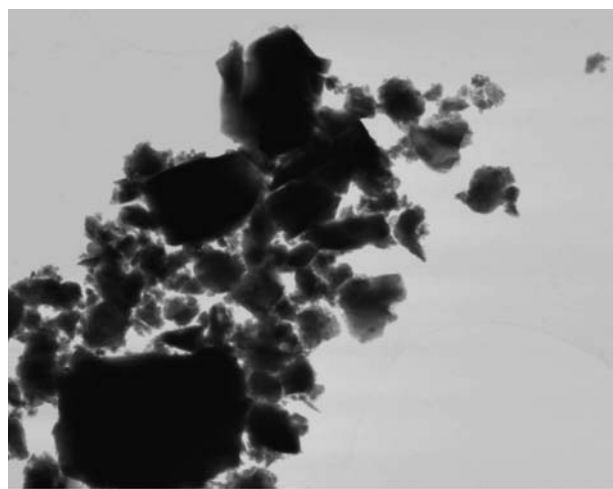
XL30-SFEG



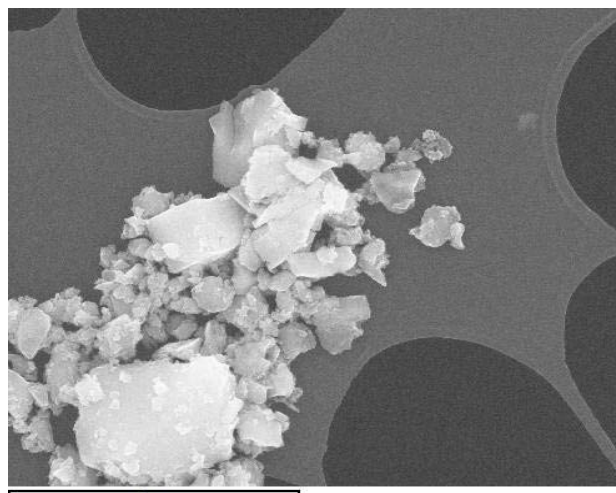
KNbO3

Processing option : All elements analysed (Normalised)

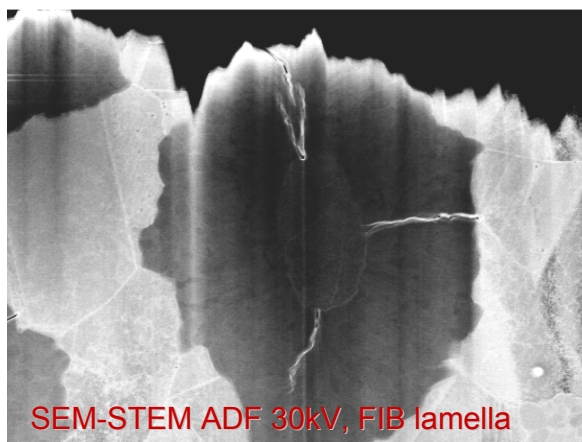
Spectrum	In stats.	O	K	Nb
Spectrum 1	Yes	56.00	21.88	22.12
Spectrum 2	Yes	58.89	19.01	22.10
Spectrum 3	Yes	43.22	27.24	29.54
Spectrum 4	Yes	67.41	16.23	16.36
Spectrum 5	Yes	56.23	20.82	22.95
Mean		56.35	21.04	22.61
Std. deviation		8.68	4.08	4.68
Max.		67.41	27.24	29.54
Min.		43.22	16.23	16.36



SEM-STEM BF

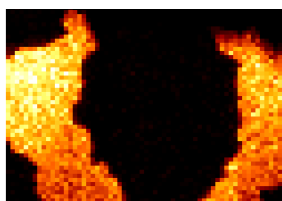


SE detector

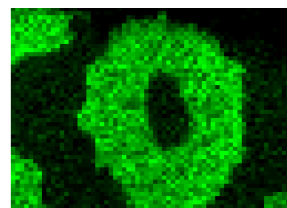


SEM-STEM ADF 30kV, FIB lamella

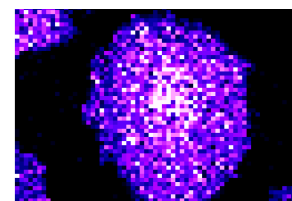
Electron Image 1



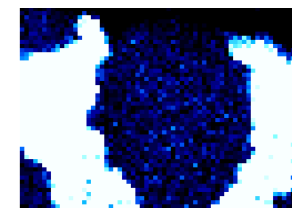
Cu Ka1



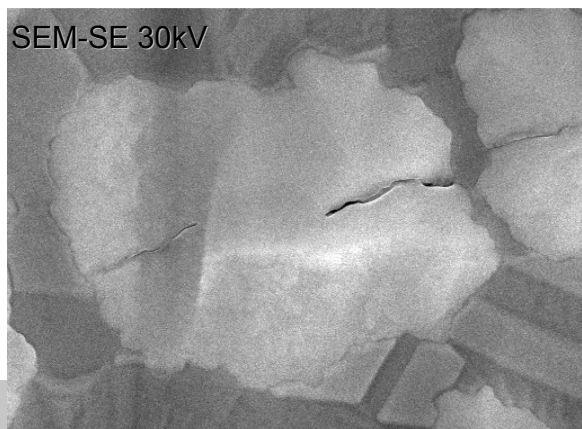
Sn La1



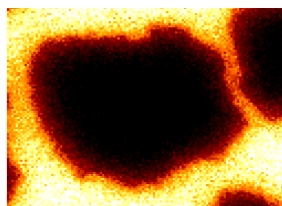
Nb Ka1



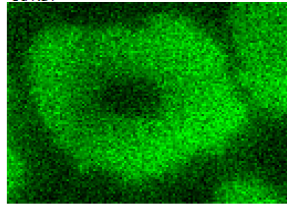
Ta La1



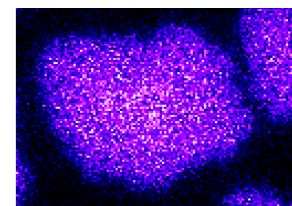
SEM-SE 30kV



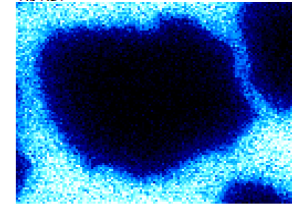
Cu Ka1



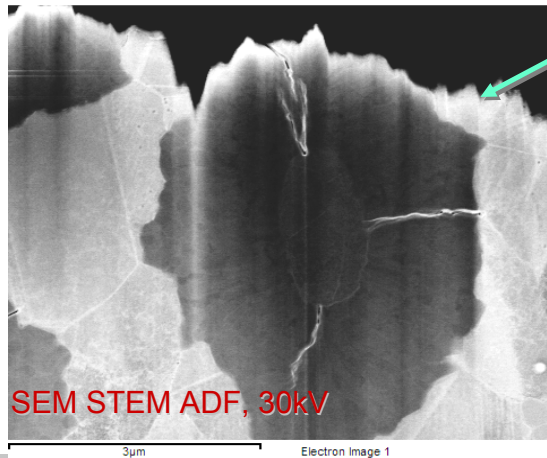
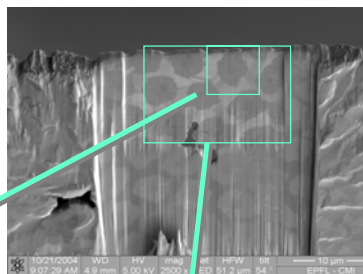
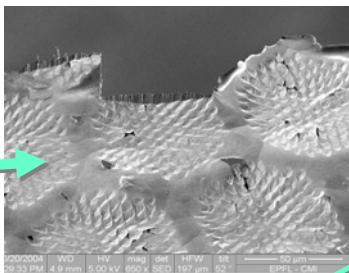
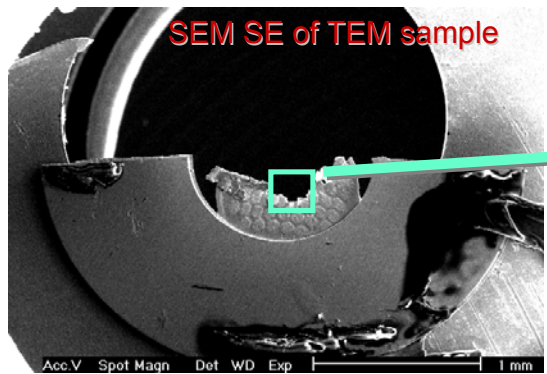
Sn La1



Nb Ka1



Ta La1



"true" STEM, CM300, 300kV, ADF

