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Microscale patterning of hydrogel stiffness through light-triggered uncaging of thiols†

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Mammalian cell behavior is strongly influenced by physical and chemical cues originating from the extracellular matrix (ECM). *In vivo*, ECM signals are displayed in a spatiotemporally complex fashion, often composed as gradients and in concentration profiles that change in time. Most *in vitro* models to study the role of ECM signals in regulating cell behavior are limited in capturing this microenvironmental complexity, as they are static and homogeneous. In order to achieve a dynamic control of the physical properties of a hydrogel network, we here designed a chemical scheme to control poly(ethylene glycol) (PEG) hydrogel stiffness in space, time and intensity. Specifically, we combined caging chemistry and Michael-type addition to enable the light-triggered local control of hydrogel crosslinking density. Thiol moieties of one of the reactive PEG macromers undergoing crosslinking were equipped with caging groups to prevent their susceptibility to the counter-reactive vinyl sulfone groups on the termini of the complementary PEG macromers. Thus, the crosslinking density of the hydrogel network could be tuned by uncaging with light which directly translated into differential patterns of hydrogel stiffness. Using this approach, user-defined stiffness patterns in a range of soft tissue microenvironments (*i.e.* between 3–8 kPa) were obtained and shown to influence the migratory behavior of primary human mesenchymal stem cells (hMSC). Stiffness gradients in the higher range (5.5–8 kPa) were able to elicit durotaxis towards the more densely crosslinked regions, whereas those in the lower range (3–5.5 kPa) showed no significant directional preference in hMSC migration. Our patterning tool should be useful for the manipulation of cell fate in various other contexts.

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Introduction

In the last two decades, synthetic hydrogels have emerged as versatile biomaterials for tissue engineering and cell biology.^{1–5} Their high water content and tunable physicochemical properties have allowed researchers to mimic the physical properties of specific ECMs of native tissues to study cell behaviour *in vitro*.^{6,7} However, most hydrogels are static and lack the possibility to recapitulate the dynamic properties of natural ECMs that are composed of multiple graded signalling cues and concentration profiles that are also changing in time. Indeed, this spatiotemporally tightly controlled display of cell-

instructive signals is, for example, crucial in directing cell behaviour during development.^{7–9}

In order to mimic dynamic ECM properties, new families of light-responsive synthetic hydrogels have been developed.^{10–12} A major focus has been on modulating by a light beam the stiffness of such polymer networks at the micrometer scale and at a desired time during an experiment.¹³ Indeed, just like in the case of biochemical signals, microscale heterogeneities in ECM stiffness have been shown to be involved in regulating cell fate during embryonic development, for example.^{14–16}

One widely used technique to generate such gels is the localized UV-induced radical crosslinking of methacrylate-, acrylate- or norbornene functionalized polymers.^{17–20} The application of this scheme has for example been used to locally alter crosslinking densities of hydrogels such as to generate stiffness gradients and manipulate cell behaviour in 2D and 3D.^{21–24} However, a drawback of radical reactions is their relatively poorly controllable crosslinking kinetics, which can lead to sharp rather than smooth gradients. Furthermore, the generated free radicals can compromise cell viability in 3D

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contexts. Although recent advances such as improved UV photoinitiators (*e.g.* Irgacure 2959, I2959, and lithium phenyl-2,4,6-trimethylbenzoylphosphine, LAP) and the new rapid step-growth mechanism of thiol-norbornene show low cytotoxicity and little effect on protein bioactivity, the need for radical-free photoinduction still persists, especially for radical-sensitive cell types.^{25–27} To overcome these limitations, researchers have begun to employ photo-releasing or ‘caging’ chemistries in order to spatiotemporally modify hydrogel properties.^{28,29} By light-induced cleavage of a bulky protecting group from a reactive site, caging technology allows to modulate biomolecular interactions and is now routinely used to investigate molecular processes in biology such as protein trafficking, and has been employed to locally control biochemical properties of hydrogels.^{29–31,32–34} Anseth *et al.* have for example adapted this concept for hydrogel engineering by

local light-mediated cleavage of a 2-nitrobenzyl linker in the PEG backbone to decrease hydrogel stiffness upon UV (365 nm), visible (405 nm) or two-photon (740 nm) light exposure.²⁸ This photo-degradation scheme was successfully applied to control the spreading of hMSCs within light-eroded channels. Although photo-degradation is an excellent approach for spatiotemporal hydrogel manipulation, it allows for altering gel features only in a subtractive rather than in an additive manner. Thus matrix stiffening is not possible using this approach. Considering this, we here introduce a novel scheme to control the intensity and location of stiffness in synthetic hydrogels. Our approach is based on photo-labile caging groups to temporarily mask one of the functional groups involved in hydrogel crosslinking (Fig. 1). Hence, local light exposure by uncaging causes reactivation of the functional group that can then undergo crosslinking, creating local

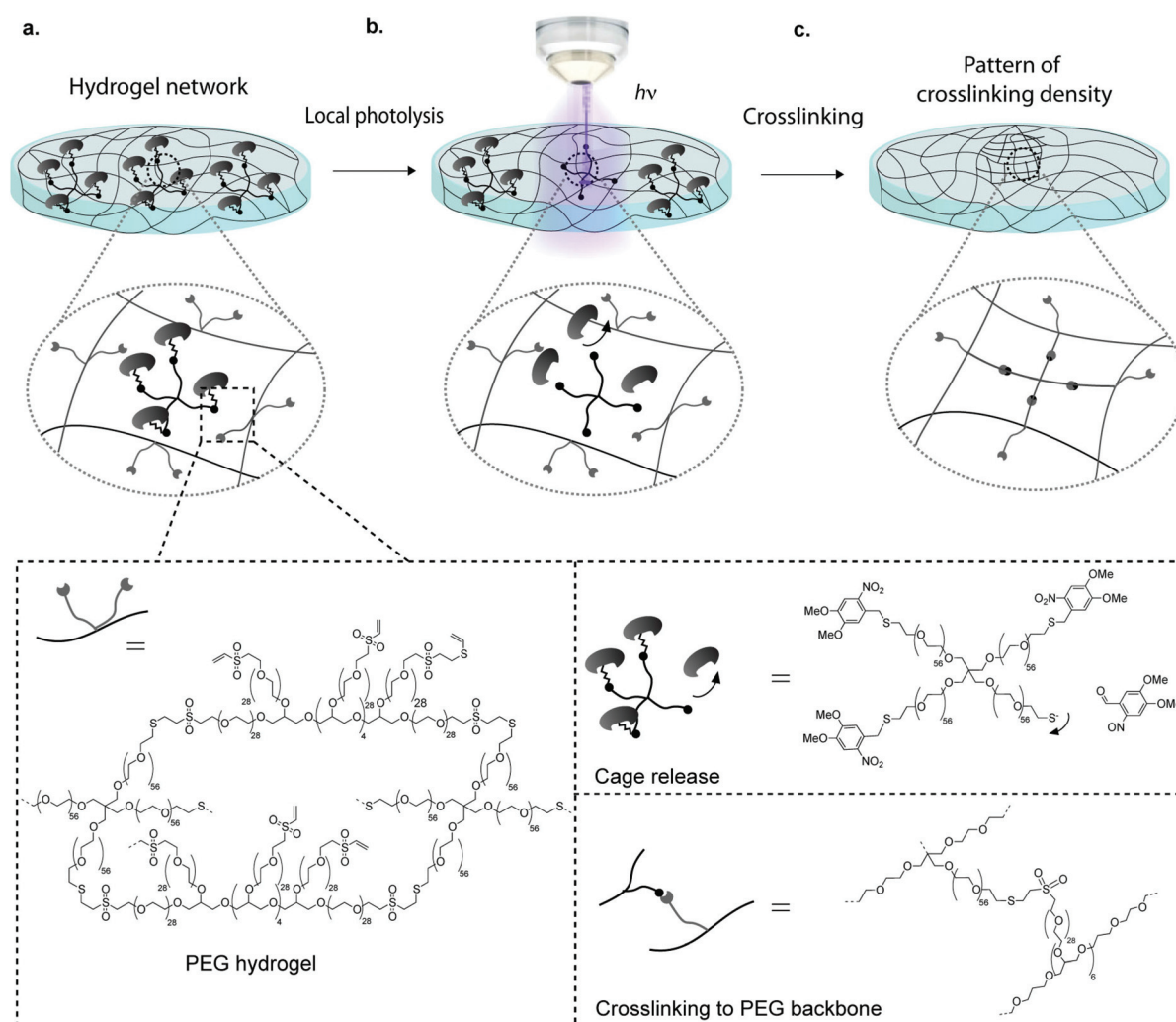


Fig. 1 Concept of light-mediated Michael-type addition for spatiotemporally controlled hydrogel crosslinking; a, A thiol-containing macromer is protected with photo-labile caging group and is dissolved in a hydrogel network comprising free vinyl sulfone groups; b, localized laser illumination causes photocleavage of the caging groups reactivating thiols; c, activated thiol-containing PEGs react to vinyl sulfone via Michael-type addition reaction creating localized stiffness patterns.

patterns of higher stiffness. We reasoned that an advantage of this approach would lie in its simplicity, specificity and the precision by which patterns of any desired shape and intensity variation could be generated.

To implement this concept, we chose a chemical hydrogel crosslinking scheme that is based on the well-established Michael-type addition of thiols onto unsaturated vinyl sulfone groups.^{35–37} This reaction is fast and reasonably specific, that is, at physiological pH competing reactions of vinyl sulfones with other nucleophiles such as amines are considerably slower.^{5,38} We reasoned that by temporally masking thiols with a photo-labile caging group, it should be possible to obtain light-dependent hydrogel crosslinking (Fig. 1). Here we show that patterning gels by photo-controlled Michael-type addition can be translated to highly localized stiffness patterns of broadly tunable shape and stiffness range. Using this approach, user-defined stiffness gradients were obtained and shown to influence the migratory behavior of hMSCs. It should be noted that during the preparation of this manuscript a somewhat similar approach of photo-controlled Michael-type addition to control hydrogel size and shape was published.³⁹ However, the authors used a different caging group as well as maleimide, a functional group that is prone to hydrolysis, in contrast to the much more stable vinyl sulfone group used here.

Experimental

Materials

Standard chemicals were purchased from Fluka AG or Sigma-Aldrich AG (Buchs, CH). Alexa Fluor 488 C5 maleimide was from Invitrogen. Hydroxyl-terminated 8arm-PEG ($M_w = 10$ kDa), thiol-functionalized 4arm-PEG-SH ($M_w = 20$ kDa) and linear methoxy-PEG-SH ($M_w = 5$ kDa) was purchased from NOF Corporation (Japan). PEG vinyl sulfone (PEG-VS) was produced and characterized as described elsewhere.⁴⁰ NMR spectra were recorded on a Bruker 400 MHz spectrometer. Reverse phase HPLC was performed on a Waters 2790 separation module with a binary gradient of water/TFA and acetonitrile coupled to UV detector (Photodiode Array Detector, Waters Corp. model 2996, Baden-Dättwil, Switzerland). Cell adhesion peptide: Ac-GRCGRDSPG-NH₂ was custom-made by GL Biochem (Shanghai).

The AFM to measure elastic moduli was Nanowizard II (JPK Instruments, Germany) in Force Mode, mounted on an epifluorescence microscope (Axiovert 100, Zeiss). We used a spherical silicon AFM cantilever (radius of curvature: 1 μm , nominal spring constants: 0.02 N m⁻¹, Microlevers, Nanoworld, Medium Sphere Type CONT). Elastic moduli were calculated using the JPK Data Processing software (JPK Instruments, Germany).

Synthesis of caged linear methoxy-PEG-SH model cross-linkers. Three different cages were used: 1-(4-hydroxyphenyl)-2-iodoethanone (HP), 1-(bromomethyl)-2-nitrobenzene (NB) and 1-(bromomethyl)-4,5-dimethoxy-2-nitrobenzene (DMNB).

HP. This compound was synthesized as reported.⁴¹ ¹H NMR in d₆-DMSO $\delta = 10.5$ (bs, 1H, OH), 7.89–7.87 (d, 2H, 2 \times CH_{aromat}), 6.86–6.84 (d, 2H, 2 \times CH_{aromat}), 4.47 (s, 2H, CH₂I).

mPEG5000-[S-HP]. 1 g (0.2 mmol) mPEG5000-SH was dissolved in a mixture of 40 mL DMF and 0.6 mL sodium methoxide (0.3 mmol) that had been degassed before with argon. Through the septum was added a solution of 393 mg (1.5 mmol, 5 eq.) HP in 5 mL (degassed) DMF. After stirring for 2 days at room temperature the solution was precipitated in cold Et₂O and filtered on paper filter. After solubilisation with CH₂Cl₂, activated charcoal was added to react for 1 hour followed by filtration through a filter cell. The solution was concentrated to about 15 mL and added to 500 mL cold Et₂O yielding yellowish powder. ¹H NMR in CDCl₃ $\delta = 8.43$ (bs, 1H, OH), 7.89–7.87 (d, 2H, 2 \times CH_{aromat}), 6.91–6.88 (d, 2H, 2 \times CH_{aromat}), 3.88 (s, 2H, CH₂S) 3.82–3.45 (m, 6H, CH₂CH₂O + SCH₂CH₂), 3.37 (s, 3H, OCH₃), 2.77 (t, 2H, SCH₂CH₂). The degree of functionalization was 97%.

mPEG5000-[S-NB]. Caged mPEG-[S-NB] was synthesized in a two-step reaction. First, mesylation of the terminal alcohol group of methoxy-PEG5000-OH was achieved by reacting 2 g (0.4 mmol) mPEG5000-OH in a mixture of 5 mL anhydrous CH₂Cl₂ with 0.333 mL triethylamine (2.4 mmol) and 155 mL of methanesulfonyl chloride (mesylate chloride) (2 mmol). Secondly, the cage molecule, 64.2 mg (0.3 mmol) was thioacetylated by reacting with potassium thioacetate, 34.3 mg (0.3 mmol) in anhydrous DMF (5 mL). After removal of salts by filtration the solution was concentrated to 15 mL of DMF and 0.9 mL (0.45 mmol) of sodium methoxide was added dropwise for 10 min, followed by the addition of 1 g (0.2 mmol) methoxy-PEG5000-mesylate. After stirring for 2 days at room temperature, polymer was precipitated in cold Et₂O, then dissolved in CH₂Cl₂, reacted with activated charcoal, filtered and precipitated in cold Et₂O. ¹H NMR (CDCl₃): 7.97–7.94 (d, 1H, CH_{aromat}), 7.57–7.49 (m, 2H, 2 \times CH_{aromat}), 7.43–7.41 (m, 1H, CH_{aromat}), 4.13 (s, 2H, CH₂S), 3.82–3.44 (m, 6H, CH₂CH₂O + SCH₂CH₂), 3.37 (s, 3H, OCH₃), 2.61 (t, 2H, SCH₂CH₂). The degree of functionalization was 96%.

mPEG5000-[S-DMNB]. 108 mg (0.022 mmol) mPEG5000-SH was dissolved in a mixture of 10 mL DMF and 10 mL phosphate buffer pH 7.4 (100 mM) that had been degassed before with argon. Through the septum was added a solution of 30 mg (0.11 mmol) DMNB in 0.5 mL (degassed) DMF. After stirring for 32 h at room temperature the solution was dialyzed (MWCO 1000) against 2.5 L milliQ water for 3 days with regularly replacing the water. The solution was filtered (0.22 μm) and the polymer recovered by lyophilisation to yield 100 mg of an orange solid. ¹H NMR (CDCl₃) 7.62 (s, 1H, CH_{aromat}), 6.99 (s, 1H, CH_{aromat}), 4.17 (s, 2H, CH₂S), 3.98 (s, 3H, OCH₃), 3.94 (s, 3H, OCH₃), 3.82–3.45 (m, 6H, CH₂CH₂O + SCH₂CH₂), 3.37 (s, 3H, OCH₃), 2.66 (t, 2H, S-CH₂CH₂). The degree of functionalization was 92%.

4arm-PEG10000-[S-DMNB]. 246 mg (0.1 mmol thiol groups) 4arm-PEG-SH was dissolved in a mixture of 15 mL DMF and 15 mL HEPES (100 mM) buffer solution containing 1 mM Na₂EDTA which had been degassed before with argon. To the

clear solution was added 62 mg (0.22 mmol) TCEP-HCl under a flow of argon. This step was done to reduce any disulfide bonds present in the 4arm-PEG. After stirring for 1 h at room temperature a solution of 206 mg (0.74 mmol) DMNB in 1.0 mL degassed DMF was added. After 16 hours the mixture was extracted with CH_2Cl_2 (3×25 mL) and the combined extracts dried over Na_2SO_4 . The solution was concentrated to about 15 mL and added to 500 mL Et_2O . After filtration this yielded 241 mg (98%) of an orange solid. For the cell experiments the 4arm-PEG-DMNB was dissolved in 10 mL milliQ H_2O and dialyzed (MWCO 3500) against 5 L milliQ H_2O for 2–3 days with regularly replacing the water to remove residual impurities like solvent that might be harmful to the cells. ^1H NMR in CDCl_3 δ = 7.61 (s, 1H, $\text{CH}_{\text{aromat}}$), 6.99 (s, 1H, $\text{CH}_{\text{aromat}}$), 4.16 (s, 2H, CH_2S), 3.99 (s, 3H, OCH_3), 3.93 (s, 3H, OCH_3), 3.83–3.44 (m, 6H, $\text{CH}_2\text{CH}_2\text{O} + \text{SCH}_2\text{CH}_2$), 2.66 (t, 2H, SCH_2CH_2). The degree of functionalization was 84%.

Analytical method to study uncaging kinetics based on HPLC. To determine uncaging kinetics, UV-illumination was performed using UV mercury lamp (100 W Hg lamp). Samples of 100 μL of 200 μM stock solutions of each caged mPEG-[S-] prepared in water and 5% DMSO with 1 mM DTT were exposed to light in the quartz cuvette at a constant distance (25 cm) from the light source. Illumination was performed for different time points (0, 5, 10, 15, 20, 30 and 60 seconds). Photocleavage of the cages was analyzed by RP-HPLC using an analytical C18 column. This analysis revealed the light-exposure dependent decrease in peak corresponding to caged PEG compounds. Each time of exposure time was triplicate ($n = 3$).

The uncaging efficiencies for all the caged PEG compounds was determined based on area of peaks corresponding to caged PEG illuminated for indicated time with UV light. The data of photodeprotection was fitted to the first-order rate equation⁴²:

$$[c]_0 - [c]_t = [c]_0 (1 - e^{-kt})$$

where $[c]_t$ is the area of caged peptide peak intensity at certain time of UV illumination, $[c]_0$ at the time zero – no UV light; k is the rate constant for deprotection, t is the time point of illumination in seconds. Consequently, we obtained k values for each cage, corresponding to a half-lives of photolysis according to the equation:

$$t_{1/2} = \ln(2)/k$$

Preparation of premix solution for formation of microgels. The premix solution was prepared by mixing a stoichiometrically balanced two precursors: 8arm PEG-VS and caged-4arm PEG-thiol in 0.3 M HEPES at pH 7.9 at a final concentration of 5% w/v. This solution was cast as a drop between quartz glasses separated by 0.5 mm thick spacers. The surface of one of the glasses was siliconized by Sigma-coating (Sigmacote, SL-2) to be hydrophobic and not adhesive to the PEG whereas another glass was left without any modification. Importantly, the prepared premixed solution did not gelled without light exposure.

Formation of stiffness patterns. First, the standard Michael-type addition-based hydrogel was formed by mixing 50% molar excess of PEG-VS over PEG-SH in HEPES buffer at pH 7.9. Additional caged PEG-[S-] was added to the hydrogel as a soluble macromer which compensates for the 50% molar stoichiometric imbalance of PEG-SH in the hydrogel. Upon light illumination the cage was released, resulting in activating soluble PEG-[S-] to crosslink to the MT-based hydrogel network, only in the irradiated locations.

General set-up for laser-assisted hydrogel patterning. An upright point-scanning confocal microscope (Zeiss, 710 LSM) equipped with a 405 nm diode laser (30 mW) and with EC Plan-Neofluar 10x/0.30 objectives was employed for local illumination of gels. Bleaching mode and user-defined Region of Interest (ROI) scanning option of the microscope software (ZEN2009) enable arbitrary patterns in x - and y -coordinates of photoactivatable hydrogel. Precise control of specific laser intensities assigned to each ROI allowed the generation of discrete gradients, for example.

Fluorescent visualization of hydrogel patterning. In order to fluorescently visualize the obtained microgels and stiffness patterns, hydrogels were immersed in a solution of 100 μM Alexa488-maleimide in PBS at pH 7.4 and allowed to react with free thiols for 3 hours at room temperature. After rinsing the hydrogel in water to remove unbound dye, confocal fluorescent microscopy was employed to visualize patterns and 3D-labeled microgels.

AFM for probing stiffness patterns in hydrogels. In order to measure the stiffness of the micropatterned PEG hydrogels, direct AFM indentations was conducted using force mode.⁴³ We probed an area of $100 \times 100 \mu\text{m}$ in the middle of each hydrogel pattern with a 5×5 indentation array with 20 μm spacing. We analysed each indentation force-response curve using a point-wise method, such that the depth-dependent apparent elastic modulus was computed as a function of indentation depth. For calculation of the point-wise apparent elastic modulus, we utilized JPK software while setting measurements for modified Hertz equation with spheroid cantilever geometry.

Cell culture and fluorescent nuclei staining. Human MSCs previously isolated from placenta were cultured in DMEM with Glutamax supplemented with 1 mM sodium pyruvate, Hepes pH 7.5, 1% Pen/Strep, 1 ng ml^{-1} hFGF and 10% fetal calf serum (FCS).⁴⁴ Cells were used between passages 7 and 14. At least 16 hours prior to seeding, Celllight Histone 2B-RFP (Invitrogen) was added at a concentration of 240 particles per cell.

Cell culture on top of stiffness-patterned hydrogels. Cell-light-stained hMSCs were seeded on the top of patterned hydrogels in standard growth medium. Cell adhesion was facilitated by homogeneously coupling of 1 mM RGDSP peptide GRGGRGDSPEPCG to the hydrogel during crosslinking *via* its thiol moiety. Cell movements across the stiffness patterns on the hydrogel was tracked by acquiring fluorescence (TRITC filter) and phase contrast images at $5\times$ magnification every hour for 36 hours starting three hours after cell seeding. An

inverted microscope (Zeiss, Till4, Visitron, USA) with motorized stage and environmental chamber (5% CO₂, 37 °C) was used. The resulting images were compiled into a stack and cell behaviour was analysed by ImageJ (Fiji).

Cell trajectory analysis. Single-focus movements of fluorescent nuclei were tracked by ImageJ using the TrackMate plugin. Cell trajectories were imported into Matlab and analysed by a customized Matlab script (kindly developed by the

Bioimaging & Optics platform at EPFL) for the displacement orientation angle (*i.e.* the angle between the direction of the net displacement of a cell and the long side of the designed regions of interest (ROIs) of the gradient), mean velocity, directionality (*i.e.* the ratio of the net displacement to the total path length, the more the cell movement tends towards one direction the more the value comes close to unity).⁴⁵ The positional assignment of the cells in respect to the partitioned areas was

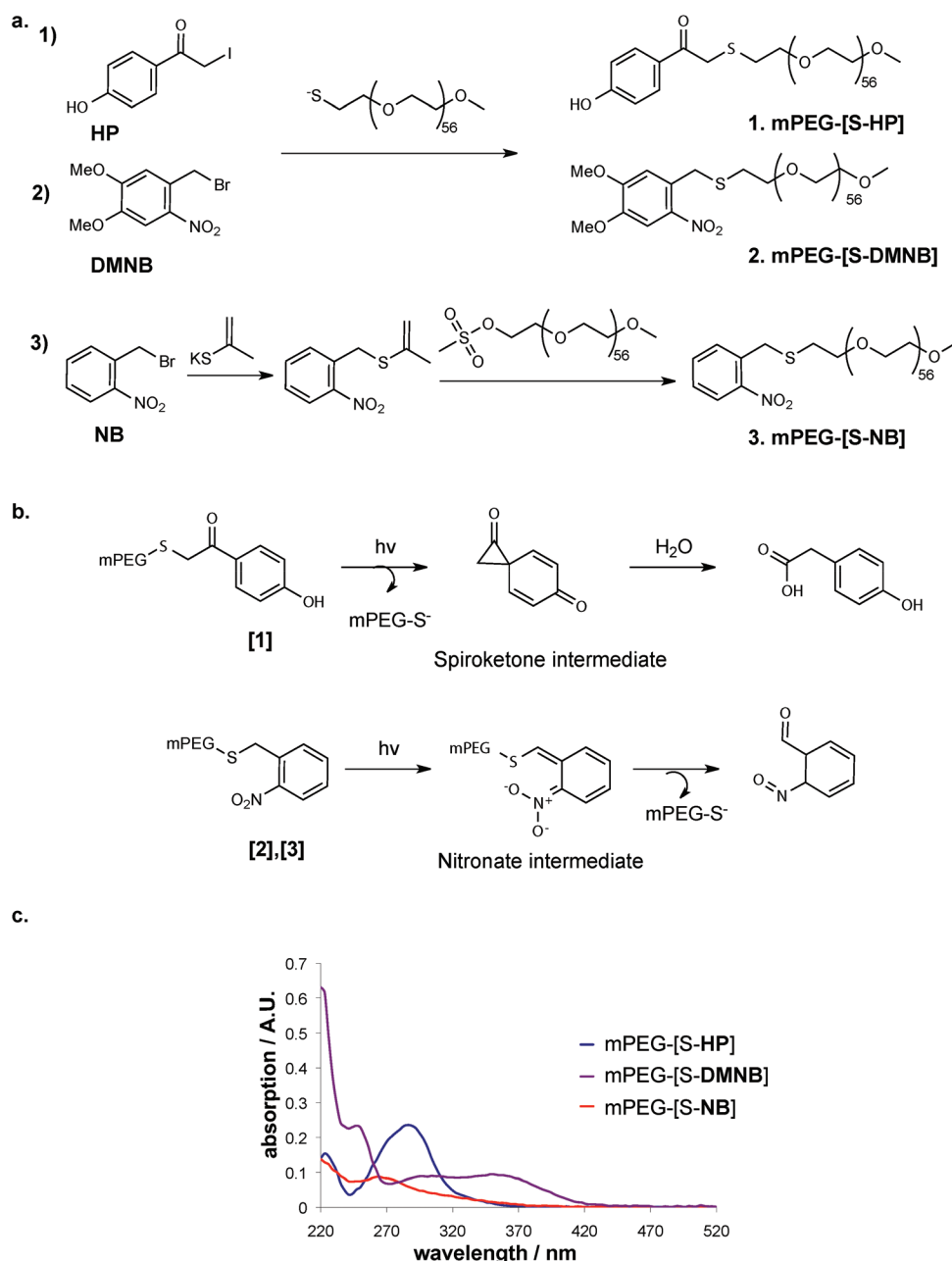


Fig. 2 Soluble models of caged linear methoxy-PEG-[S-]; a, synthetic schemes involving the caging groups (1): 1-(4-hydroxyphenyl)-2-iodoethanone (HP), (2): 1-(bromomethyl)-2-nitrobenzene (NB), (3): 1-(bromomethyl)-4,5-dimethoxy-2-nitrobenzene (DMNB); b, simplified mechanism of photolysis reaction for hydroxyphenacyl group (upper reaction) and nitrobenzene family of cages (lower panel); c, measured UV absorption spectra of the cages.

determined by the mean position of the cells. Non-motile cells and cells migrating for less than 12 hours were excluded.

Results and discussion

Selection of caging chemistry for rapid photo-release in the visible light absorption spectrum

We chose three thiol-caging groups based on the key criteria of rapid photo-release kinetics, in order to obtain patterning without extensive illumination of the hydrogel sample, as well as cell-friendly, near UV-VIS wavelengths such as to be able to apply the system in the presence of cells with good depth penetration. Consequently, caging groups with promising characteristics for hydrogel patterning applications were chosen based on the criteria $\phi\epsilon > 100$ (with ϵ = extinction coefficient, corresponding to the strength of light absorption at a given wavelength; ϕ = quantum yield, corresponding to the efficiency of a caused effect by absorbed light).^{46,47}

The cages *o*-nitrobenzyl (NB) and its derivative dimethoxy nitrobenzyl (DMNB) (Fig. 2a) belong to the *o*-nitrobenzyl family and are the most commonly used photo-labile molecules for controlling biochemical processes. Their main advantages over other cages are the fast deprotection rate and the sensitivity to near UV-VIS. Furthermore, NB is reported to possess higher deprotection rate but at lower wavelengths, while DMNB was reported to have excellent absorption in the range of near UV-VIS light.^{31,47–49}

The third caging group that was selected belongs to the family of *p*-hydroxyphenacyl (HP) cages. Specifically, we used hydroxyphenacyl iodide (Fig. 2a) that is suitable for conjugation to PEG due to iodide as a very good leaving group.⁵⁰ HP cages have an advantageous photolytic chemical mechanism compared to other cage, in which the photoactivation happens immediately in the very first step of photolysis (Fig. 2b).

We first investigated the rate of photo-release of these cages when attached to a mono-functional methoxy-PEG-thiol (mPEG-SH, 5 kDa) as a soluble model system for the cross-linked PEG hydrogel. The different cages were conjugated to

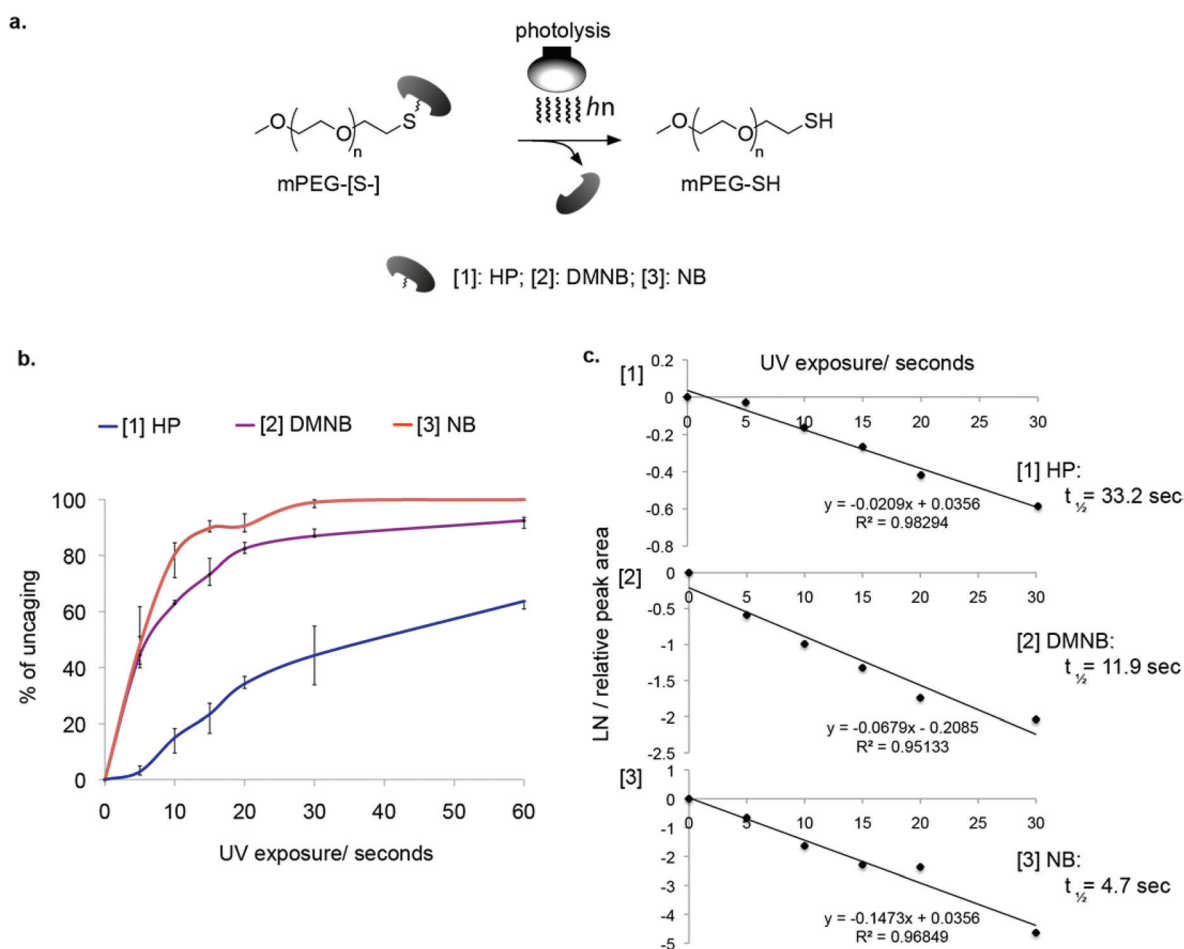


Fig. 3 Model for studying photolysis kinetics of the caged-PEGs; a, Schematic of photolysis leading to reactive thiols; b, Uncaging kinetics determined by RP-HPLC, revealing the rate of photo-release of the cages (– HP, [2] – NB, [3] – DMNB); c, Analysis of photo-release kinetics fitted to first-order rate equation. The slope of the linear regression gives a value for $-k$ and, consequently, half-life of photolysis was obtained ($t_{1/2} = \ln 2/k$) corresponding to HP: $t_{1/2} = 33.2$, NB: $t_{1/2} = 4.7$, DMNB: $t_{1/2} = 11.9$ seconds.

the sulfhydryl group of the mPEG-SH (Fig. 2a) leading to caged mPEG-SH (mPEG-[S-]). When the UV absorption of the obtained caged linear PEGs was compared spectrophotometrically (Fig. 2c), a broad absorption spectrum of DMNB reaching visible light was observed, whereas in the case of HP high absorption was only detected around 300 nm.

To analyze the kinetics of light-induced cleavage of the cages, we utilized an HPLC-based method that allowed quantification of the rate of cage release. Illumination was performed using a UV mercury lamp for different time periods and the samples were injected into an analytical C18 column (Fig. 3a). This analysis revealed the light-dependent decrease in peaks corresponding to caged PEG (Fig. 3b). By measuring the decrease in signal intensity from caged mPEG-[S-] and fitting a linear regression, we obtained the half-lives of photolysis (Fig. 3c, ESI† Table 1). The slowest reaction was

obtained for mPEG-[S-HP] with $t_{1/2}$ of 33.2 seconds. Much better performance was obtained for the nitrobenzyl cages, ranging from $t_{1/2}$ = 4.7 seconds for NB to $t_{1/2}$ = 11.9 seconds for DMNB. Based on these results we chose DMNB for further studies.

Photo-controllable Michael-type addition affords precise spatial control of hydrogel stiffness

To render hydrogel crosslinking *via* Michael-type addition susceptible to light, 4arm-PEG-SH macromers (10 kDa) were modified with DMNB. Mixing of two aqueous solutions containing caged 4arm-PEG-[S-]⁴⁶ and 8arm-PEG-VS at equal stoichiometry of functional groups did not result in any gelation. In contrast, illumination of these precursors with a confocal microscope laser at defined regions of interests (ROI) resulted in microscale hydrogel patches (Fig. 4). This experiment

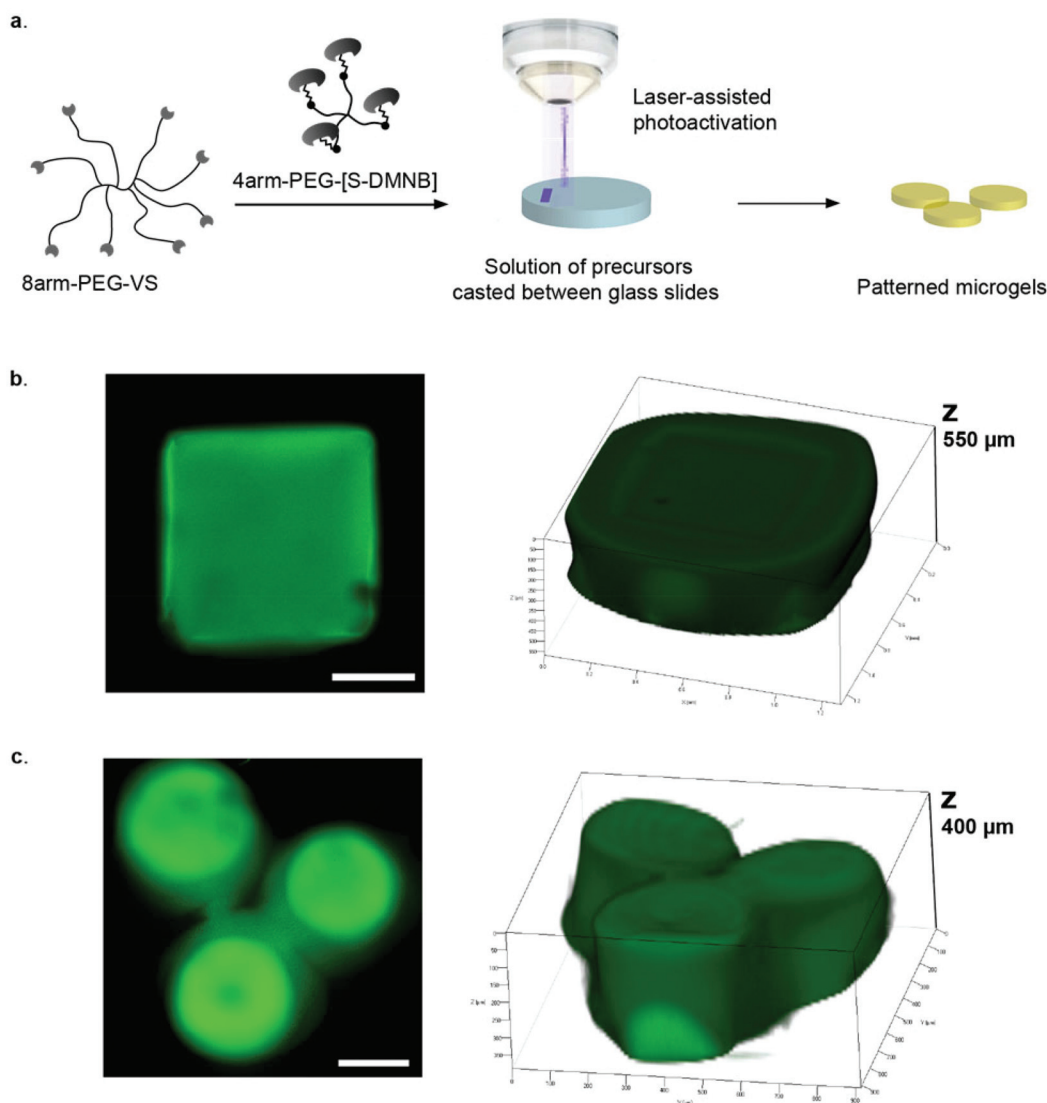


Fig. 4 a, Concept of photo-controllable Michael-type addition for bulk hydrogel crosslinking. An aqueous solution of PEG-vinylsulfone (VS) and caged PEG-[S-DMNB] is non-reactive due to caged thiols; b,c, Fluorescent microgels obtained by spots of LSM confocal laser; b, Cubic hydrogel (scale bar = 300 μm); c, Smaller cylindrical hydrogels (scale bar = 200 μm).

proves that thiol uncaging is very efficient and a combination of photochemistry and Michael-type addition affords precise microscale synthesis of hydrogels. Furthermore, the use of confocal lasers allows for scanning with high resolution practically any designed patterns by simply drawing user-defined shapes. The current setting allows a patterning down to single cell resolution of about $10\ \mu\text{m}$ (data not shown). In theory, the actual limit to the resolution is the objective or the microscope that is used, respectively, and thus can be further increased according to the needs of the user.

We next focused on locally modulating stiffness in already formed hydrogel samples according to the scheme indicated

in Fig. 1. To this end, soluble caged 4arm-PEG-[S-] was added to a pre-made hydrogel having 50% excess vinyl sulfone groups available for crosslinking with equimolar amounts of photo-activated thiols (Fig. 5a). The pre-made hydrogels were then exposed to a confocal laser (405 nm diode laser, 30 mW) using defined laser intensities. Specifically, 100% of the maximal laser power for $840\ \mu\text{m} \times 840\ \mu\text{m}$ square and a series of different intensities corresponding to 12.5, 25, 50 and 100% of the maximal laser power assigned to rectangular ROIs of $420\ \mu\text{m}$ width were used. The time of laser irradiation was equal to the sum of 50 iterations of laser scanning. Notably, patterns were readily visible by phase

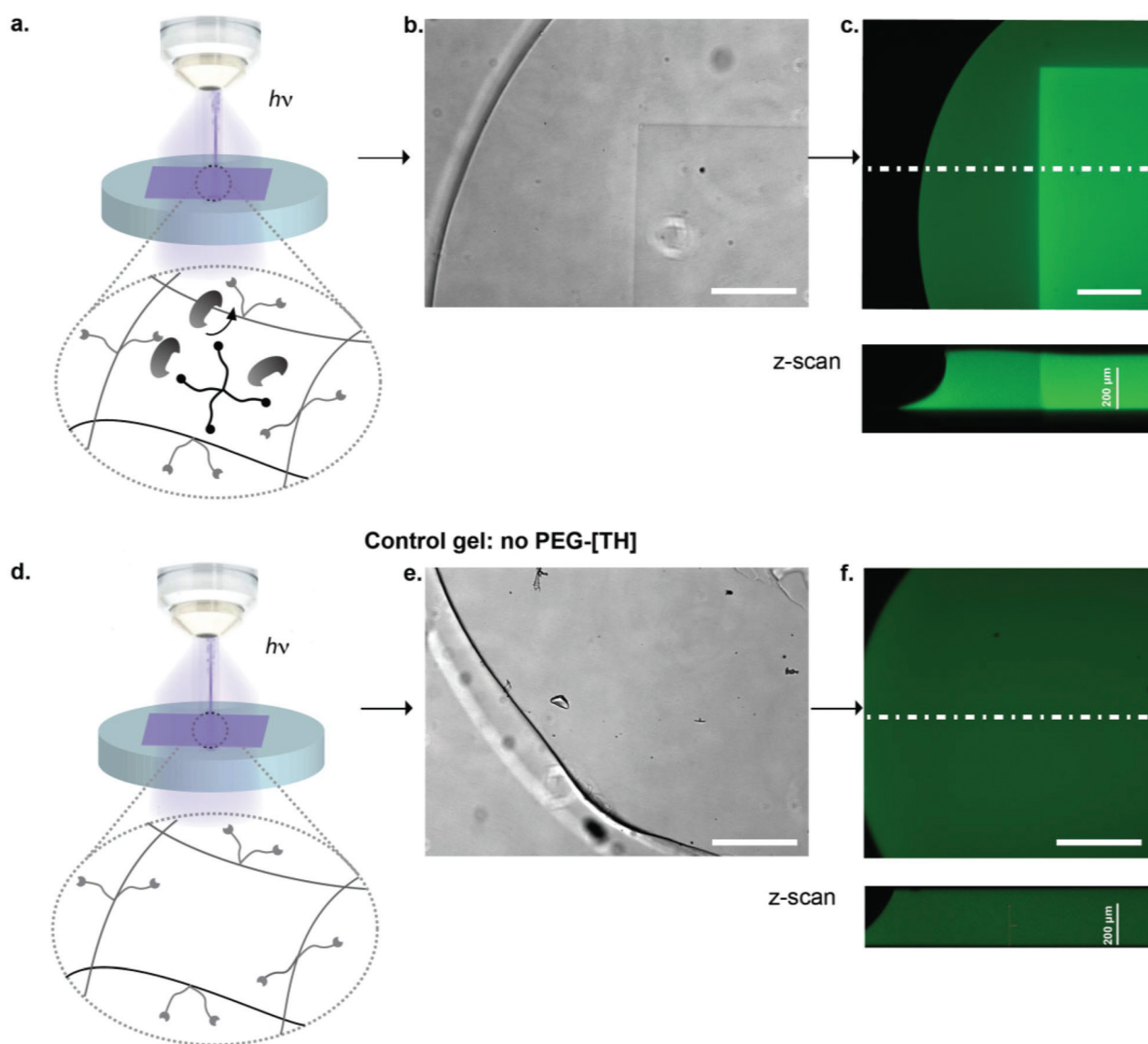


Fig. 5 Generation of physical heterogeneities in hydrogel; a, Schemes of photo-patterning hydrogel crosslink densities to locally stiffen hydrogels; b, Phase contrast image of hydrogel pattern obtained after photo-activated crosslinking; c, Pattern visualization by fluorescent labeling with thiol-reactive, maleimide-functionalized dye; d, Negative control of photo-patterning procedure with Michael-type addition hydrogel that lacks the photo-labile PEG-[S-DMNB] linker during photolysis; e–f, No patterns can be observed. Scale bars: $200\ \mu\text{m}$ in b–e, $300\ \mu\text{m}$ in c–f.

contrast microscopy (Fig. 5b), whereas in a negative control lacking 4arm-PEG-[S-] no pattern was formed. Moreover, it was possible to visualize patterned areas by fluorescently labeling thiols using a maleimide dye and confocal imaging (Fig. 5c). Importantly, the intensity of the fluorescent signal obtained by maleimide-based labeling directly correlated with the laser intensity. Again, no fluorescent pattern was observed in the case of a negative control missing caged 4arm-PEG-[S-]. These data confirm the successful photolysis of caging groups from 4arm-PEG-[S-] macromers in an intensity-dependent manner.

Next, we measured the elasticity of the obtained micropatterns on the surface of PEG hydrogels using atomic force microscopy (AFM).⁴³ We analyzed each indentation force-response curve with a point-wise method, such that the depth-dependent, apparent elastic modulus could be computed as a function of indentation depth using a modified Hertz model.⁵¹ These measurements showed a discrete microstiffness gradient, ranging from 3.3 (± 0.1) to 8.2 (± 0.3) kPa measured in micro-squares of $420\ \mu\text{m} \times 420\ \mu\text{m}$ (Fig. 6). We observed minimal stiffness variation within each section of the discrete gradient. Therefore, patterning gels by photo-controlled Michael-type-addition is not limited to the production of simple binary or linear changes in stiffness, as stiffness patterns of practically any desired shape and stiffness range can be used.

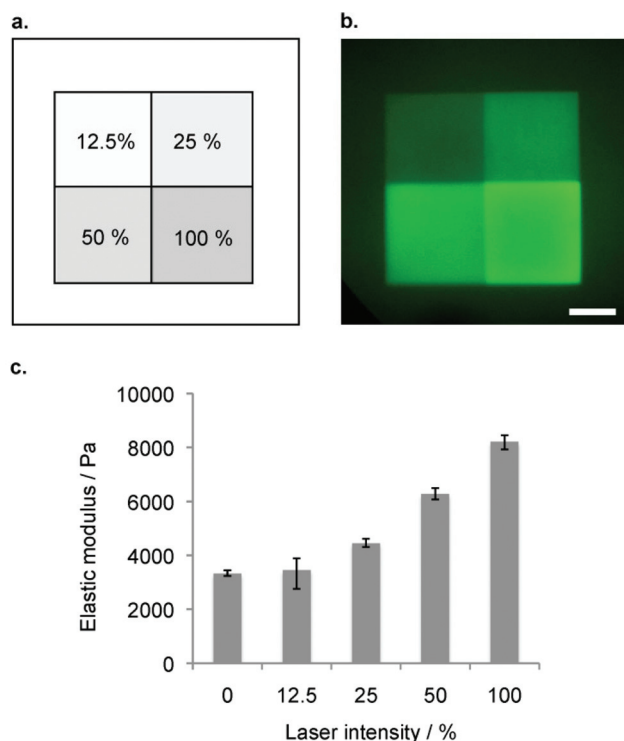


Fig. 6 AFM to quantify local hydrogel stiffness. a, Design of regions of interest (ROIs) for laser exposure; b, Hydrogel patterns in hydrogel visualized fluorescently by labeling with a fluorescent dye (scale bar = $200\ \mu\text{m}$); c, Obtained mean values of elastic moduli measured by AFM.

Directed hMSC migration on photomodulated stiffness gradients

To demonstrate the utility of our system for biological applications, we tested the hydrogel stiffness-patterning platform in a cellular context. We examined the influence of local gel crosslinking density on the migratory behavior of hMSCs. It has been well established that these cells can respond to substrate elasticity by changing morphology, motility and even lineage specification.^{21,22,52} Lo and others have already demonstrated that many different cell types undergo durotaxis, that is, the directed migration along a stiffness gradient.^{22,53–56} Specifically, guiding hMSC migration by mimicking these heterogeneous elasticity features of the substrate by complex and dynamic hydrogel designs might be a valuable tool to study mechanisms of wound healing, morphogenesis and cancer progression, and eventually exploit them in tissue engineering devices.

To probe the migratory behavior of hMSCs on a discrete stiffness gradient, we designed regions of interest (ROIs) with 20 different stiffness ranges (Fig. 7a), with elastic moduli in the range of soft tissues⁷ (between 3 and 8 kPa) and spanning a region of approximately $1000 \times 900\ \mu\text{m}$. In addition, the PEG hydrogels were rendered cell-adhesive by tethering to the gels 1 mM of the fibronectin-derived adhesion peptide RGDSP during initial crosslinking to ensure homogenous distribution. When hMSCs were seeded at a cell density of $1000\ \text{cells cm}^{-2}$, they initially attached homogeneously on the hydrogel surface (Fig. 7b). Single cell migration tracks on the patterns were imaged by time-lapse microscopy and analyzed by the Track-Mate plugin (Fiji) (Fig. 7c and ESI Movie†). The mean velocity was found to be significantly increased on the softer area of the gradient (ESI Fig. 1a†). Overall, hMSCs seeded on the mechanically modulated area of increasing substrate rigidity showed a loss of random motility, *i.e.* cell migration seemed to be more directionally biased in comparison to the softer non-patterned outside (ESI Fig. 1b†). Another means to assess directional migration is the persistence time which was not significantly different in our setup (data not shown).^{57,58} Notably, the assessment of the direction of the displacement by the analysis of the orientation angle of the net displacement revealed that the stiffness gradient of 5.5 kPa onwards was able to elicit durotaxis towards the more densely crosslinked regions (Fig. 7d and e). In contrast, cell migration on the stiffness gradient in the lower range (3–5.5 kPa) showed no significant directional preference. This finding indicated a threshold of about 5.5 kPa to initiate durotactic behavior in hMSCs in our system, *i.e.* in a modulus range-dependent manner with constant gradient strength. These initial data reveal a significant effect of local, gradual change of the crosslinking density on cell migration. Stiffness levels and steepness of a gradient are important regulators in durotaxis of cells and our platform should provide means for the further investigation of these parameters.^{22,55,56}

Taken together, we think that the combination of caging chemistry and Michael-type addition is an efficient strategy for

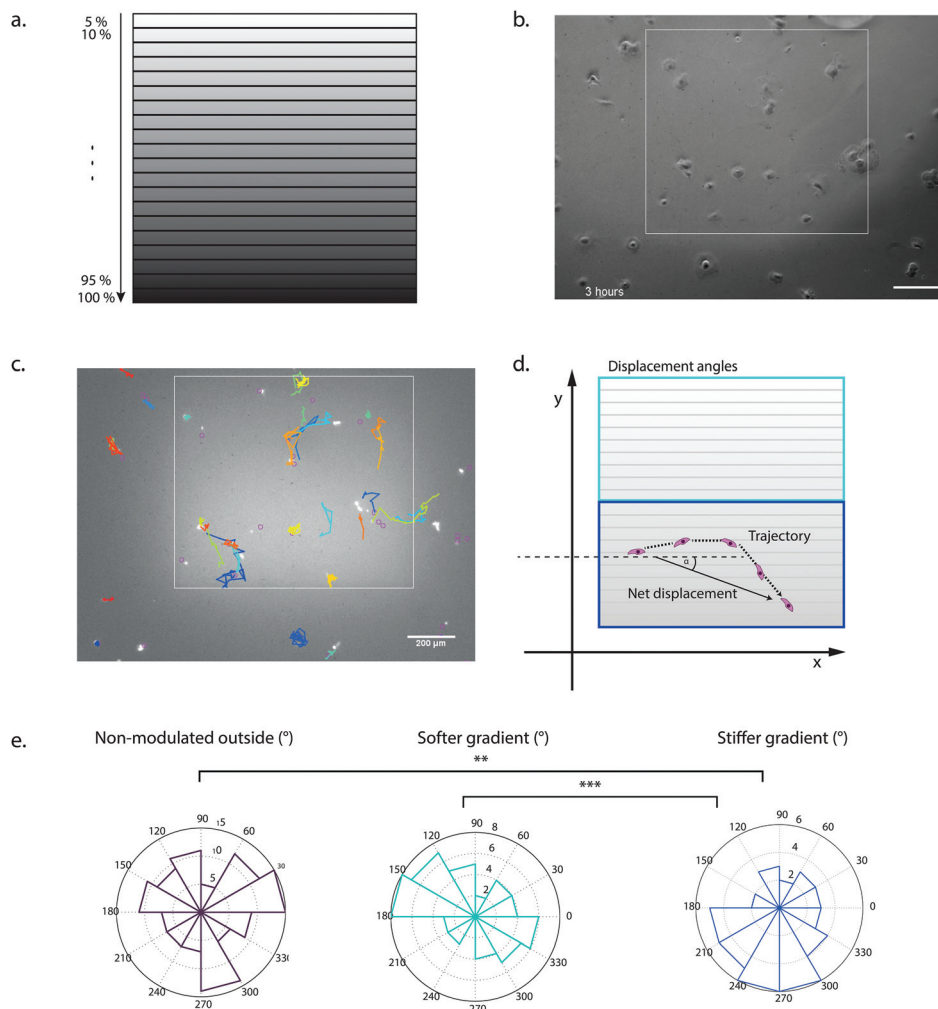


Fig. 7 hMSC behavior on stiffness patterns. **a**, Design of patterns by translation of stepwise increasing laser intensity into regions of increasing stiffness. **b**, hMSCs are cultured on top of photopatterned hydrogels and imaged by time-lapse microscopy for 36 hours. The gradient was visualized by a fluorescent dye and borders are indicated by a white frame. hMSCs initially attached homogenously on the hydrogel surface (Scale bar = 200 μm). **c**, Single cell nuclei are stained by Celllight and are tracked by the TrackMate plugin (Fiji). Single cell tracks are indicated by different colors. **d**, Cell tracks for the non-modulated soft area, on patterned soft areas (3–5.5 kPa, light blue) and stiffer areas (5.5–8 kPa, dark blue) are analyzed. Displacement orientation angle α is defined by the angle between the direction of the net displacement of a cell and the long side of the designed ROIs of the gradient. **e**, Displacement orientation angles are significantly different between the stiffer area and the non-modulated and patterned softer area respectively and mainly oriented towards the stiffer region indicating that the cells undergo durotaxis from 5.5 kPa onwards (** $p < 0.01$, *** $p < 0.001$, error bars represent s.d., $n = 3$ independent experiments, 113, 51 and 42 cells were tracked for the outside, softer gradient and stiffer gradient regions respectively).

the light-triggered control of hydrogel stiffness. This patterning concept should be also useful in the context of 3D cell cultures to manipulate the behaviour of various cell types.

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