



Fracture & Fatigue of Advanced Metallic Alloys

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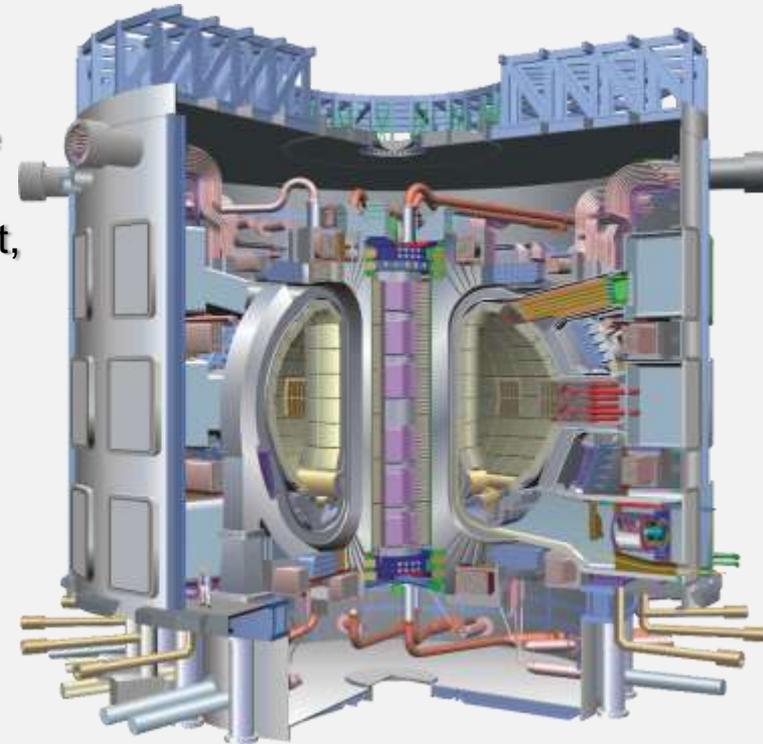
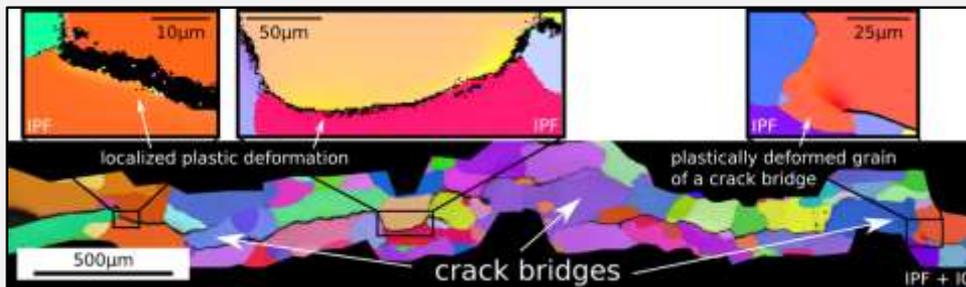


Mechanical Properties of Structural Materials

Past research (Master & PhD theses):

○ Fracture behavior of W-materials

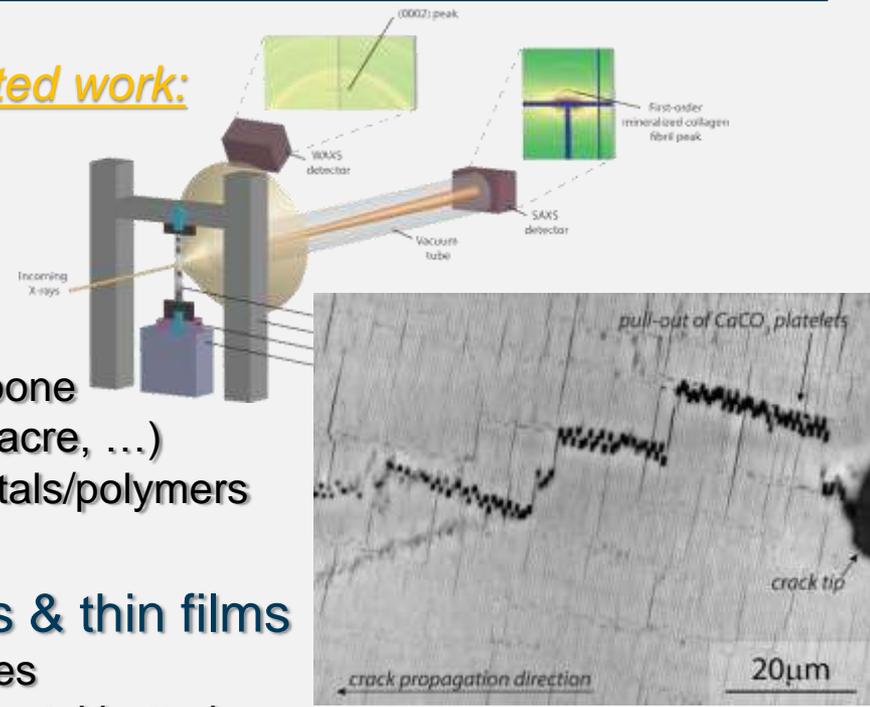
- understanding mechanisms determining brittle fracture at low temperatures & ductile-to-brittle transition
- influence of grain size, dislocation density/arrangement, texture & impurities on fracture mechanism
- fracture behavior in the range RT to 1000C



Current research (Post-doctoral work) – bio-related work:

Mechanical properties of biological & nature-inspired materials

- using *in-situ* VP-SEM & SAXS/WAXD techniques
- understanding effects like aging, strain rate, diseases (OI) and drugs on structural integrity of bone
- investigating natural materials (fish scales, skin, nacre, ...) & mimicking natural structures using ceramics/metals/polymers

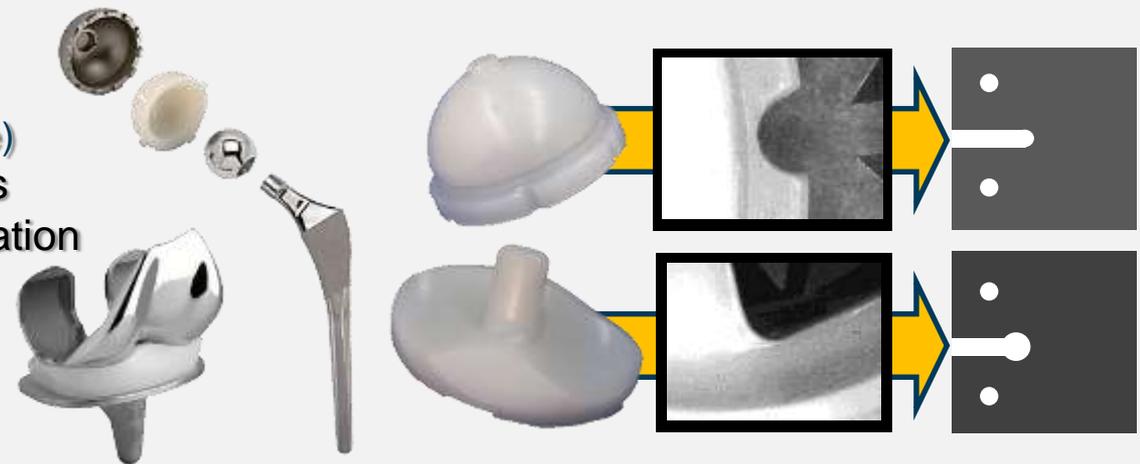


Strength of nano-particle reinforced fibers & thin films

- understanding mechano-optical sensing capabilities
- verifying usability of block copolymers for lithium metal batteries

Fatigue behavior of UHMWPE (ultra-high-molecular-weight polyethylene)

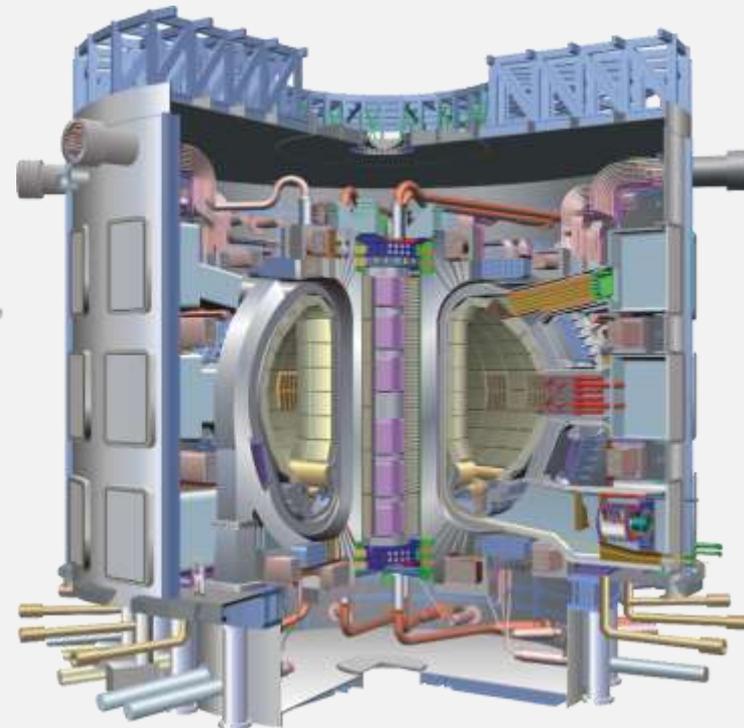
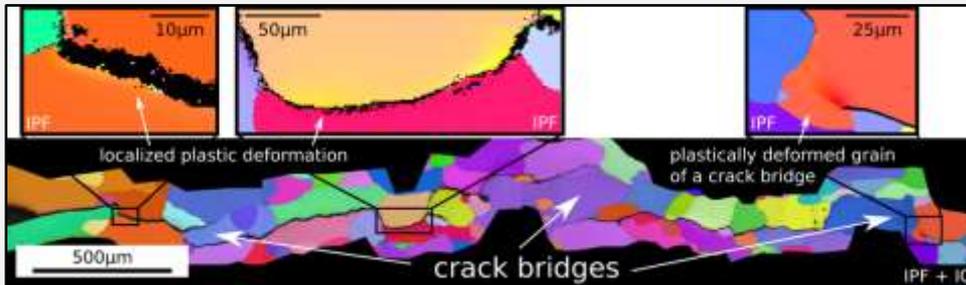
- used in hip and knee replacements
- evaluation of fatigue crack propagation from various notch geometries



Mechanical Properties of Structural Materials

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 - fracture behavior in the range RT to 1000C



Current research (Post-doctoral work) – engineering materials:

- Strength and toughness of nuclear graphite
 - investigating deformation mechanisms and damage tolerance in the temperature range room temperature to 1000C using synchrotron x-ray tomography
- Fracture & fatigue of bulk-metallic glasses (BMGs)
- Fracture behavior of high-entropy alloys (HEAs)



EPFL – Seminar, 1st February 2016

Fracture & Fatigue of Multi-Component Alloys

Fracture and fatigue of bulk-metallic glasses (BMGs)

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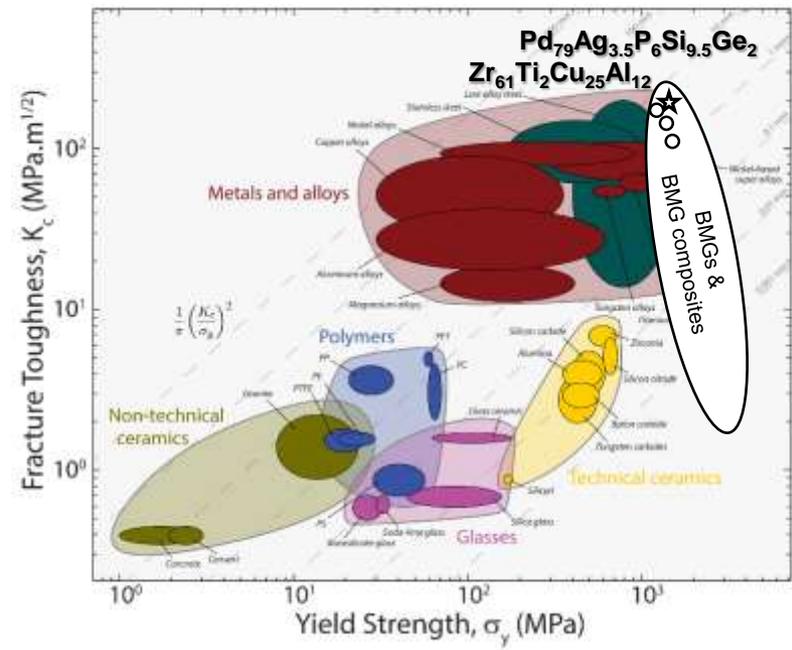
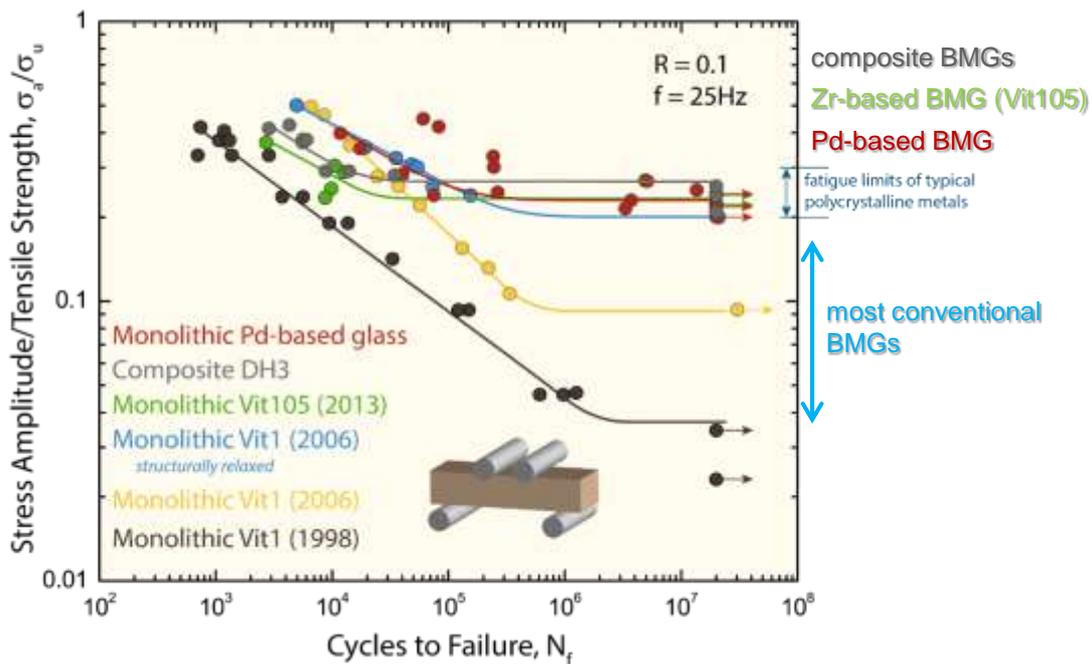
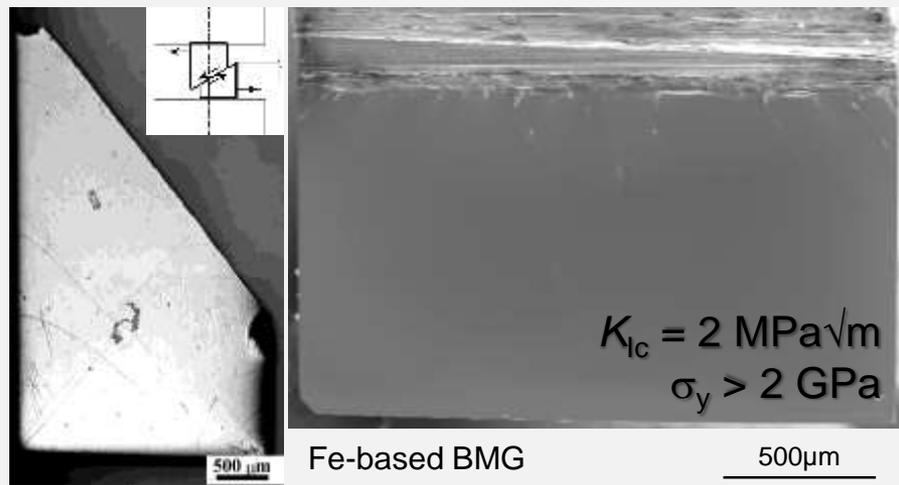
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Mechanical Properties of Metallic Glasses

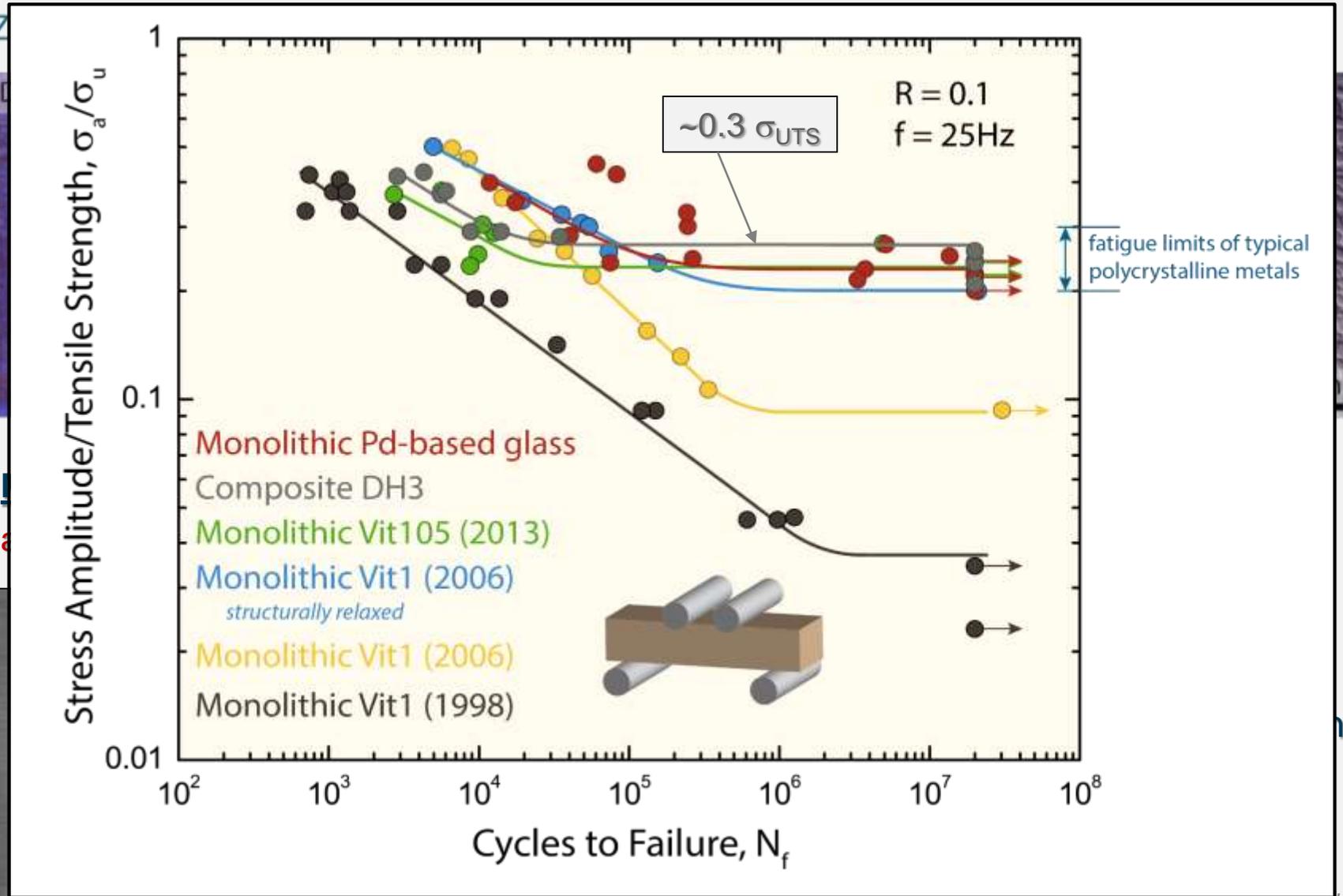
- high strength (0.5 – 4 (5) GPa)
- low stiffness (<100 GPa)
- limited ductility
(tension: ~ 0% / compression: < 2%)
- varying fatigue strength
(0.05 – 0.2 (0.3) σ_{UTS})
- 'OK' fracture properties
(fracture toughness 10 – 100 (200) $\text{MPa}\sqrt{\text{m}}$)





Bulk-metallic glass matrix composites

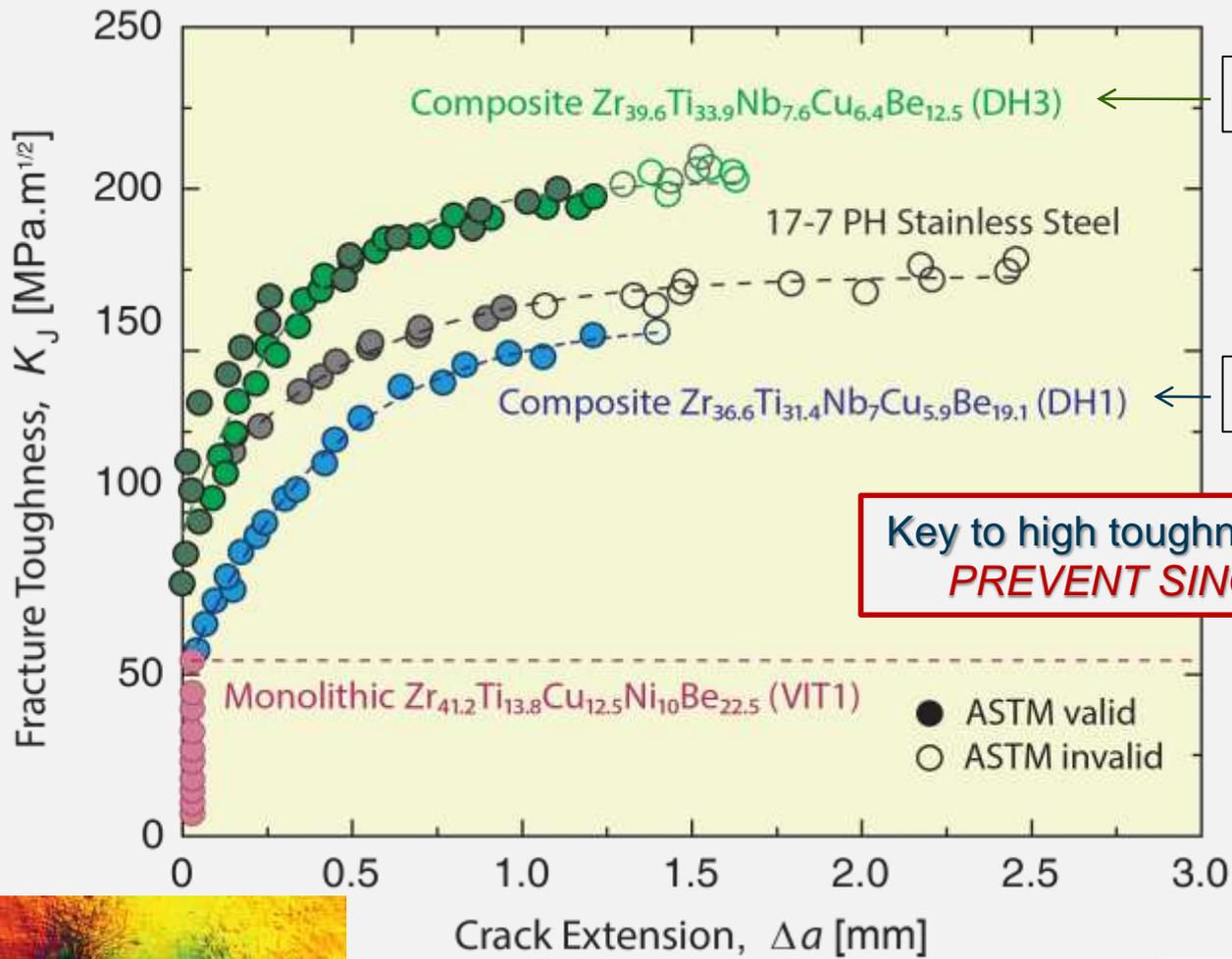
BMG in situ Matrix Composites



Launey, Hofmann, Johnson, Ritchie *PNAS* 2009

Hofmann, Sun, Weist, Duan, Lind, Demetriou, Johnson *Nature* 2008

Toughness of BMG in situ Matrix Composites



$\sigma_y \sim 1.1$ GPa

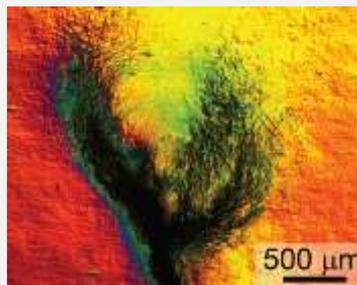
$$K_{JIC} = (J_{IC} E')^{1/2}$$

$\sigma_y \sim 1.5$ GPa

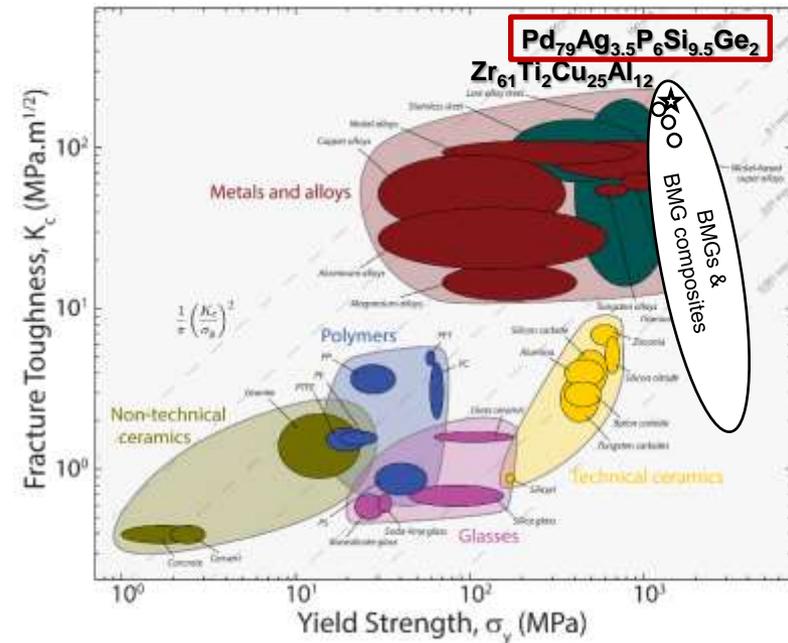
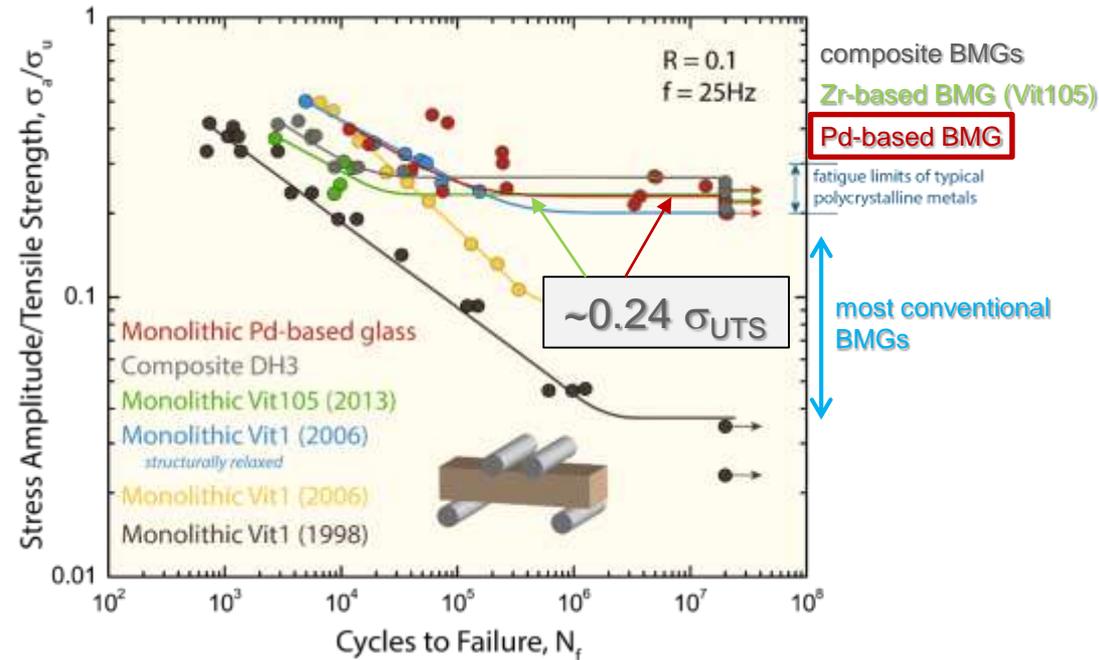
Key to high toughness & excellent fatigue behavior:
PREVENT SINGLE-SHEAR BAND FAILURE

DH3: $K_{JC} = 157$ MPa.m^{0.5}
DH1: $K_{JC} = 97$ MPa.m^{0.5}

BMG composites display very high damage tolerance

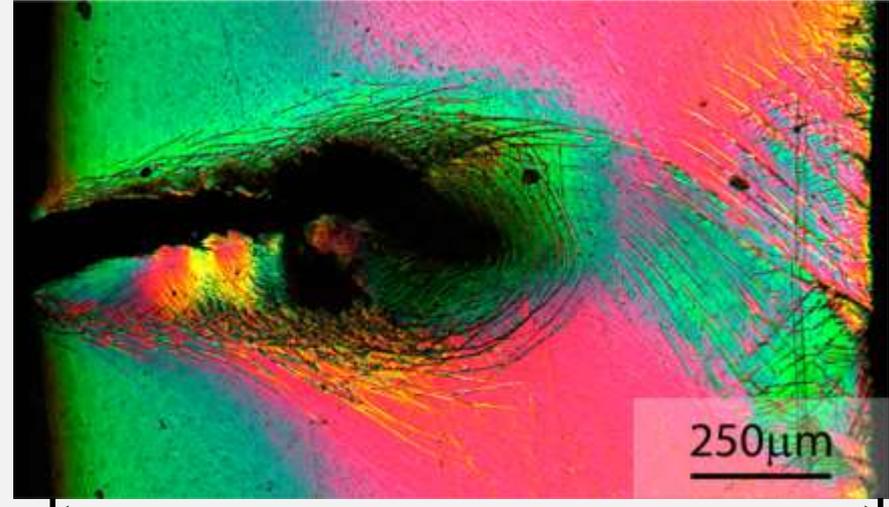
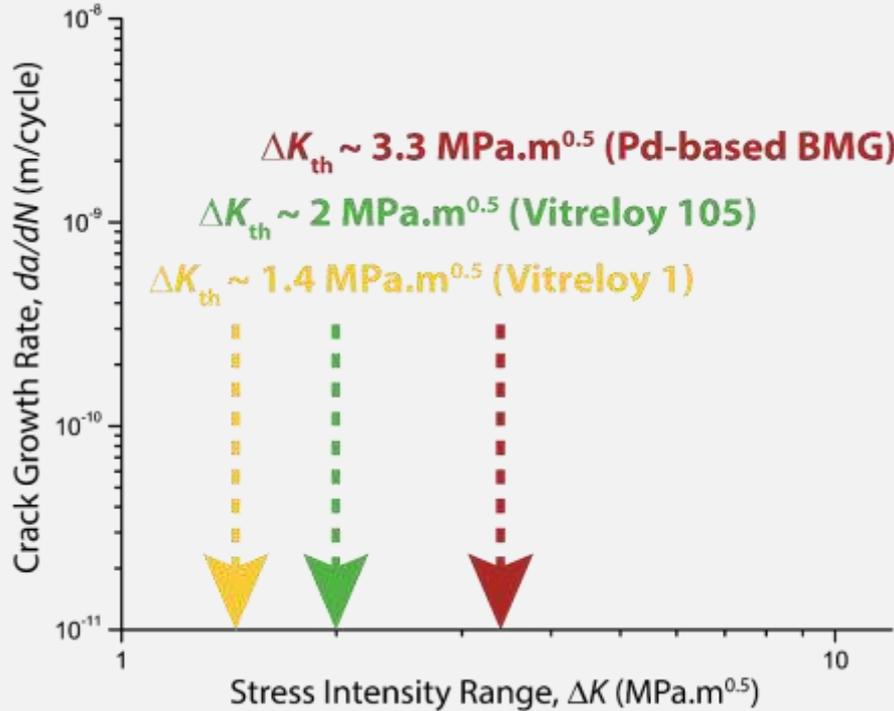


Fully Amorphous Monolithic Glasses

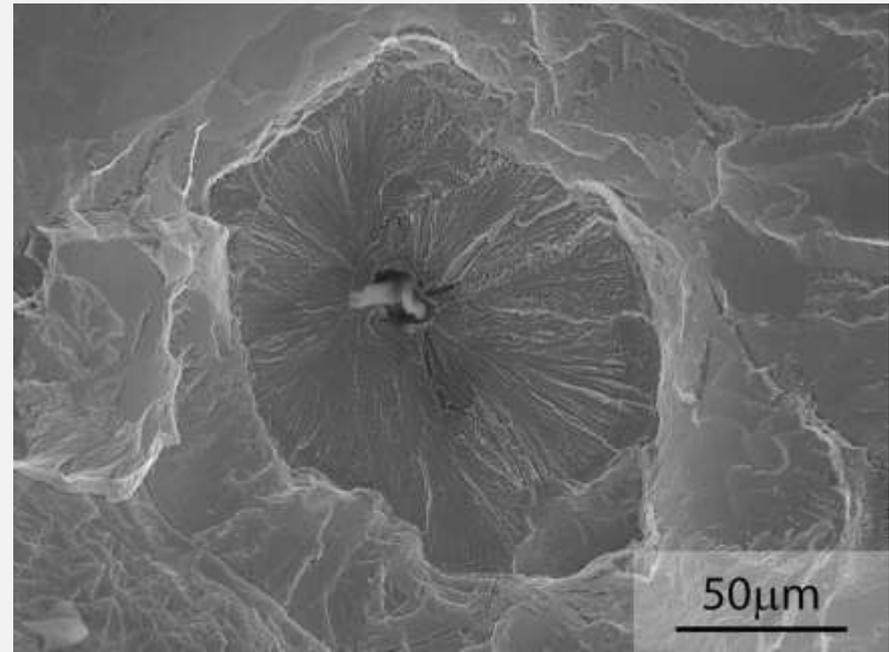


Pd-based BMG: Threshold & flaw sensitivity

high threshold → high endurance limit



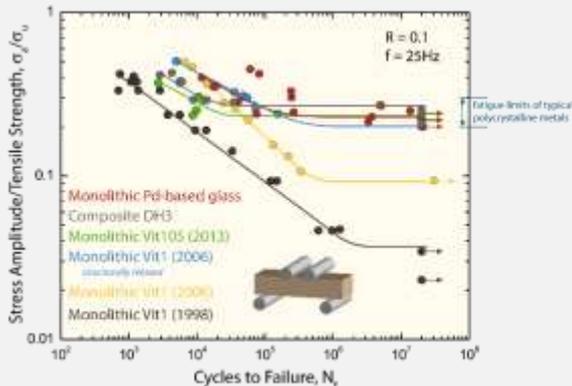
thickness of 4-point bending stress-life (S-N) sample



$E = 88 \text{ GPa}$

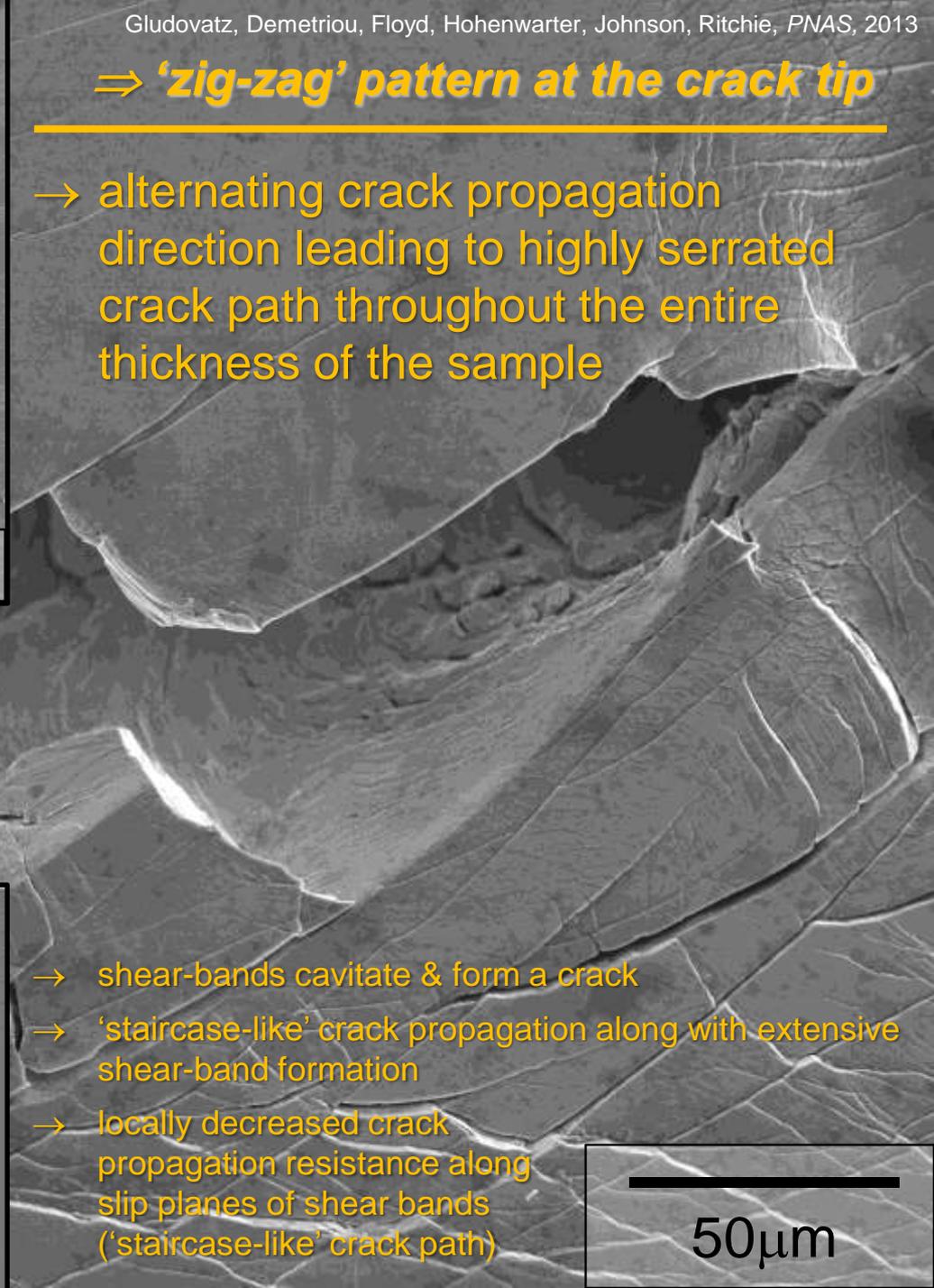
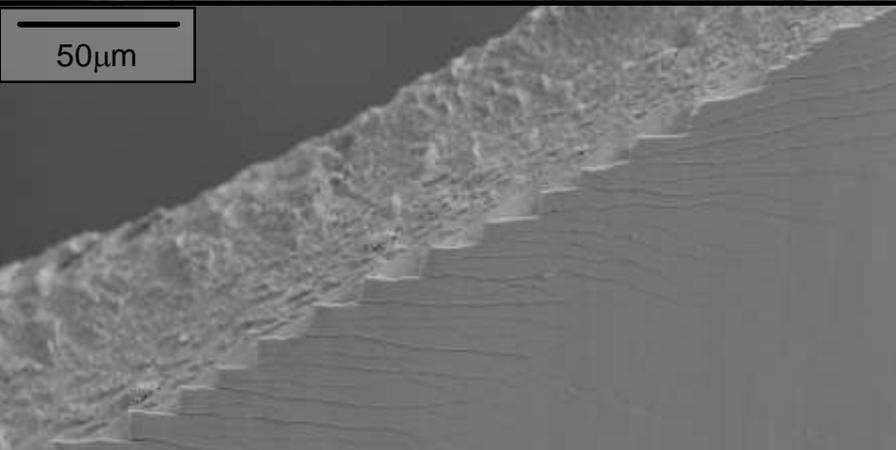
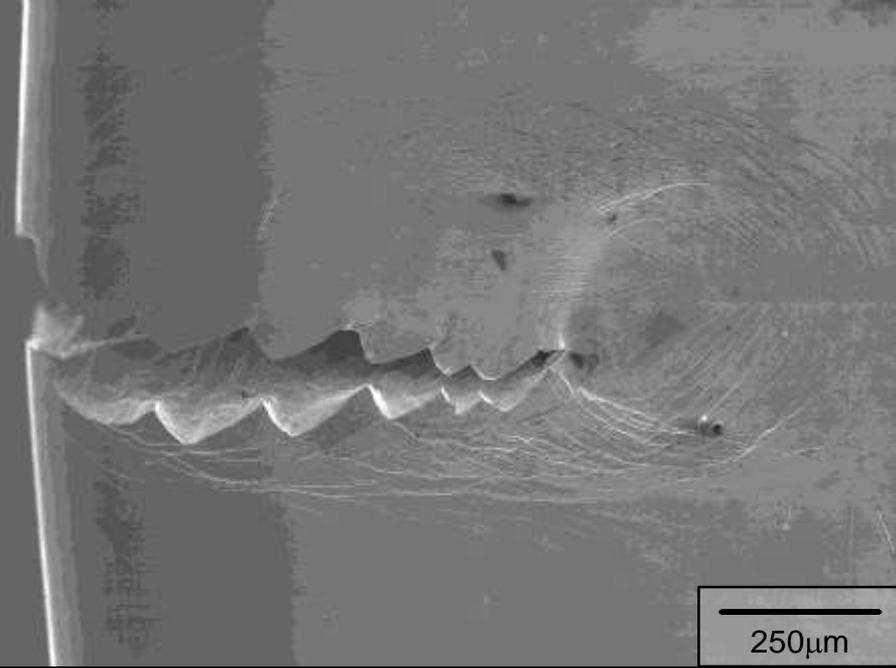
$B = 172 \text{ GPa}$

$G = 31 \text{ GPa}$



⇒ **'zig-zag' pattern at the crack tip**

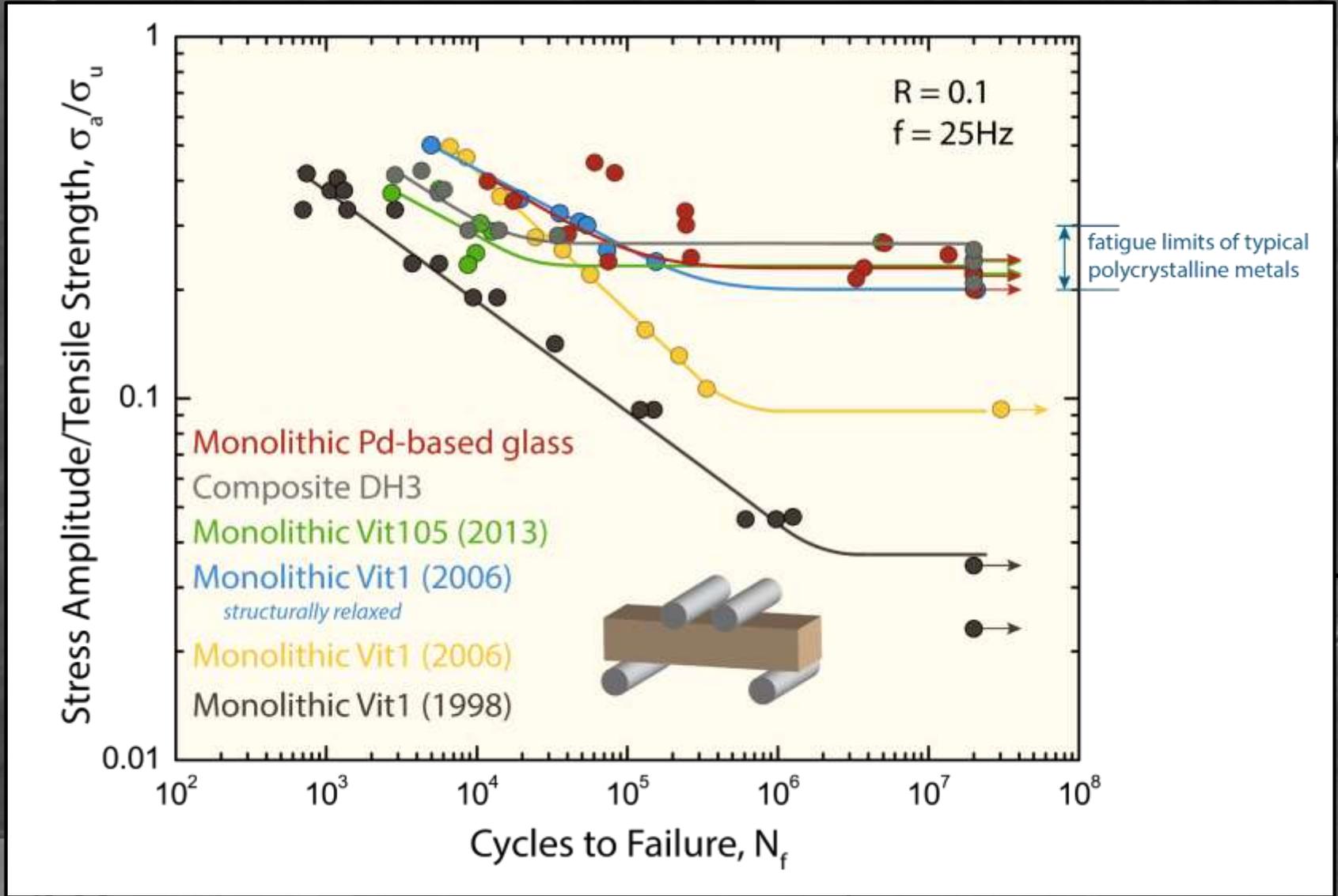
→ alternating crack propagation direction leading to highly serrated crack path throughout the entire thickness of the sample



→ shear-bands cavitate & form a crack
→ 'staircase-like' crack propagation along with extensive shear-band formation
→ locally decreased crack propagation resistance along slip planes of shear bands ('staircase-like' crack path)

50 μm

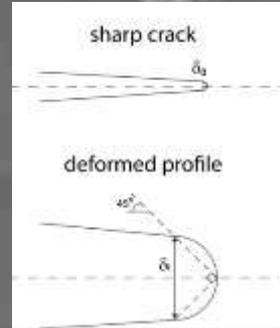
Roughness-induced crack closure



10 μm

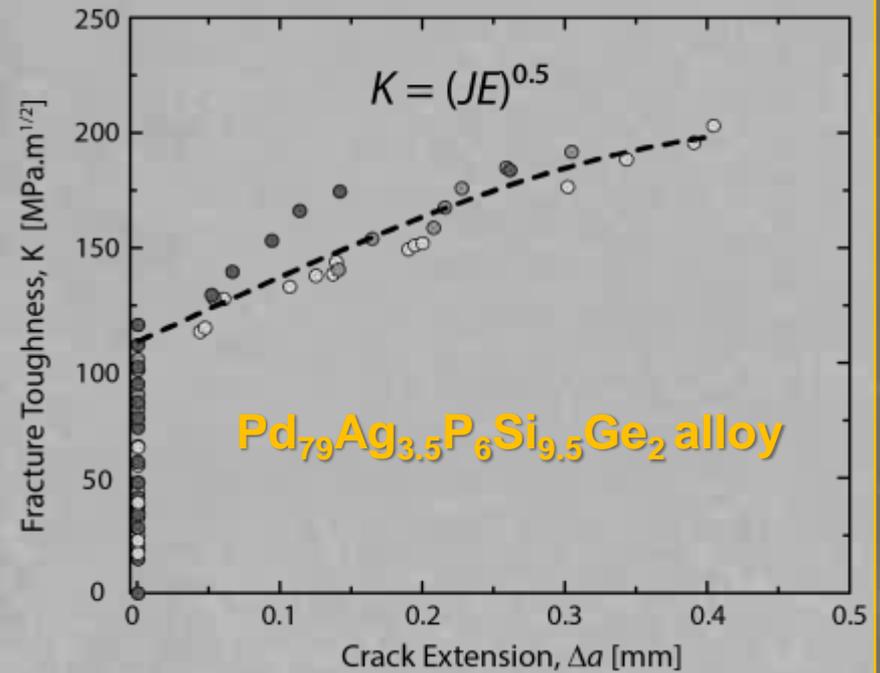
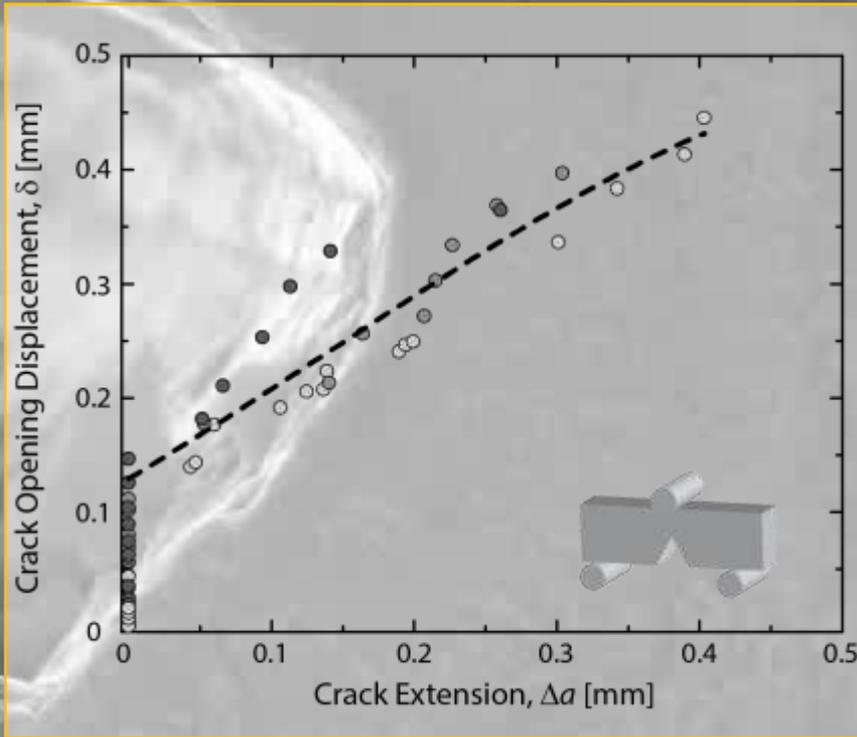
25 μm

- multiple stable shear-band formation and proliferation
 → excessive crack tip blunting and plastic-zone size formation,
 prior to initiation and propagation of shear-band cracks



- akin to general yielding and fully ductile fracture
 in crystalline alloys

- large crack-tip open displacements (CTODs) lead to very high toughness



Pd-based monolithic BMGs

WD16.0mm 15.0kV x3.0k 10um

Current issues with metallic glasses

Major problem: **VARIABILITY** of mechanical properties

- i) Processing
- ii) Mechanical testing

processing & structure-property relation

various processing techniques:

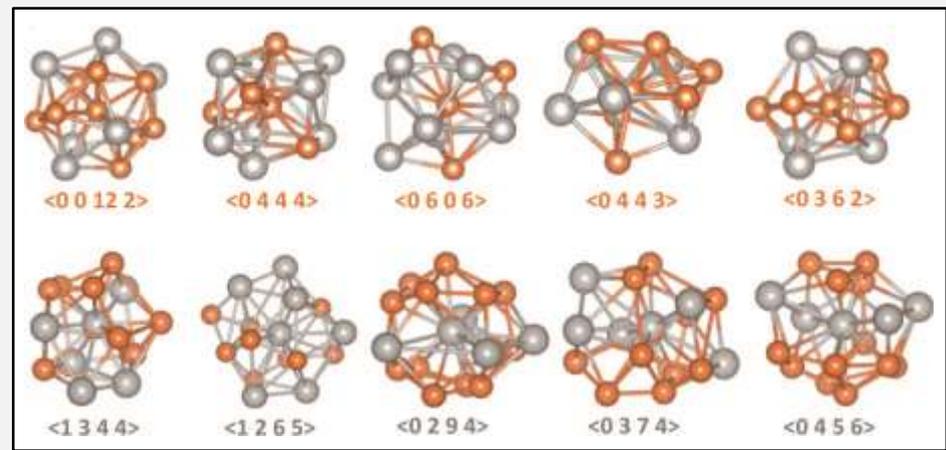
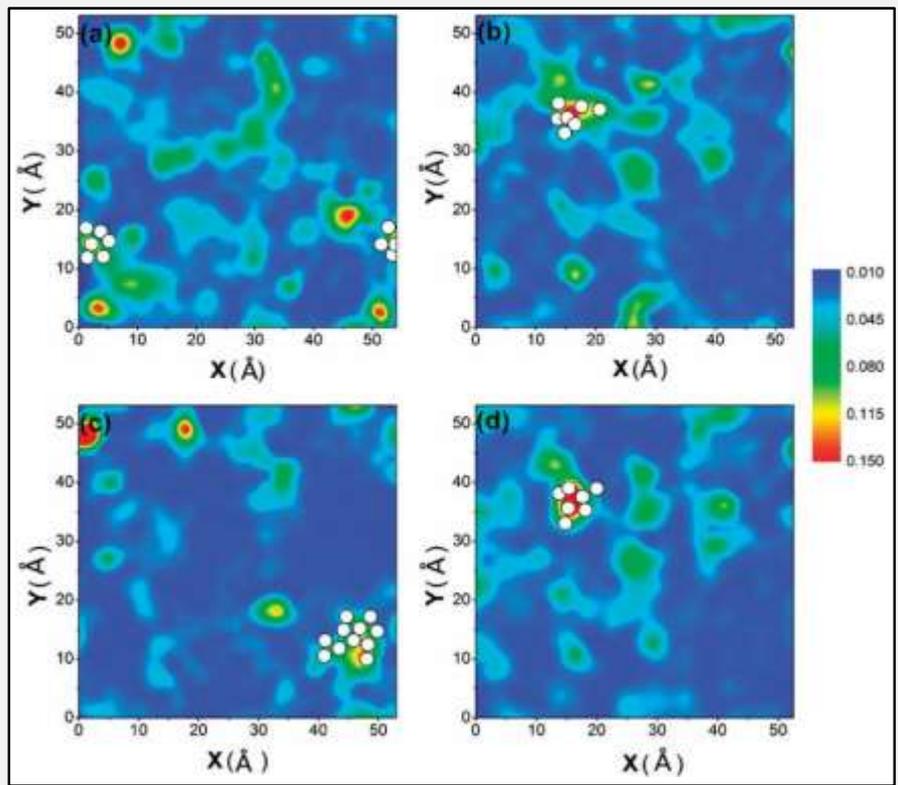
- suction casting
- centrifugal casting
- counter-gravity casting
- thermo-plastic forming
- ...

limitations:

- sample size
- cooling rate
- shape
- ...



Fecht, Johnson, *Mat Sci Eng A*, 2004



- variations in local atomic packing structure
- quasi-localized soft spots
- strongly influence shear deformation

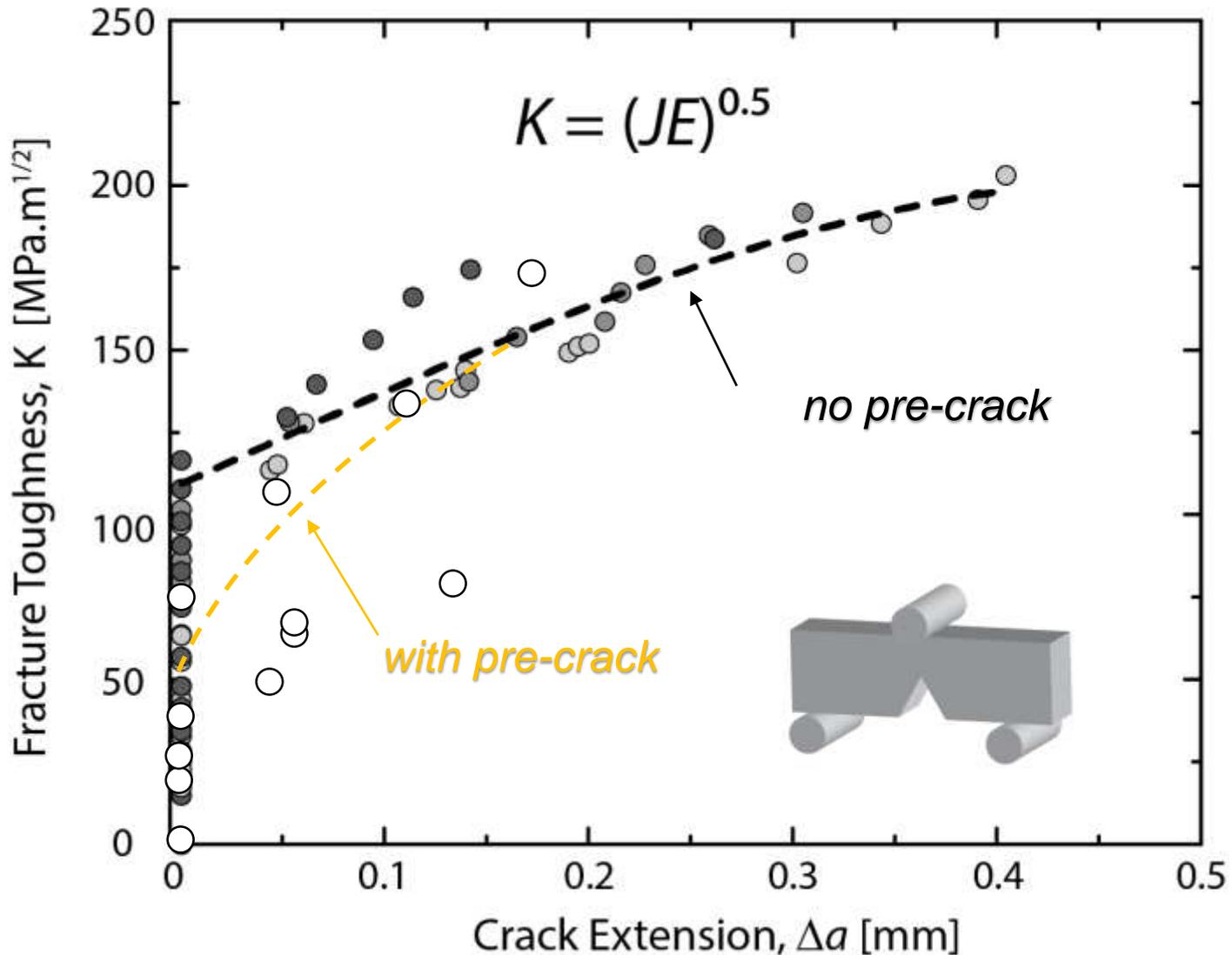
Current issues with metallic glasses

Major problem: **VARIABILITY** of mechanical properties

- i) Processing: - 'structure'-property relation
- ii) **Mechanical testing: - notch sensitivity**

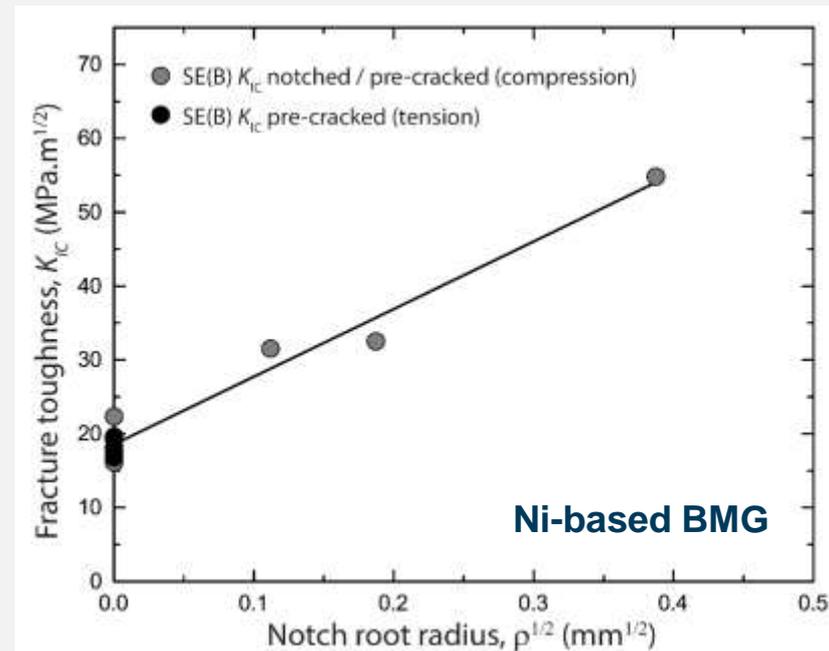
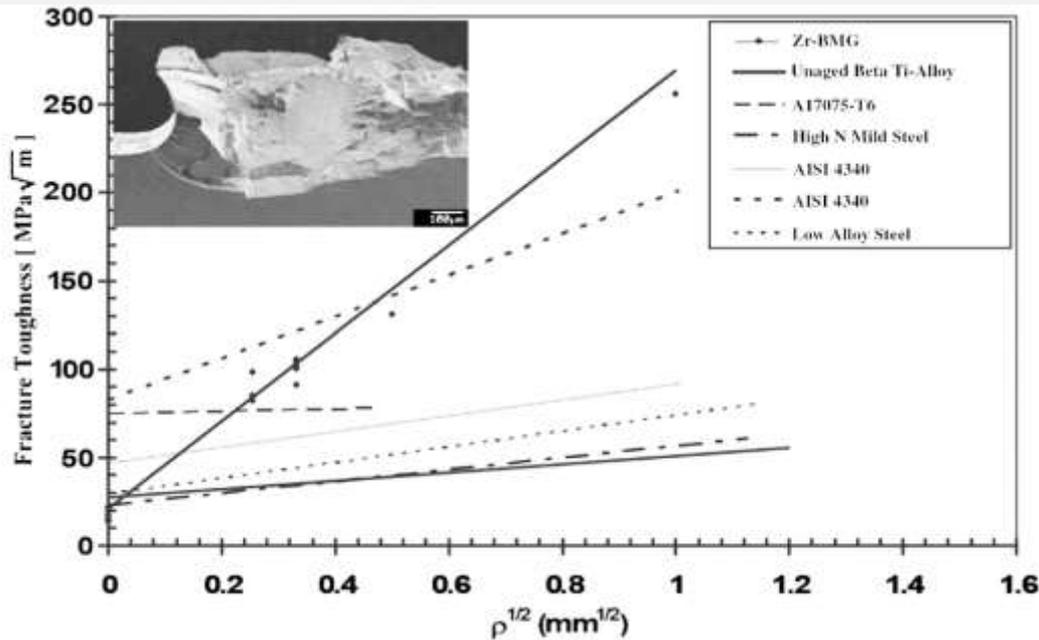
100 μm

mechanical testing: i) notch sensitivity



mechanical testing: i) notch sensitivity

- increased notch root radius, $\rho \uparrow \Rightarrow$ **artificially** elevates toughness, $K_Q \uparrow$
- notch root sensitivity **varies for different compositions**
- effect possibly more pronounced than in crystalline materials
- K_Q values of samples tested with **same ρ comparable**



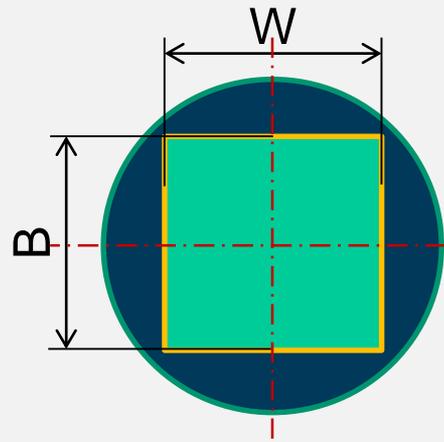
Current issues with metallic glasses

Major problem: **VARIABILITY** of mechanical properties

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 - **sample size**

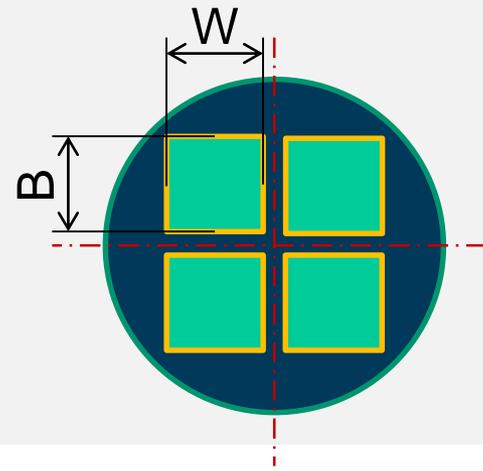
mechanical testing: ii) sample size

SE(B) *without* pre-crack:

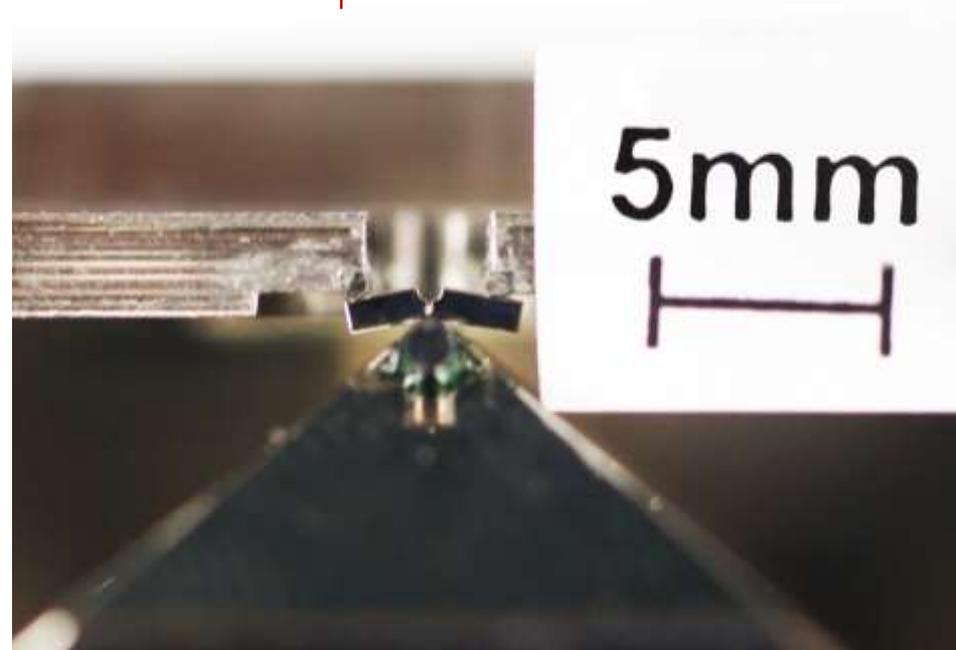
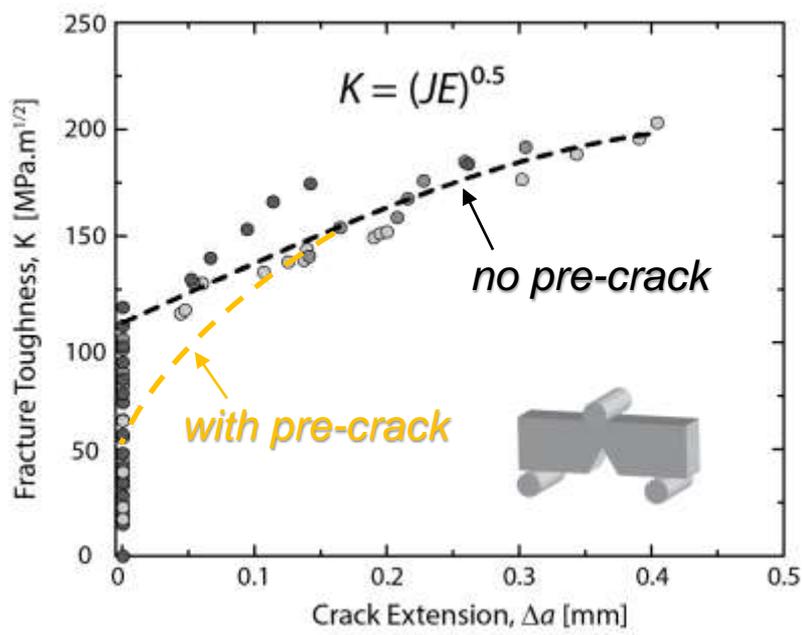


S ~ 15 mm
 B ~ 2.1 mm
 W ~ 2.1 mm
 a ~ 0.5 W
 → b ~ 1 mm

SE(B) *with* pre-crack:

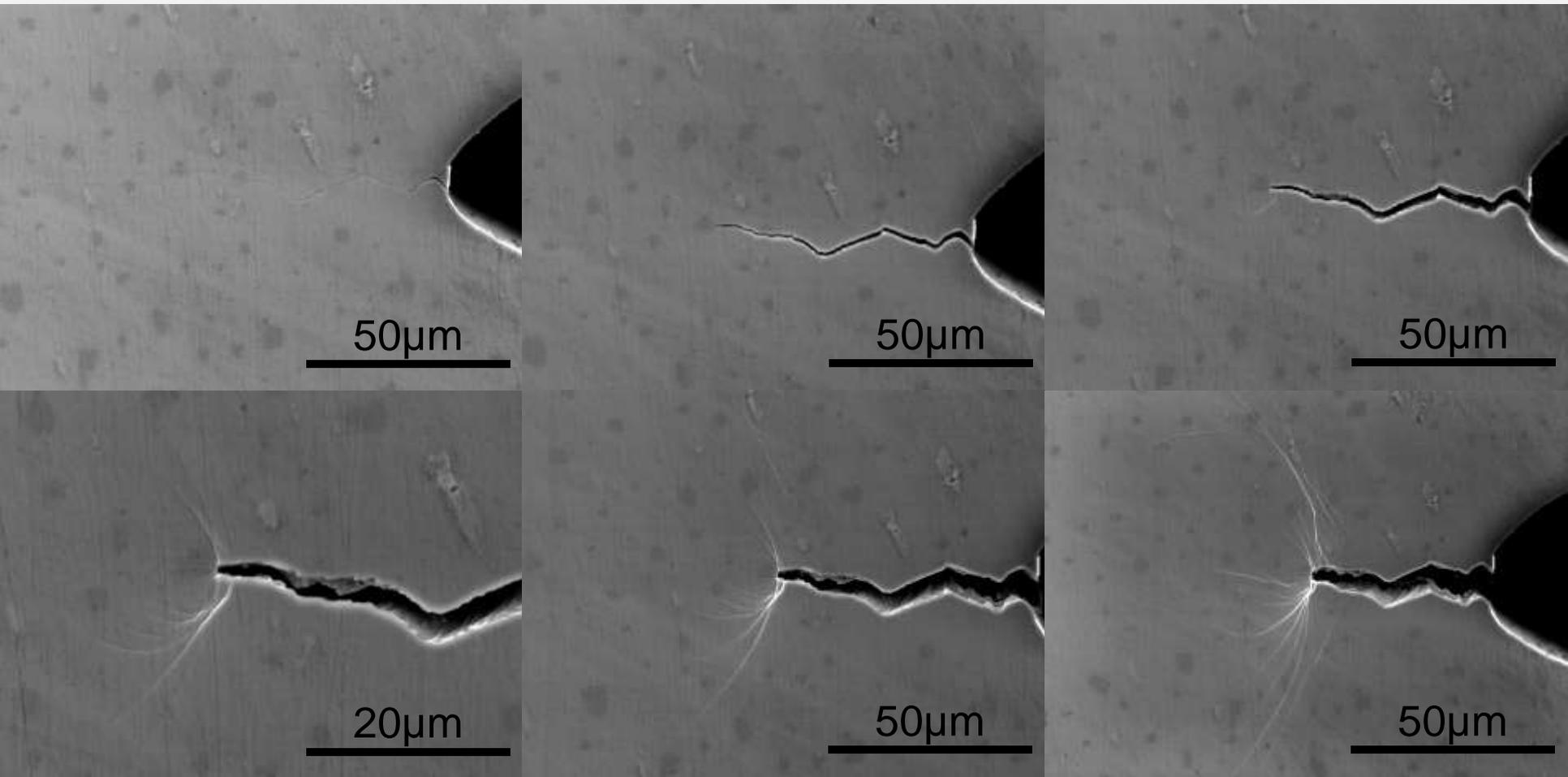


S ~ 2.9 mm
 B ~ 0.7 mm
 W ~ 0.7 mm
 a ~ 0.5 W
 → b ~ 0.35 mm

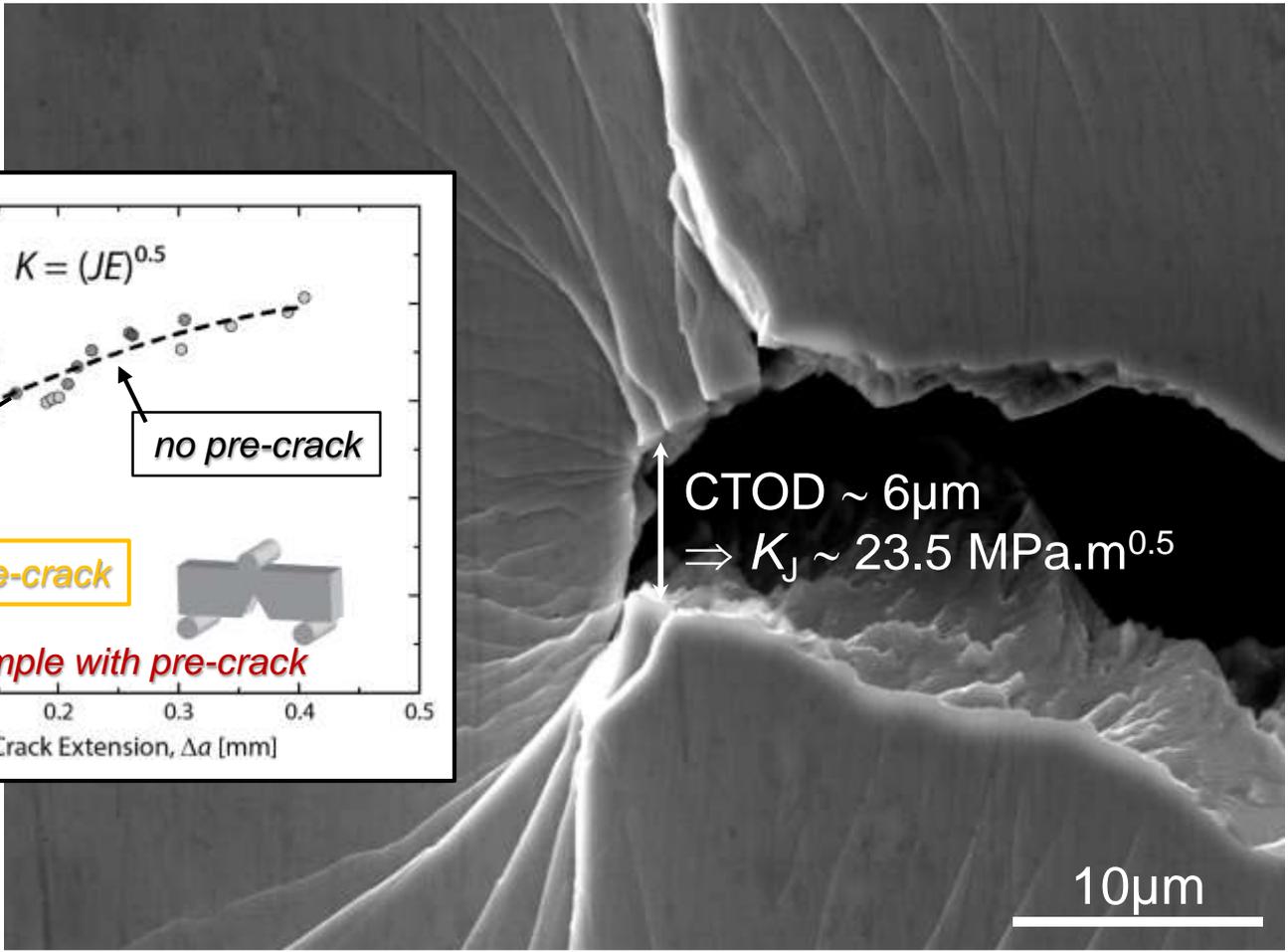


SE(B) *with* pre-crack:

(S ~ 8 mm, B ~ 2.1 mm, W ~ 2.1 mm, a ~ 0.5 W, *b ~ 1mm*)

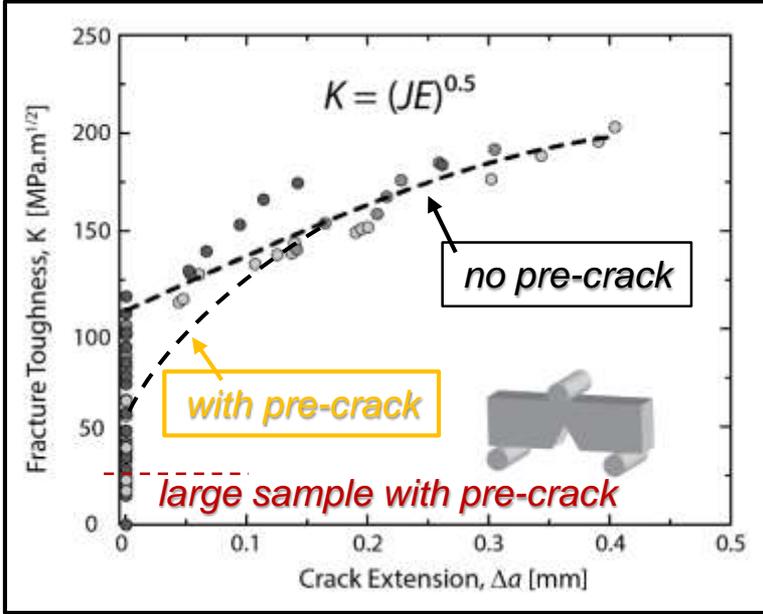


mechanical testing: ii) sample size



CTOD ~ 6µm
⇒ $K_J \sim 23.5 \text{ MPa}\cdot\text{m}^{0.5}$

1mm



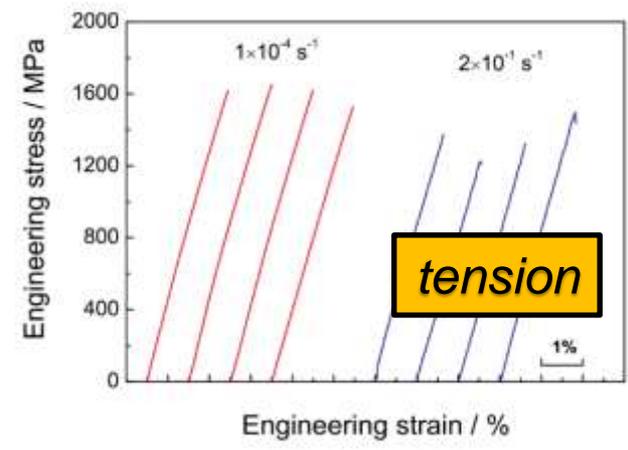
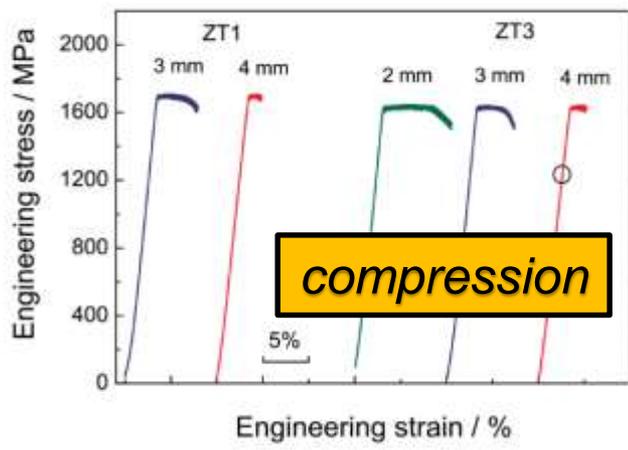
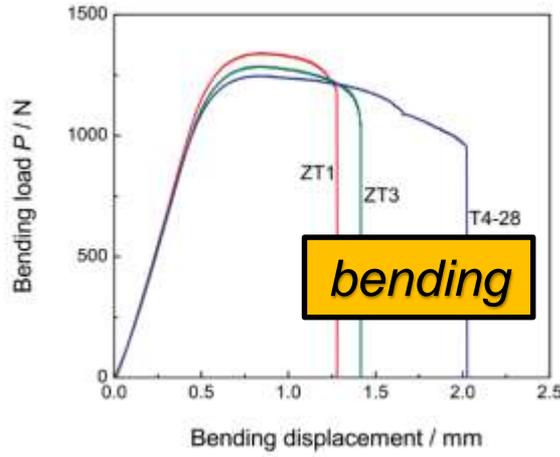
Current issues with metallic glasses

Major problem: **VARIABILITY** of mechanical properties

- i) Processing: - 'structure'-property relation
- ii) **Mechanical testing:** - notch sensitivity
 - sample size
 - **loading condition**

mechanical testing: iii) loading condition

BMGs show major differences when being loaded in ...



SE(B) & SE(T) samples:

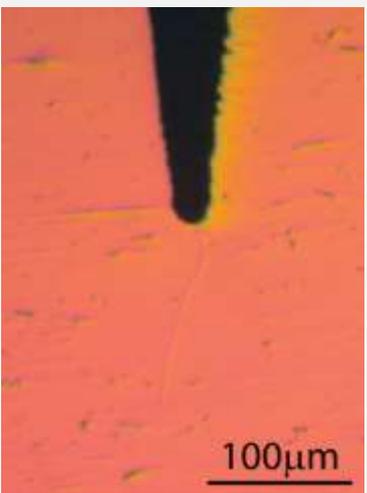
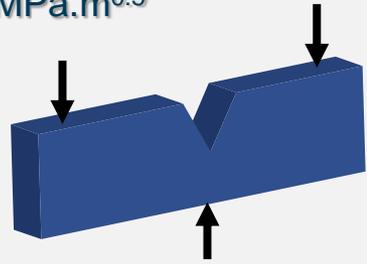
$B \sim 3.5 \text{ mm} / W \sim 7 \text{ mm} / a \sim 0.5 W$

pre-cracking in bending:

$\Delta K = 5.5 - 7 \text{ MPa.m}^{0.5}$

$R = 0.1$

$f = 25 \text{ Hz}$



He, Xu, *J Mater Sci Technol*, 2012

$\text{Zr}_{56}\text{Ni}_{25}\text{Al}_{15}\text{Nb}_4$ (ZKAN) - brittle

| BENDING | TENSION |
|-------------------------------|-------------------------------|
| $\sim 48 \text{ MPa.m}^{0.5}$ | $\sim 83 \text{ MPa.m}^{0.5}$ |

$\text{Zr}_{52.5}\text{Cu}_{17.9}\text{Ni}_{14.6}\text{Al}_{10}\text{Ti}_5$ (Vit105) - medium

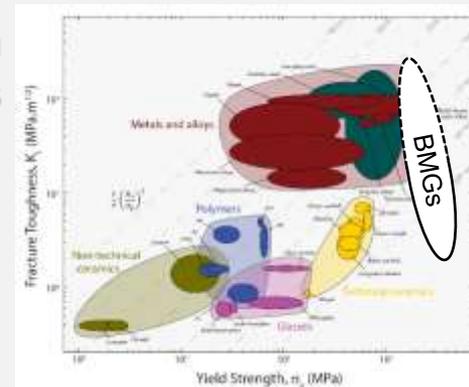
| BENDING | TENSION |
|--------------------------------|-------------------------------|
| $\sim 59+ \text{ MPa.m}^{0.5}$ | $\sim 88 \text{ MPa.m}^{0.5}$ |

$\text{Zr}_{61}\text{Ti}_2\text{Cu}_{25}\text{Al}_{12}$ (ZT1) - ductile

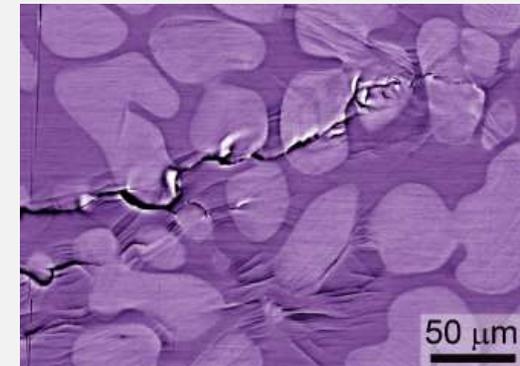
| BENDING | TENSION |
|--------------------------------|--------------------------------|
| $\sim 64+ \text{ MPa.m}^{0.5}$ | $\sim 129 \text{ MPa.m}^{0.5}$ |

? K_{Ic} (bending) > K_{Ic} (tension) ?

Metallic glasses show good combination of strength and toughness

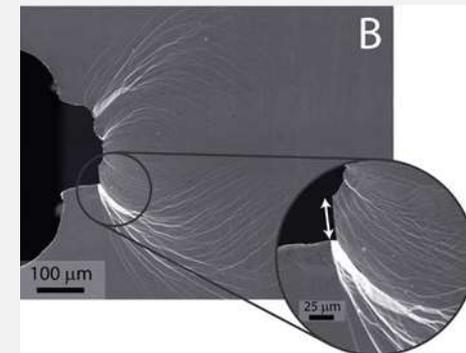


- BMGs ... fascinating class of potential structural materials with a decent combination of mechanical properties:
 - low stiffness & high strength (0.5 – 4 GPa)
 - reasonable fracture toughness (10 – 100 MPa·m^{0.5})
 - ‘OK’ fatigue strength (0.05 – 0.2 σ_{UTS})
- Some glasses/glass-like materials can significantly enhance the fracture/fatigue properties through ductility from “plasticity” via multiple shear-band formation:



- **Glass-composite alloys with second phase dendrites**
 - **ARRESTING** shear bands before they can form cracks
 - microstructural length-scales, $\lambda \leftrightarrow a_c$, mechanical length-scales
- **Fully amorphous glasses**
 - **FORMATION & PROLIFERATION** of multiple shear bands ... **BEFORE** crack initiation and propagation

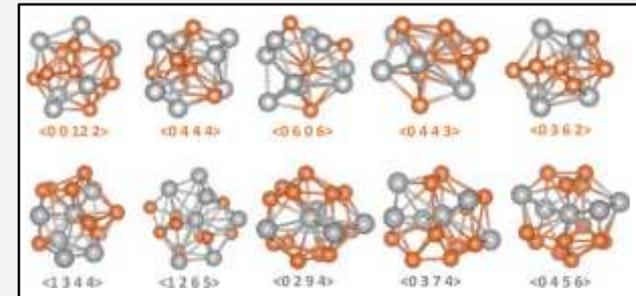
⇒ **excellent toughness & fatigue behavior!**



⇒ Major problem: Variability of mechanical properties

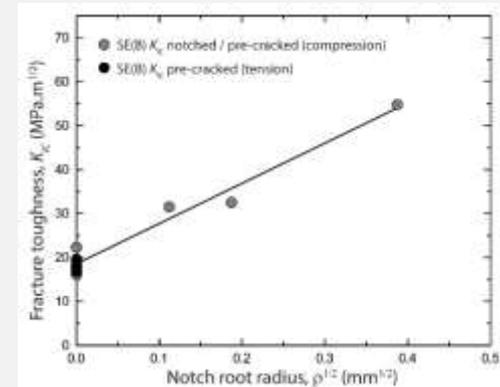
Processing

- **structure-property relationship**
 - local atomic packing structure / quasi-localized soft spots strongly influence shear deformation



Mechanical testing ...

- **notch root radius / flaw sensitivity**
 - larger notch root radii artificially increase the fracture toughness
 - notch root sensitivity varies for different compositions
- **sample size**
 - smaller samples seem to behave more 'plastic' (K_{Ic} , K_{JIc} , $K_Q \uparrow$) & show stable crack propagation rather than catastrophic failure
- **loading condition**
 - no obvious bending ductility
 - large variation in the results of both bending and tension
 - trend to higher numbers in tension



NEED to understand variability of mechanical properties in BMGs to be able to use them in structural applications



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Fracture & Fatigue of Multi-Component Alloys

***Fracture toughness of the CrCoFeMnNi high-entropy alloy
at cryogenic temperatures***

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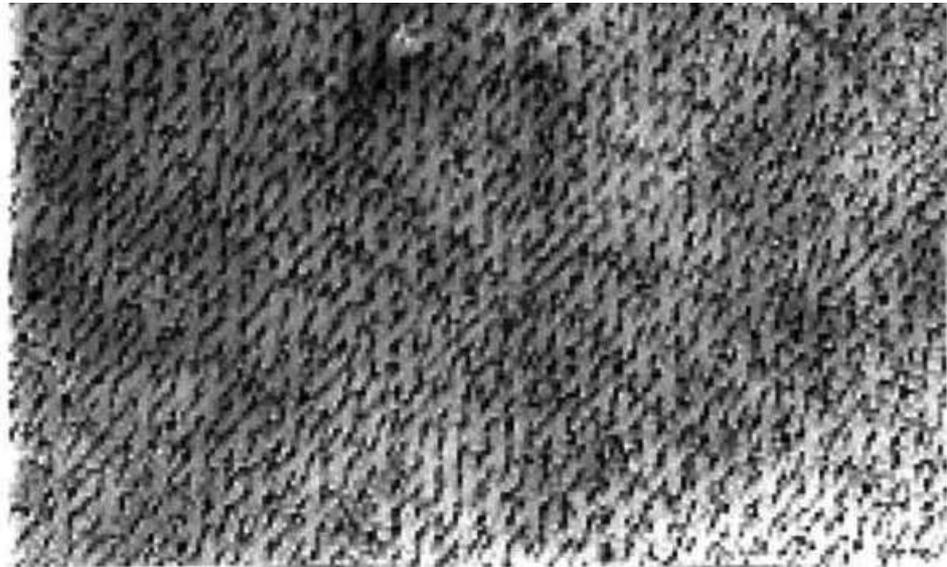
Anton Hohenwarter
University of Leoben, Austria



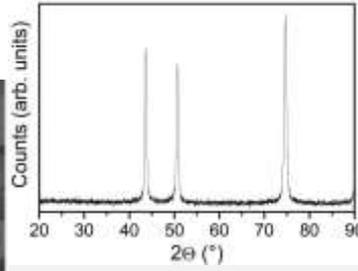
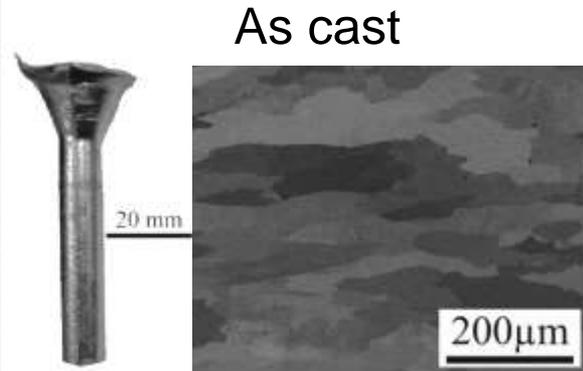
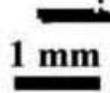
Work supported by the Office of Science (Basic Energy Sciences) of the Department of Energy (Berkeley, Oakridge)
and by the National Science Foundation (Caltech).

A new class of equiatomic alloys ...

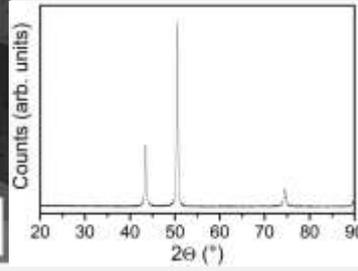
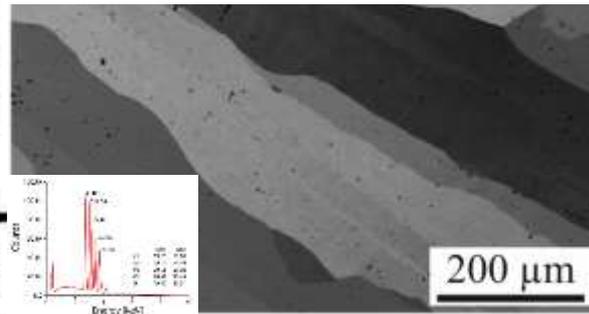
Cast dendritic structure but single-phase FCC



First identified 2004 by Cantor et al.
CrCoFeMnNi



Annealed 72h @ 1000C



Stable single-phase FCC solid solution

Arc-melted and drop-cast alloy is indeed single phase FCC

Why HEA?

| | | | | | |
|-------------------|-----|-----|-----|-----|-----|
| Element | Ni | Fe | Cr | Co | Mn |
| Crystal structure | fcc | bcc | bcc | hcp | A12 |

⇒ Hume-Rothery rules for solid solubility do **NOT** apply

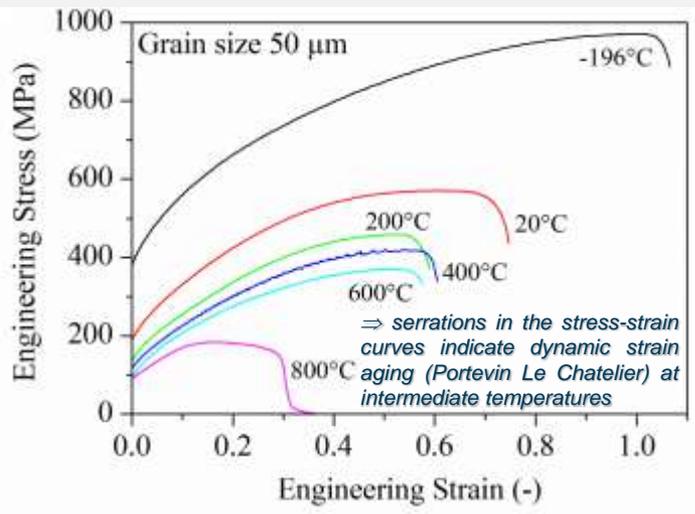
⇒ **Configurational entropy** can stabilize solid solutions (relative to compound/precipitate formation)

⇒ **High-entropy alloys** (number of elements ≥ 5)

$$\Delta G = \Delta H - T\Delta S$$

Cantor et al., Mater. Sci Eng. 2004
Yeh et al., Adv. Eng. Mater. 2004
Otto et al., Acta Mater. 2013

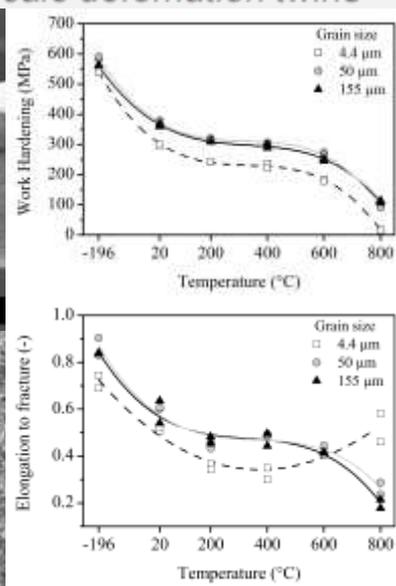
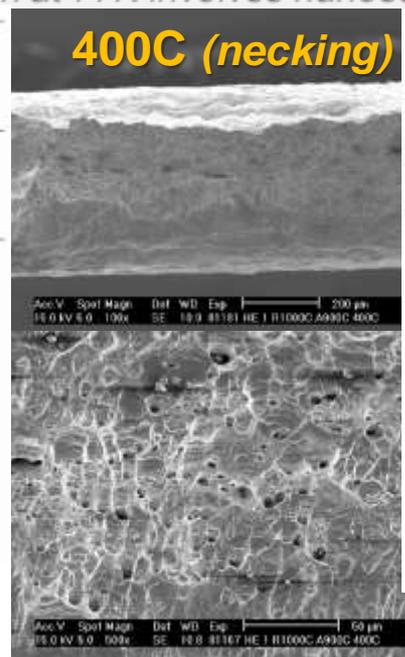
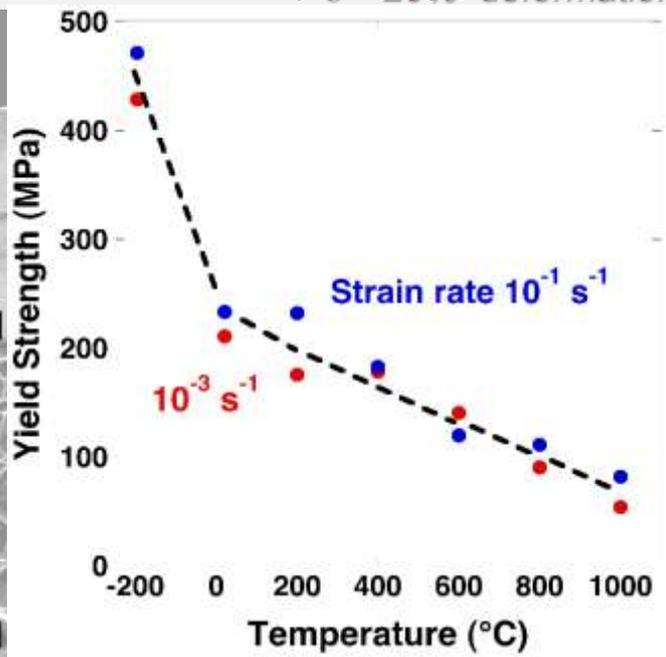
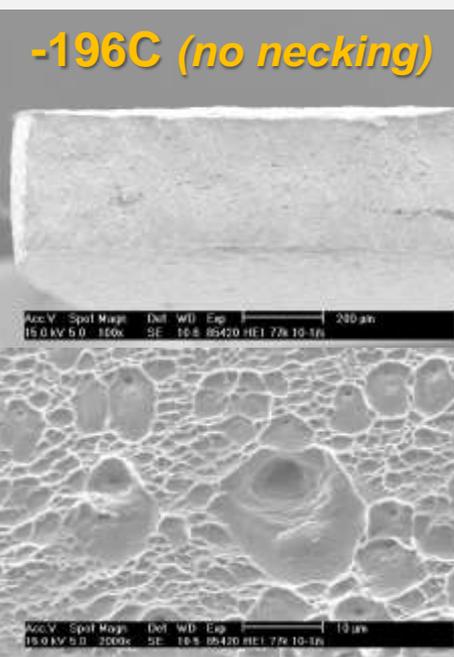
Tensile stress-strain behavior



- \Rightarrow **strong temperature** dependence of strength and ductility
- \Rightarrow **highest ductilities at -196C** (likely due to prevention of necking)
- \Rightarrow degree of **work hardening** ($\sigma_u - \sigma_y$) **highest at -196C**
- \Rightarrow σ_y approx doubles from RT to -196C (**thermally activated yielding**)
- \Rightarrow **σ_y insensitive to strain rate** (unusual for thermally activated yield)

Deformation:

- \Rightarrow $\epsilon < 2\%$: deformation by **planar slip** on $\{111\}\langle 110 \rangle$ (**ALL** temperatures)
- \Rightarrow $\epsilon > 20\%$: deformation at 77K involves **nanoscale deformation twins**



Microstructure, elastic & mechanical properties

Microstructure:

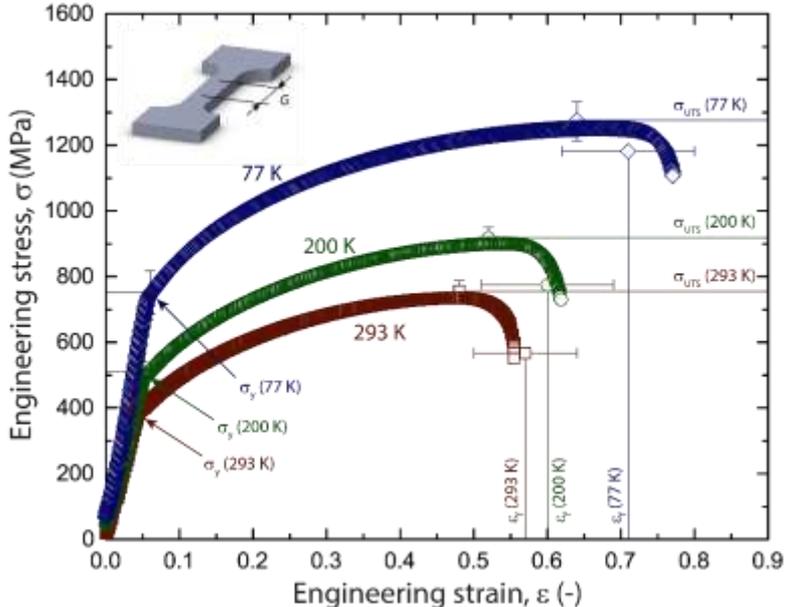
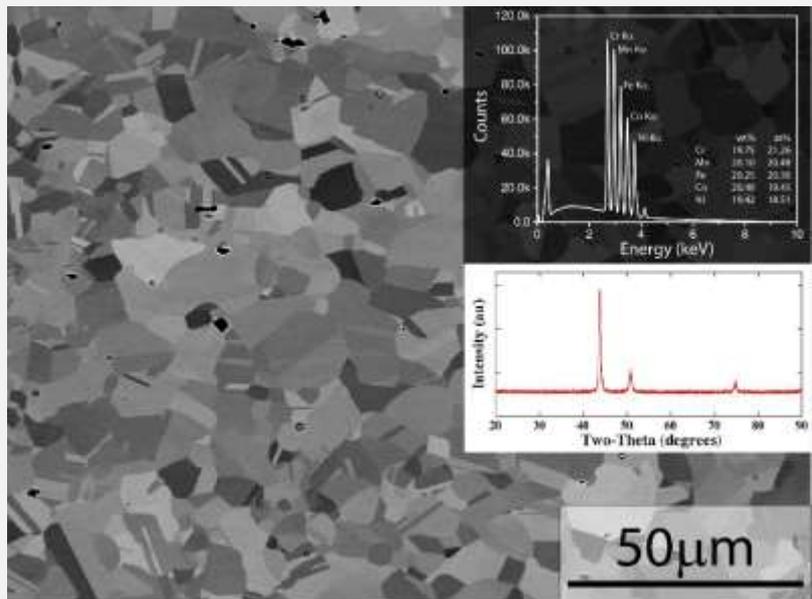
- equiatomic, single-phase material
- equiaxed grains
- grain size ~ 6 μm
- numerous recrystallization twins

Elastic properties:

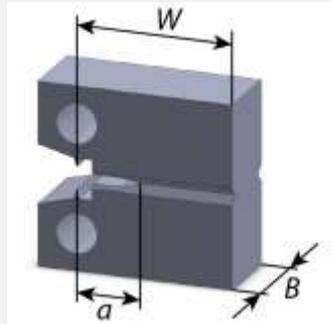
| temp (K) | E (GPa) | ν |
|----------|---------|-------|
| 77 | 214.5 | 0.256 |
| 200 | 209 | 0.263 |
| 293 | 202 | 0.267 |

Mechanical properties:

| temp (K) | σ_y (MPa) | σ_{UTS} (MPa) | ϵ_f | n (0.06-0.15) |
|----------|--------------------------------|------------------------------------|----------------------------------|---------------|
| 77 | ↑ 85% increase in σ_y ↑ | ↑ 70% increase in σ_{UTS} ↑ | ↑ 25% increase in ϵ_f ↑ | 0.36 |
| 200 | | | | 0.41 |
| 293 | | | | 0.41 |

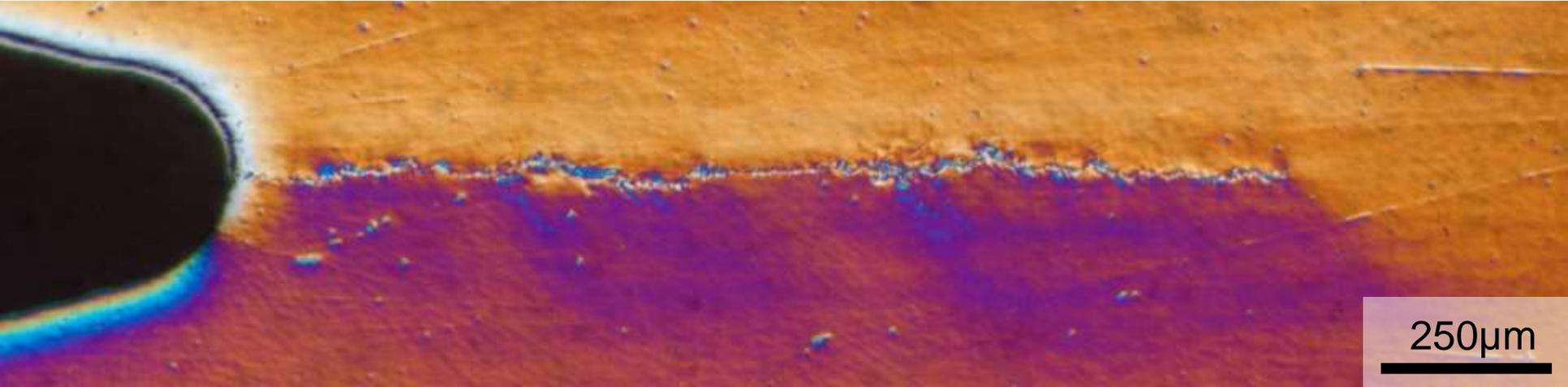
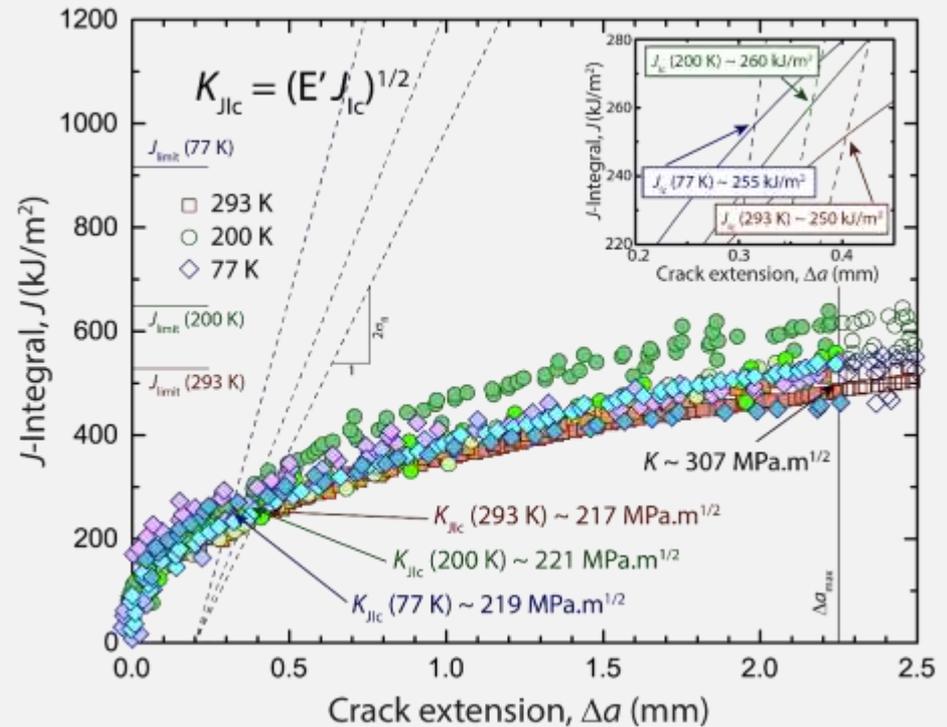


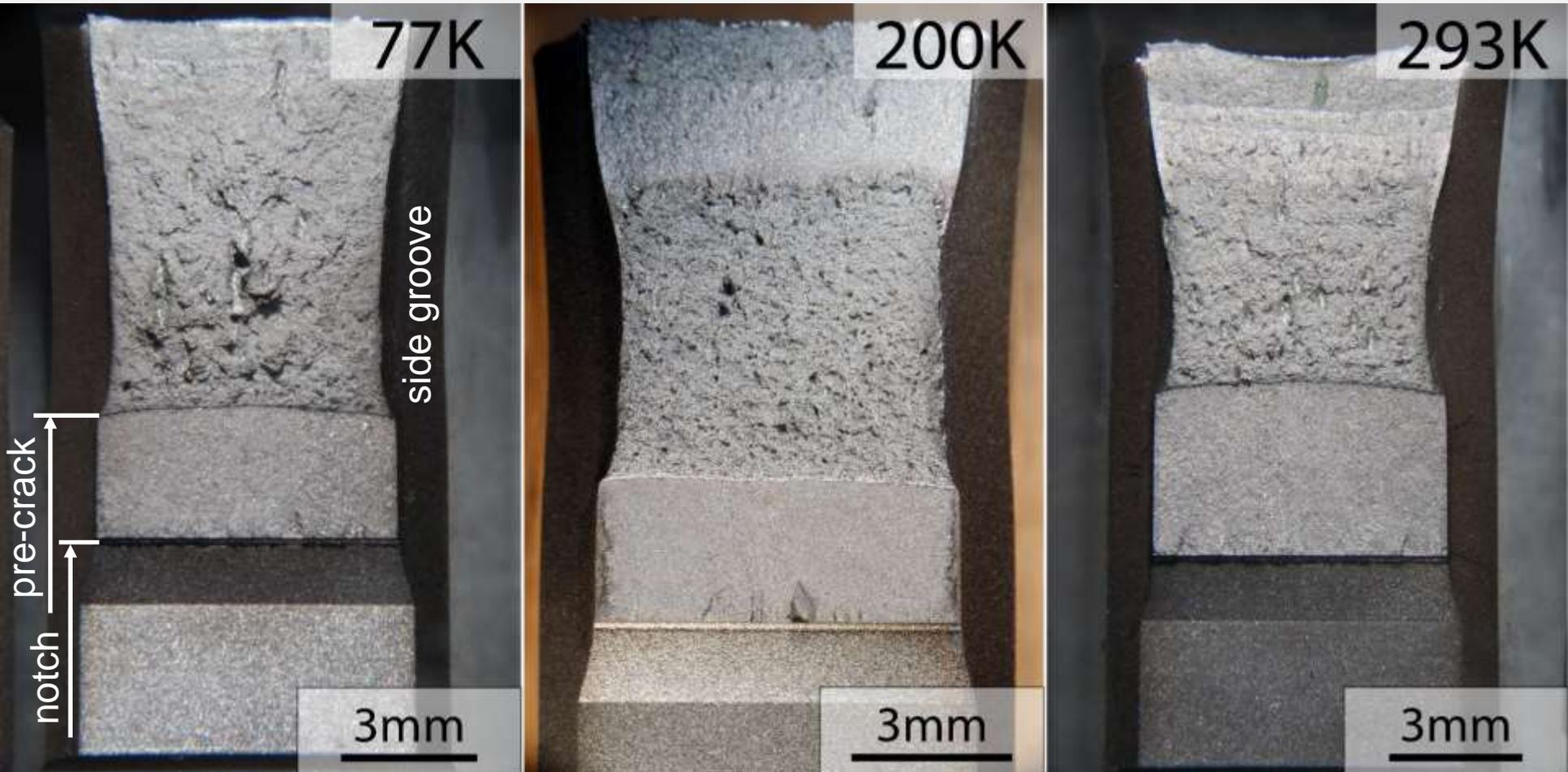
Fracture toughness measurements



C(T)-samples:
 $W = 18 \text{ mm}$
 $B \sim 9 \text{ mm} / B_N \sim 7 \text{ mm}$

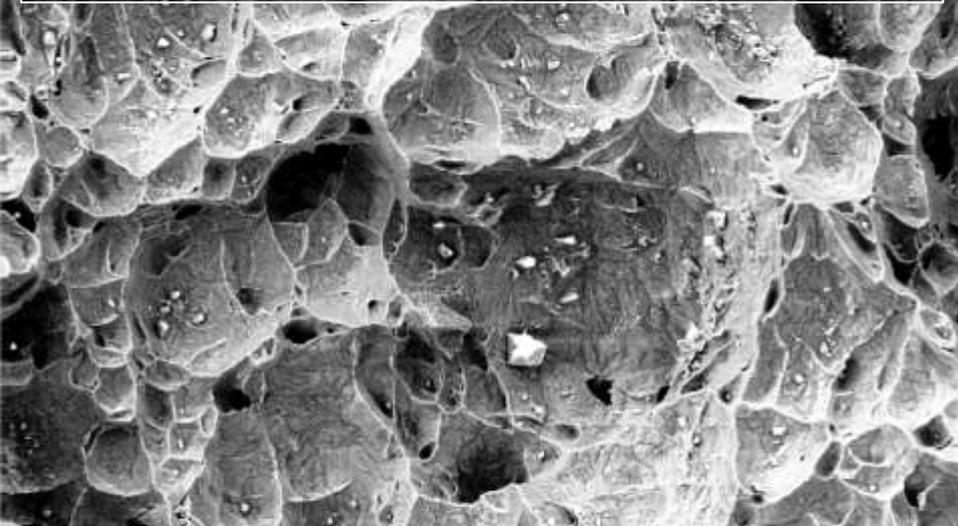
- samples machined by EDM
- surfaces polished using SiC-paper
- pre-cracked in tension
($R = 0.1, \Delta K = 12 - 13 \text{ MPa}\cdot\text{m}^{1/2}$)
- side-grooved by EDM
- tested at 293K, ~200K & ~77K



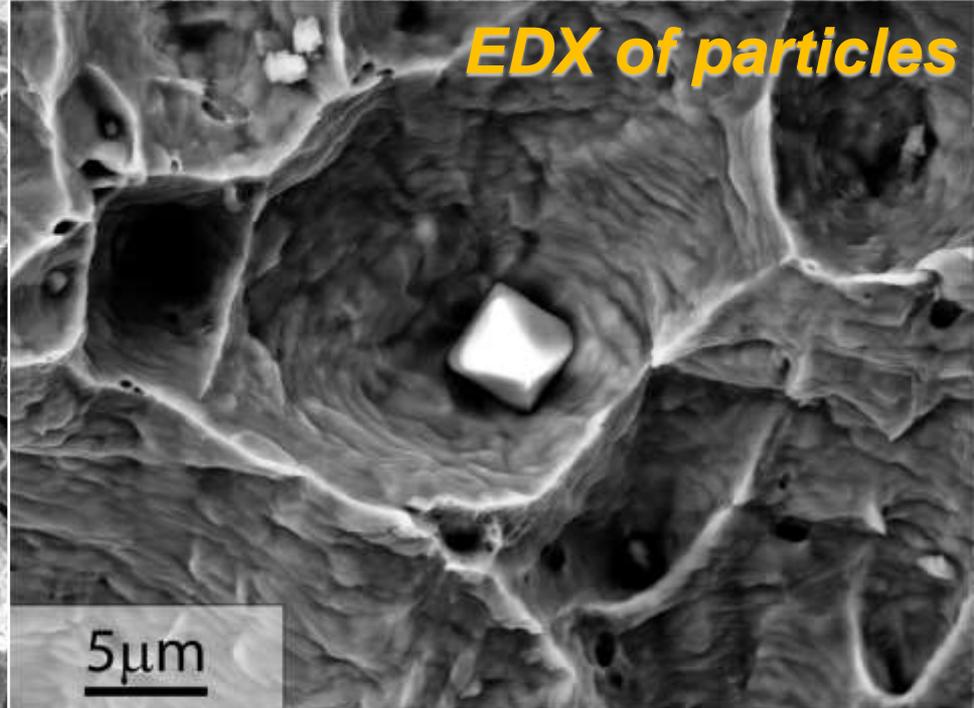


- after the last loading/unloading cycle, one sample of each temperature was fatigued to failure in order to investigate the fracture surfaces

- average particle size $\sim 1.6 \mu\text{m}$
- average particle spacing $\sim 49.6 \mu\text{m}$



EDX of particles

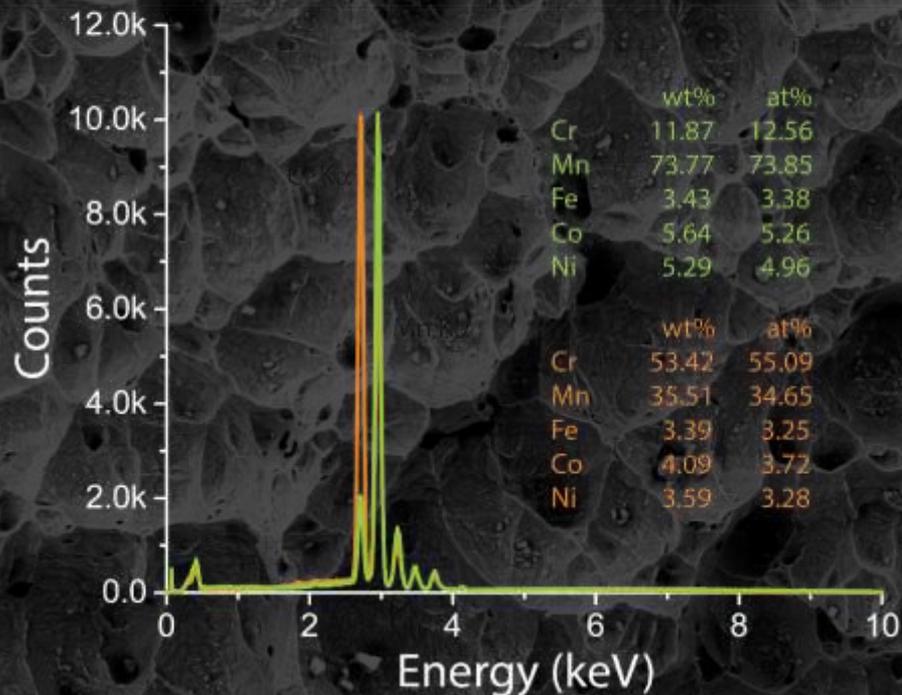


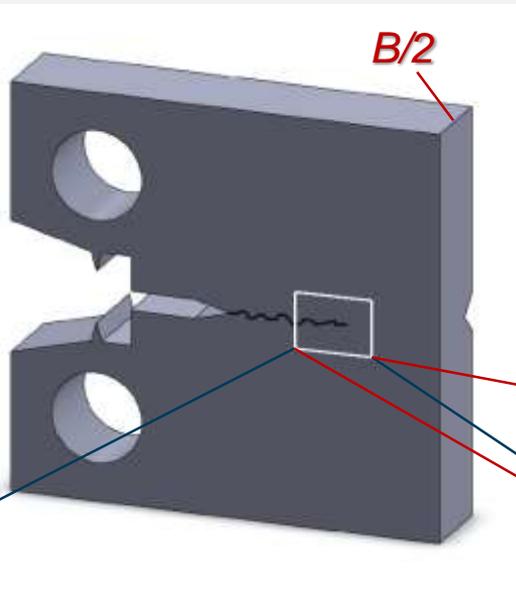
5 μm

- particles are likely oxides
- most particles:
Cr-rich (50% +) with $\sim 35\%$ Mn
(+ small amounts of Fe, Co, Ni)
- some particles:
up to 75% Mn, 12% Cr
(+ minor conc of Fe, Co and Ni)
- particles **NOT** seen in CoCrFeNi alloy



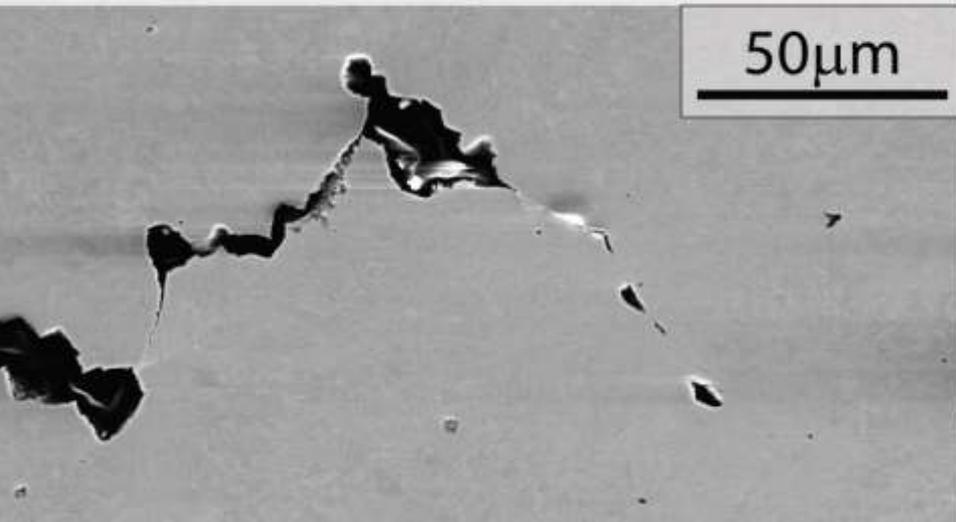
100 μm



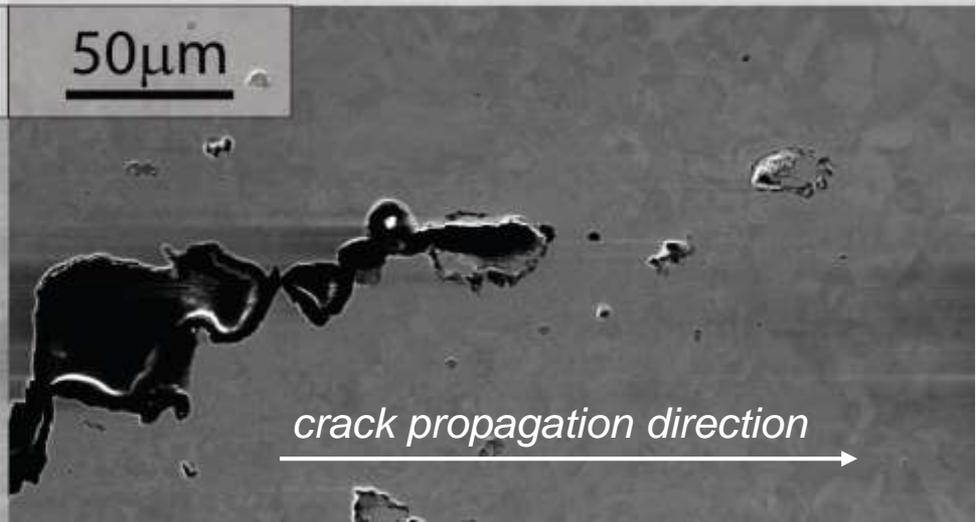


- some samples (293K / 77K) sliced in halves
- each half embedded in conductive / non-conductive resin
- both halves examined using secondary electron (SE) and back-scattered electron (BSE) modes in the SEM

liquid nitrogen temperature (77 K)

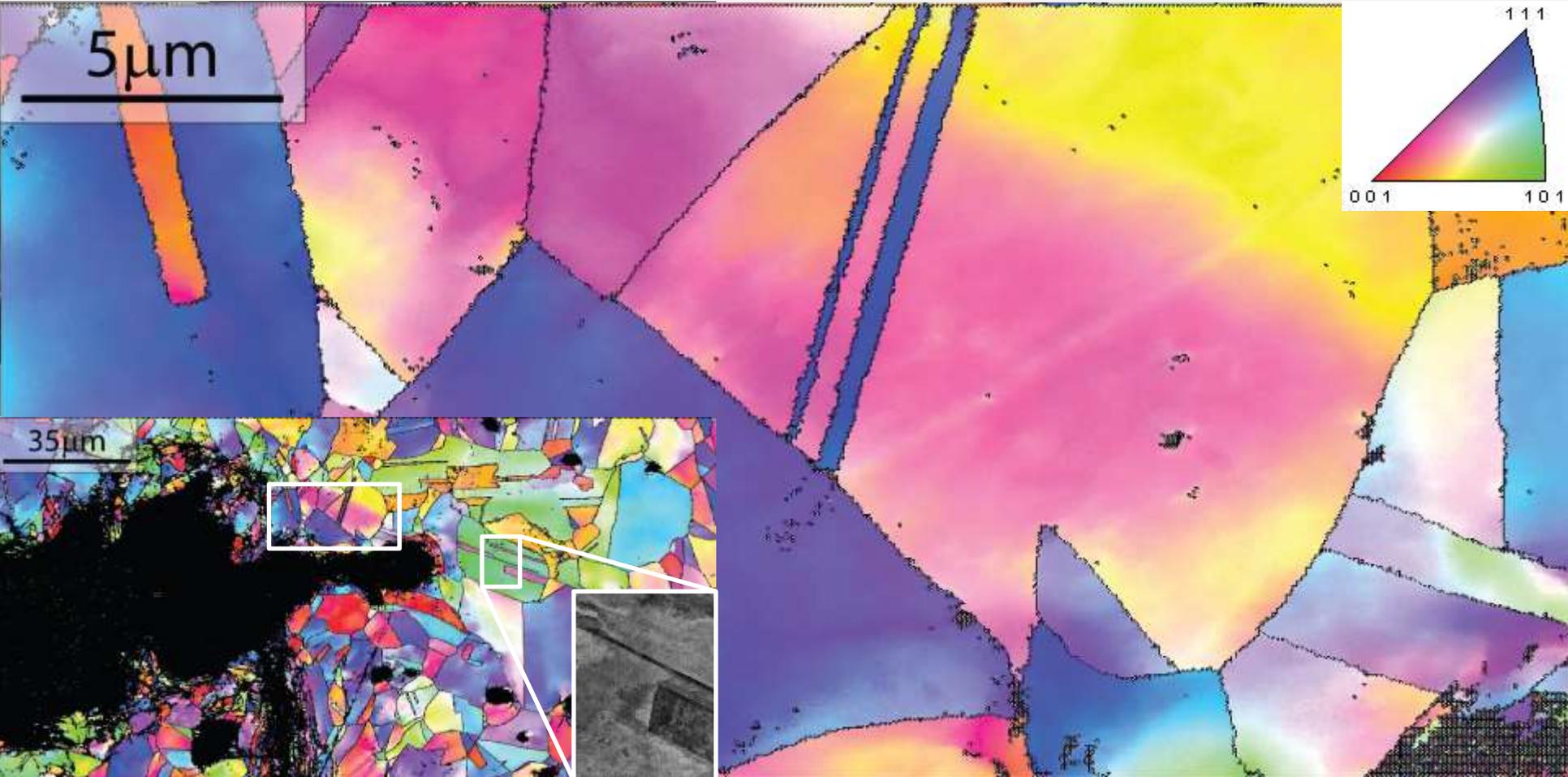


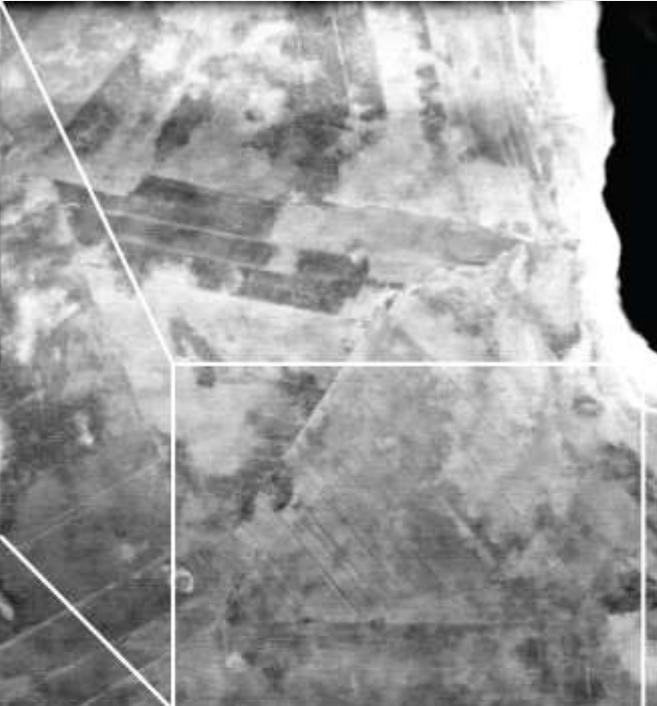
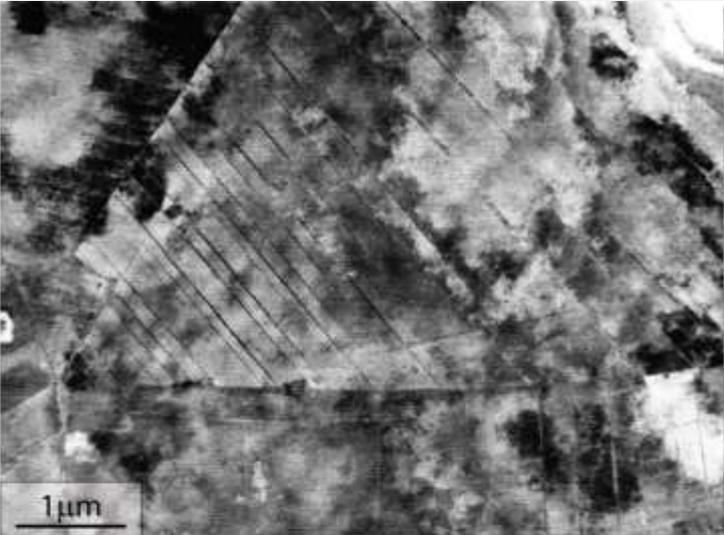
room temperature (293 K)



EBSD & BSE at the interior of the crack tip @ 293K

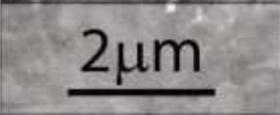
- deformation mainly by dislocation motion (planar slip) → grain misorientations
- no evidence of pronounced deformation-induced nano-twinning
- mainly annealing/recrystallization twins visible





Significant plastic deformation trough:

- pronounced cell structures
- significant dislocation activity
- extensive deformation-induced nano-twinning



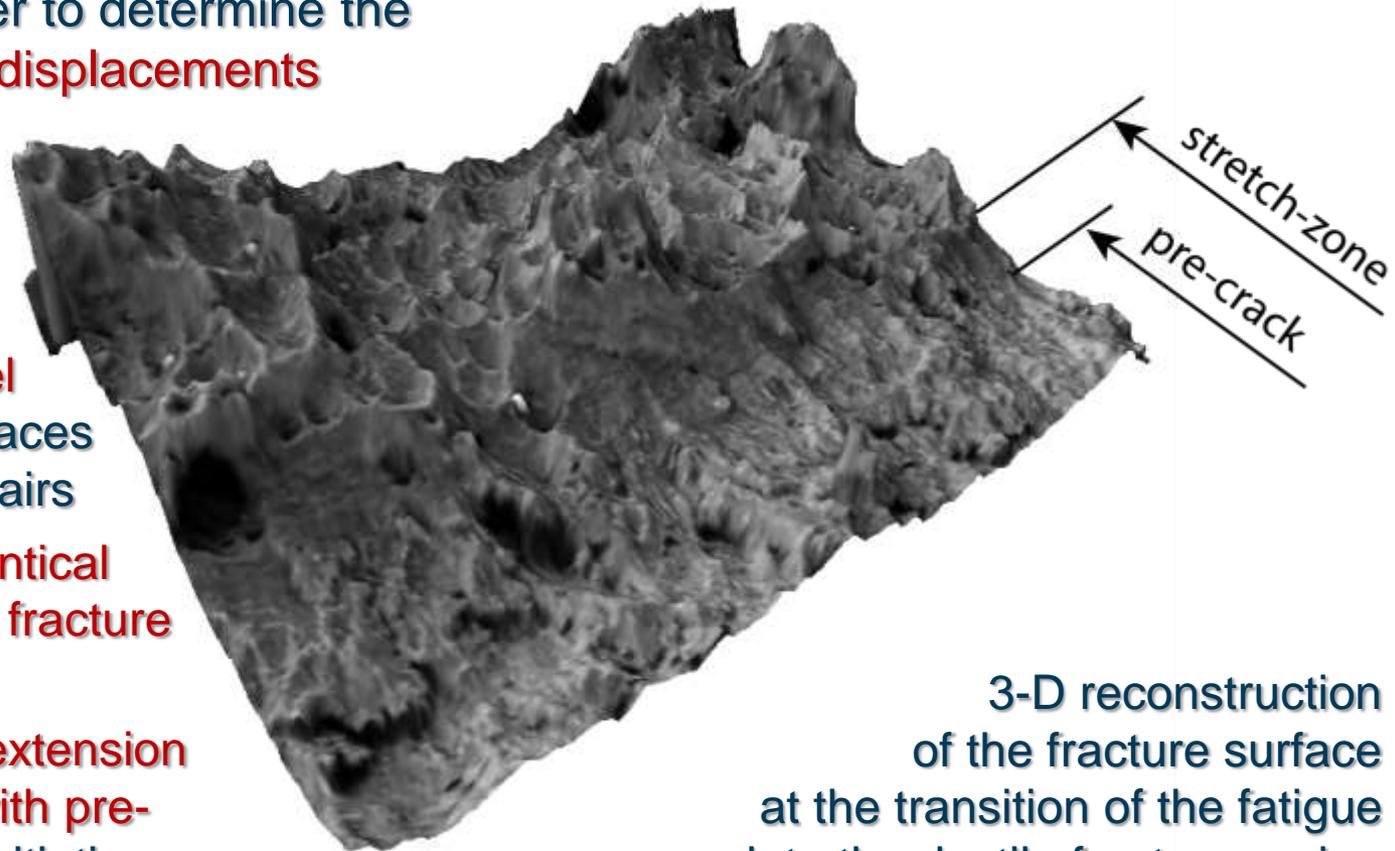
Stereophotogrammetry

- an independent assessment of the toughness can be achieved from the fracture surface
- crack-initiation toughness K_i can also be determined using *stereophotogrammetry* in the SEM in order to determine the crack-tip opening displacements (CTOD) at the onset of crack extension
- digital surface model of both fracture surfaces from SEM images pairs
- determination of identical crack paths on both fracture surfaces
- first physical crack extension from coalescence with pre-crack → $CTOD_i$ at initiation

Global K_{JIC} measurements:

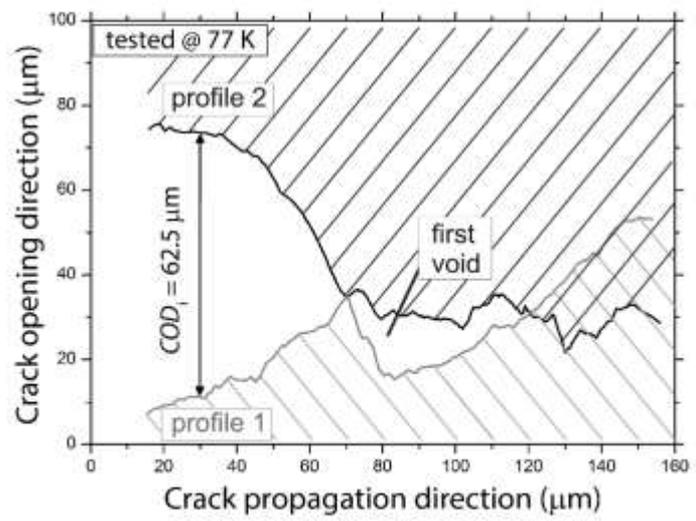
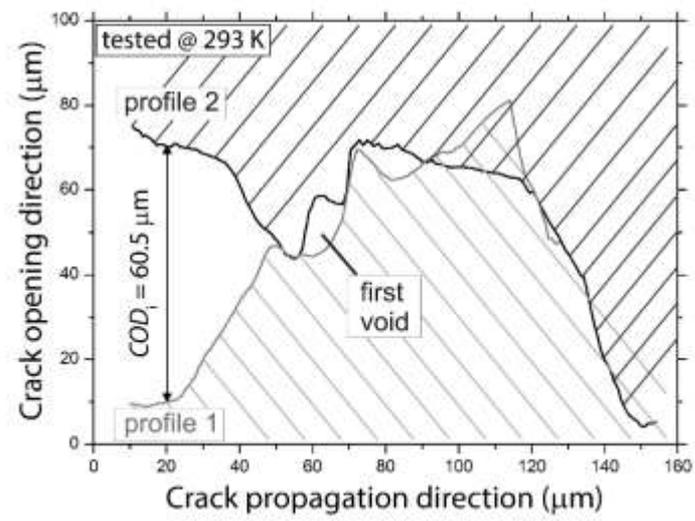
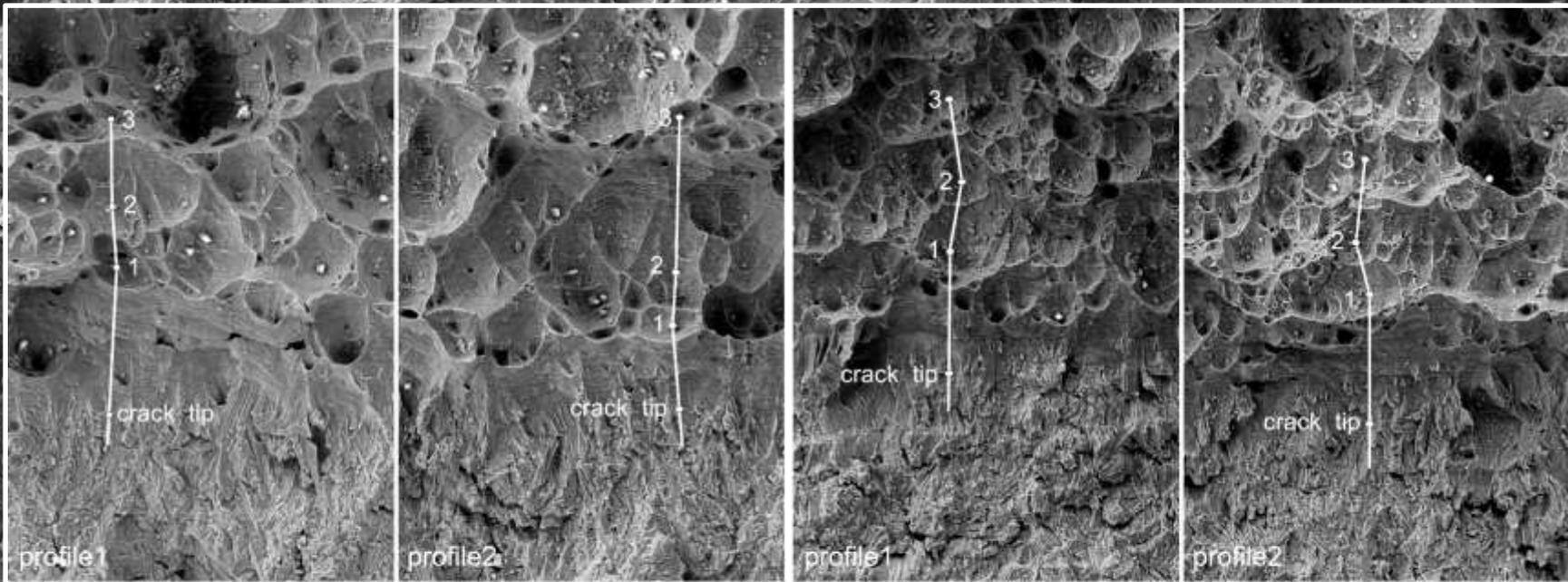
K_{JIC} (293K) ~ 217 MPa√m

K_{JIC} (77K) ~ 219 MPa√m



3-D reconstruction of the fracture surface at the transition of the fatigue pre-crack to the ductile fracture region

side 2 (position 1) @ 77K



Profile 1
31.5 μm

32 μm

32 μm

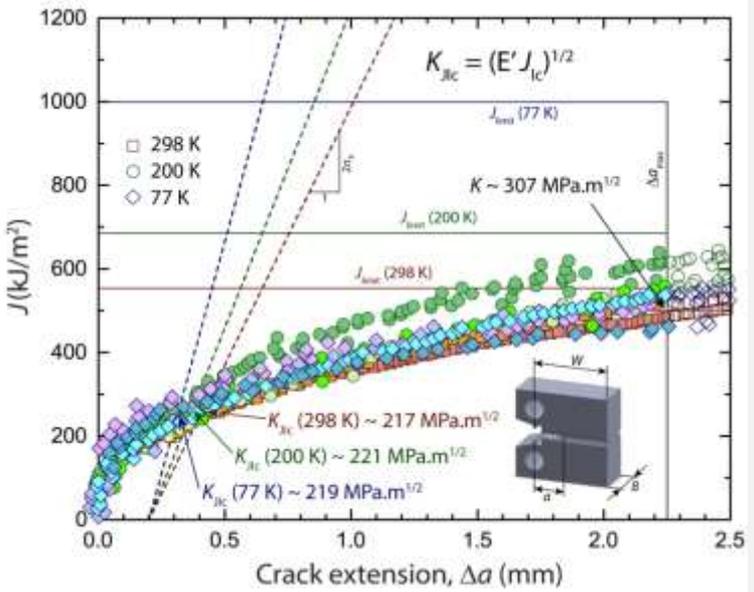
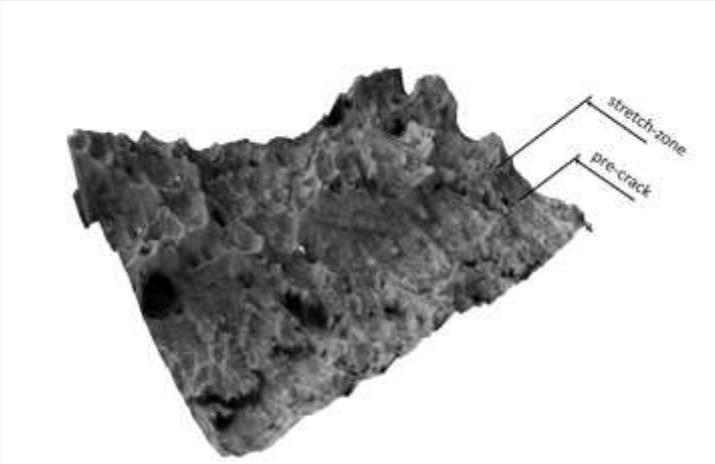
59.4 μm

Fracture toughness of CoCrFeMnNi

using ASTM E1820 & stereophotogrammetry

$$J_i = \frac{1}{d_n} \sigma_0 COD_i$$

$$J_{IC} = K_{JIC}^2 / E'$$



CTOD_i

σ_y

σ_{UTS}

J_i

K_i (Δa → 0)

K_{JIC} (Δa = 200 μm)

K_{SS} (stable crack growth)

293K

57 ± 19 μm

410 MPa

763 MPa

195 kJ/m²

191 MPa.m^{1/2}

217 MPa.m^{1/2}

>300 MPa.m^{1/2}

77K

49 ± 13 μm

759 MPa

1280 MPa

219 kJ/m²

203 MPa.m^{1/2}

219 MPa.m^{1/2}

>300 MPa.m^{1/2}

Ashby map (strength vs. toughness) & Conclusions

- High-Entropy Alloys

→ new aspect of metallurgy in the quest for new materials with interesting properties

- CoCrFeMnNi single-phase fcc alloy

→ excellent damage-tolerance properties which don't degrade at cryogenic temperatures

- @ 293K: $\sigma_{UTS} \sim 763 \text{ MPa}$
 $K_{JIC} = 217 \text{ MPa}\cdot\text{m}^{1/2}$ ($K_I = 191 \text{ MPa}\cdot\text{m}^{1/2}$)
 deformation by *planar dislocation slip*

- @ 77K: $\sigma_{UTS} \sim 1280 \text{ MPa}$
 $K_{JIC} = 219 \text{ MPa}\cdot\text{m}^{1/2}$ ($K_I = 203 \text{ MPa}\cdot\text{m}^{1/2}$)
 deformation associated with *nanoscale deformation twinning*

- Toughness → associated with continuous steady hardening ($n \sim 0.4$)
 → suppressing plastic instability & localization
 → appears to be a characteristic of plastic deformation by twinning

- **!!! $\sigma_{UTS} \sim 1.3 \text{ GPa}$ & $K_{JIC} \sim 220 \text{ MPa}\sqrt{\text{m}}$ @ 77K → extremely damage-tolerant !!!**
(properties exceed those of many materials including many austenitic stainless steels)

